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ORGANIZATION (CAMIRO) EXPLORATION DIVISION**

**CAMIRO PROJECT 10E01
Phase 1**

**Quality Control Assessment of Portable XRF Analyzers:
Development of Standard Operating Procedures,
Performance on Variable Media and Recommended Uses**

APPENDIX 2a

**University of California X-Ray Data Booklet
Saved as independent pdf file (appendix_2a.pdf)**

**Center for X-Ray Optics
and
Advanced Light Source**

**X-RAY DATA
BOOKLET**

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CENTER FOR X-RAY OPTICS
ADVANCED LIGHT SOURCE

X-RAY DATA BOOKLET

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UPDATES AND FURTHER INFORMATION

The content of the *X-Ray Data Booklet*, together with additional information and further references, can be found on the web at <http://xdb.lbl.gov>. This site will also be updated with corrections to any errors found in this edition of the *Booklet*.

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PREFACE

This edition of the *X-Ray Data Booklet* has minor revisions to some sections since the second edition in 2001. The booklet has been very popular. All 12,000 copies of the second edition were distributed several years ago. This edition is being published to provide a new supply of booklets. The booklet was first published in 1985. The *Booklet* is also posted on the web at <http://xdb.lbl.gov>, together with additional detail and further references for many of the sections.

As before, the compilers are grateful to a host of contributors who furnished new material or reviewed and revised their original sections. Also, as in the original edition, many sections draw heavily on work published elsewhere, as indicated in the text and figure captions. I would like to thank Doug Vaughan for his excellent work as co-editor of the second edition. I also want to express my continuing thanks to Janos Kirz, who conceived the *Booklet* as a service to the community and who remains an active contributor to this edition.

I take full responsibility for any errors in this new edition, and I invite readers to bring them to my attention at the Center for X-Ray Optics, 2-400, Lawrence Berkeley National Laboratory, Berkeley, California 94720, or by e-mail at xdb@grace.lbl.gov. Corrections will be posted on the web and incorporated in subsequent printings.

Albert C. Thompson
1 September 2009

SECTION 1

X-RAY PROPERTIES OF THE ELEMENTS

1.1 ELECTRON BINDING ENERGIES

Gwyn P. Williams

Table 1-1 gives the electron binding energies for the elements in their natural forms. The energies are given in electron volts relative to the vacuum level for the rare gases and for H₂, N₂, O₂, F₂, and Cl₂; relative to the Fermi level for the metals; and relative to the top of the valence bands for semiconductors.

Values have been taken from Ref. 1 except as follows:

*Values taken from Ref. 2, with additional corrections

†Values taken from Ref. 3.

^aOne-particle approximation not valid owing to short core-hole lifetime.

^bValue derived from Ref. 1.

Thanks also to R. Johnson, G. Ice, M. Olmstead, P. Dowben, M. Seah, E. Gullikson, F. Boscherini, W. O'Brien, R. Alkire, and others.

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2. M. Cardona and L. Ley, Eds., *Photoemission in Solids I: General Principles* (Springer-Verlag, Berlin, 1978).
3. J. C. Fuggle and N. Mårtensson, "Core-Level Binding Energies in Metals," *J. Electron Spectrosc. Relat. Phenom.* **21**, 275 (1980).

Table I-1. Electron binding energies, in electron volts, for the elements in their natural forms.

Element	K 1s	L ₁ 2s	L ₂ 2p _{1/2}	L ₃ 2p _{3/2}	M ₁ 3s	M ₂ 3p _{1/2}	M ₃ 3p _{3/2}	M ₄ 3d _{3/2}	M ₅ 3d _{5/2}	N ₁ 4s	N ₂ 4p _{1/2}	N ₃ 4p _{3/2}
1 H	13.6											
2 He	24.6*											
3 Li	54.7*											
4 Be	111.5*											
5 B	188*											
6 C	284.2*											
7 N	409.9*	37.3*										
8 O	543.1*	41.6*										
9 F	696.7*											
10 Ne	870.2*	48.5*	21.7*	21.6*								
11 Na	1070.8†	63.5†	30.65	30.81								
12 Mg	1303.0†	88.7	49.78	49.50								
13 Al	1559.6	117.8	72.95	72.55								
14 Si	1839	149.7* ^b	99.82	99.42								
15 P	2145.5	189*	136*	135*								
16 S	2472	230.9	163.6*	162.5*								
17 Cl	2822.4	270*	202*	200*								
18 Ar	3205.9*	326.3*	250.6†	248.4*	29.3*	15.9*	15.7*					
19 K	3608.4*	378.6*	297.3*	294.6*	34.8*	18.3*	18.3*					
20 Ca	4038.5*	438.4†	349.7†	346.2†	44.3 †	25.4†	25.4†					
21 Sc	4492	498.0*	403.6*	398.7*	51.1*	28.3*	28.3*					
22 Ti	4966	560.9†	460.2†	453.8†	58.7†	32.6†	32.6†					

Table 1-1. Electron binding energies (continued).

Element	K 1s	L ₁ 2s	L ₂ 2p _{1/2}	L ₃ 2p _{3/2}	M ₁ 3s	M ₂ 3p _{1/2}	M ₃ 3p _{3/2}	M ₄ 3d _{3/2}	M ₅ 3d _{5/2}	N ₁ 4s	N ₂ 4p _{1/2}	N ₃ 4p _{3/2}
48 Cd	26711	4018	3727	3538	772.0†	652.6†	618.4†	411.9†	405.2†	109.8†	63.9†a	63.9†a
49 In	27940	4238	3938	3730	827.2†	703.2†	665.3†	451.4†	443.9†	122.9†	73.5†a	73.5†a
50 Sn	29200	4465	4156	3929	884.7†	756.5†	714.6†	493.2†	484.9†	137.1†	83.6†a	83.6†a
51 Sb	30491	4698	4380	4132	946†	812.7†	766.4†	537.5†	528.2†	153.2†	95.6†a	95.6†a
52 Te	31814	4939	4612	4341	1006†	870.8†	820.0†	583.4†	573.0†	169.4†	103.3†a	103.3†a
53 I	33169	5188	4852	4557	1072*	931*	875*	630.8	619.3	186*	123*	123*
54 Xe	34561	5453	5107	4786	1148.7*	1002.1*	940.6*	689.0*	676.4*	213.2*	146.7	145.5*
55 Cs	35985	5714	5359	5012	1211*b	1071*	1003*	740.5*	726.6*	232.3*	172.4*	161.3*
56 Ba	37441	5989	5624	5247	1293*b	1137*b	1063*b	795.7†	780.5*	253.5†	192	178.6†
57 La	38925	6266	5891	5483	1362*b	1209*b	1128*b	853*	836*	274.7*	205.8	196.0*
58 Ce	40443	6549	6164	5723	1436*b	1274*b	1187*b	902.4*	883.8*	291.0*	223.2	206.5*
59 Pr	41991	6835	6440	5964	1511	1337	1242	948.3*	928.8*	304.5	236.3	217.6
60 Nd	43569	7126	6722	6208	1575	1403	1297	1003.3*	980.4*	319.2*	243.3	224.6
61 Pm	45184	7428	7013	6459	—	1471	1357	1052	1027	—	242	242
62 Sm	46834	7737	7312	6716	1723	1541	1420	1110.9*	1083.4*	347.2*	265.6	247.4
63 Eu	48519	8052	7617	6977	1800	1614	1481	1158.6*	1127.5*	360	284	257
64 Gd	50239	8376	7930	7243	1881	1688	1544	1221.9*	1189.6*	378.6*	286	271
65 Tb	51996	8708	8252	7514	1968	1768	1611	1276.9*	1241.1*	396.0*	322.4*	284.1*
66 Dy	53789	9046	8581	7790	2047	1842	1676	1333	1292.6*	414.2*	333.5*	293.2*
67 Ho	55618	9394	8918	8071	2128	1923	1741	1392	1351	432.4*	343.5	308.2*
68 Er	57486	9751	9264	8358	2207	2006	1812	1453	1409	449.8*	366.2	320.2*
69 Tm	59390	10116	9617	8648	2307	2090	1885	1515	1468	470.9*	385.9*	332.6*
70 Yb	61332	10486	9978	8944	2398	2173	1950	1576	1528	480.5*	388.7*	339.7*

Table I-1. Electron binding energies (continued).

Element	N ₄ 4d _{3/2}	N ₅ 4d _{5/2}	N ₆ 4f _{5/2}	N ₇ 4f _{7/2}	O ₁ 5s	O ₂ 5p _{1/2}	O ₃ 5p _{3/2}	O ₄ 5d _{3/2}	O ₅ 5d _{5/2}	P ₁ 6s	P ₂ 6p _{1/2}	P ₃ 6p _{3/2}
48 Cd	11.7†	10.7†										
49 In	17.7†	16.9†										
50 Sn	24.9†	23.9†										
51 Sb	33.3†	32.1†										
52 Te	41.9†	40.4†										
53 I	50.6	48.9										
54 Xe	69.5*	67.5*			23.3*	13.4*	12.1*					
55 Cs	79.8*	77.5*			22.7	14.2*	12.1*					
56 Ba	92.6†	89.9†			30.3†	17.0†	14.8†					
57 La	105.3*	102.5*			34.3*	19.3*	16.8*					
58 Ce	109*	—	0.1	0.1	37.8	19.8*	17.0*					
59 Pr	115.1*	115.1*	2.0	2.0	37.4	22.3	22.3					
60 Nd	120.5*	120.5*	1.5	1.5	37.5	21.1	21.1					
61 Pm	120	120	—	—	—	—	—					
62 Sm	129	129	5.2	5.2	37.4	21.3	21.3					
63 Eu	133	127.7*	0	0	32	22	22					
64 Gd	—	142.6*	8.6*	8.6*	36	28	21					
65 Tb	150.5*	150.5*	7.7*	2.4*	45.6*	28.7*	22.6*					
66 Dy	153.6*	153.6*	8.0*	4.3*	49.9*	26.3	26.3					
67 Ho	160*	160*	8.6*	5.2*	49.3*	30.8*	24.1*					
68 Er	167.6*	167.6*	—	4.7*	50.6*	31.4*	24.7*					
69 Tm	175.5*	175.5*	—	4.6	54.7*	31.8*	25.0*					
70 Yb	191.2*	182.4*	2.5*	1.3*	52.0*	30.3*	24.1*					

Table 1-1. Electron binding energies (continued).

Element	K 1s	L ₁ 2s	L ₂ 2p _{1/2}	L ₃ 2p _{3/2}	M ₁ 3s	M ₂ 3p _{1/2}	M ₃ 3p _{3/2}	M ₄ 3d _{3/2}	M ₅ 3d _{5/2}	N ₁ 4s	N ₂ 4p _{1/2}	N ₃ 4p _{3/2}
71 Lu	63314	10870	10349	9244	2491	2264	2024	1639	1589	506.8*	412.4*	359.2*
72 Hf	65351	11271	10739	9561	2601	2365	2108	1716	1662	538*	438.2†	380.7†
73 Ta	67416	11682	11136	9881	2708	2469	2194	1793	1735	563.4†	463.4†	400.9†
74 W	69525	12100	11544	10207	2820	2575	2281	1872	1809	594.1†	490.4†	423.6†
75 Re	71676	12527	11959	10535	2932	2682	2367	1949	1883	625.4†	518.7†	446.8†
76 Os	73871	12968	12385	10871	3049	2792	2457	2031	1960	658.2†	549.1†	470.7†
77 Ir	76111	13419	12824	11215	3174	2909	2551	2116	2040	691.1†	577.8†	495.8†
78 Pt	78395	13880	13273	11564	3296	3027	2645	2202	2122	725.4†	609.1†	519.4†
79 Au	80725	14353	13734	11919	3425	3148	2743	2291	2206	762.1†	642.7†	546.3†
80 Hg	83102	14839	14209	12284	3562	3279	2847	2385	2295	802.2†	680.2†	576.6†
81 Tl	85530	15347	14698	12658	3704	3416	2957	2485	2389	846.2†	720.5†	609.5†
82 Pb	88005	15861	15200	13035	3851	3554	3066	2586	2484	891.8†	761.9†	643.5†
83 Bi	90524	16388	15711	13419	3999	3696	3177	2688	2580	939†	805.2†	678.8†
84 Po	93105	16939	16244	13814	4149	3854	3302	2798	2683	995*	851*	705*
85 At	95730	17493	16785	14214	4317	4008	3426	2909	2787	1042*	886*	740*
86 Rn	98404	18049	17337	14619	4482	4159	3538	3022	2892	1097*	929*	768*
87 Fr	101137	18639	17907	15031	4652	4327	3663	3136	3000	1153*	980*	810*
88 Ra	103922	19237	18484	15444	4822	4490	3792	3248	3105	1208*	1058	879*
89 Ac	106755	19840	19083	15871	5002	4656	3909	3370	3219	1269*	1080*	890*
90 Th	109651	20472	19693	16300	5182	4830	4046	3491	3332	1330*	1168*	966.4†
91 Pa	112601	21105	20314	16733	5367	5001	4174	3611	3442	1387*	1224*	1007*
92 U	115606	21757	20948	17166	5548	5182	4303	3728	3552	1439* ^b	1271* ^b	1043†

Table 1-1. Electron binding energies (continued).

Element	N ₄ 4d _{3/2}	N ₅ 4d _{5/2}	N ₆ 4f _{5/2}	N ₇ 4f _{7/2}	O ₁ 5s	O ₂ 5p _{1/2}	O ₃ 5p _{3/2}	O ₄ 5d _{3/2}	O ₅ 5d _{5/2}	P ₁ 6s	P ₂ 6p _{1/2}	P ₃ 6p _{3/2}
71 Lu	206.1*	196.3*	8.9*	7.5*	57.3*	33.6*	26.7*					
72 Hf	220.0†	211.5†	15.9†	14.2†	64.2†	38*	29.9†					
73 Ta	237.9†	226.4†	23.5†	21.6†	69.7†	42.2*	32.7†					
74 W	255.9†	243.5†	33.6*	31.4†	75.6†	45.3*b	36.8†					
75 Re	273.9†	260.5†	42.9*	40.5*	83†	45.6*	34.6*b					
76 Os	293.1†	278.5†	53.4†	50.7†	84*	58*	44.5†					
77 Ir	311.9†	296.3†	63.8†	60.8†	95.2*b	63.0*b	48.0†					
78 Pt	331.6†	314.6†	74.5†	71.2†	101.7*b	65.3*b	51.7†					
79 Au	353.2†	335.1†	87.6†	84.0	107.2*b	74.2†	57.2†					
80 Hg	378.2†	358.8†	104.0†	99.9†	127†	83.1†	64.5†	9.6†	7.8†			
81 Tl	405.7†	385.0†	122.2†	117.8†	136.0*b	94.6†	73.5†	14.7†	12.5†			
82 Pb	434.3†	412.2†	141.7†	136.9†	147*b	106.4†	83.3†	20.7†	18.1†			
83 Bi	464.0†	440.1†	162.3†	157.0†	159.3*b	119.0†	92.6†	26.9†	23.8†			
84 Po	500*	473*	184*	184*	177*	132*	104*	31*	31*			
85 At	533*	507	210*	210*	195*	148*	115*	40*	40*			
86 Rn	567*	541*	238*	238*	214*	164*	127*	48*	48*	26		
87 Fr	603*	577*	268*	268*	234*	182*	140*	58*	58*	34	15	15
88 Ra	636*	603*	299*	299*	254*	200*	153*	68*	68*	44	19	19
89 Ac	675*	639*	319*	319*	272*	215*	167*	80*	80*	—	—	—
90 Th	712.1†	675.2†	342.4†	333.1†	290*a	229*a	182*a	92.5†	85.4†	41.4†	24.5†	16.6†
91 Pa	743*	708*	371*	360*	310*	232*	232*	94*	94*	—	—	—
92 U	778.3†	736.2†	388.2*	377.4†	321*ab	257*ab	192*ab	102.8†	94.2†	43.9†	26.8†	16.8†

1.2 X-RAY EMISSION ENERGIES

Jeffrey B. Kortright and Albert C. Thompson

In Table 1-2, characteristic K , L , and M x-ray line energies are given for elements with $3 \leq Z \leq 95$. Only the strongest lines are included: $K\alpha_1$, $K\alpha_2$, $K\beta_1$, $L\alpha_1$, $L\alpha_2$, $L\beta_1$, $L\beta_2$, $L\gamma_1$, and $M\alpha_1$. Wavelengths, in angstroms, can be obtained from the relation $\lambda = 12,398/E$, where E is in eV. The data in the table were based on Ref. 1, which should be consulted for a more complete listing. Widths of the $K\alpha$ lines can be found in Ref. 2.

Table 1-3 provides a listing of these, and additional, lines (arranged by increasing energy), together with relative intensities. An intensity of 100 is assigned to the strongest line in each shell for each element. Figure 1-1 illustrates the transitions that give rise to the lines in Table 1-3.

REFERENCES

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2. M. O. Krause and J. H. Oliver, "Natural Widths of Atomic K and L Levels, $K\alpha$ X-Ray Lines and Several KLL Auger Lines," *J. Phys. Chem. Ref. Data* **8**, 329 (1979).

Table 1-2. Photon energies, in electron volts, of principal K-, L-, and M-shell emission lines.

Element	K α_1	K α_2	K β_1	L α_1	L α_2	L β_1	L β_2	L γ	M α_1
3 Li	54.3								
4 Be	108.5								
5 B	183.3								
6 C	277								
7 N	392.4								
8 O	524.9								
9 F	676.8								
10 Ne	848.6	848.6							
11 Na	1,040.98	1,040.98	1,071.1						
12 Mg	1,253.60	1,253.60	1,302.2						
13 Al	1,486.70	1,486.27	1,557.45						
14 Si	1,739.98	1,739.38	1,835.94						
15 P	2,013.7	2,012.7	2,139.1						
16 S	2,307.84	2,306.64	2,464.04						
17 Cl	2,622.39	2,620.78	2,815.6						
18 Ar	2,957.70	2,955.63	3,190.5						
19 K	3,313.8	3,311.1	3,589.6						
20 Ca	3,691.68	3,688.09	4,012.7	341.3	341.3	344.9			
21 Sc	4,090.6	4,086.1	4,460.5	395.4	395.4	399.6			

Table 1-2. Energies of x-ray emission lines (continued).

Element	$K\alpha_1$	$K\alpha_2$	$K\beta_1$	$L\alpha_1$	$L\alpha_2$	$L\beta_1$	$L\beta_2$	$L\gamma$	$M\alpha_1$
22 Ti	4,510.84	4,504.86	4,931.81	452.2	452.2	458.4			
23 V	4,952.20	4,944.64	5,427.29	511.3	511.3	519.2			
24 Cr	5,414.72	5,405.509	5,946.71	572.8	572.8	582.8			
25 Mn	5,898.75	5,887.65	6,490.45	637.4	637.4	648.8			
26 Fe	6,403.84	6,390.84	7,057.98	705.0	705.0	718.5			
27 Co	6,930.32	6,915.30	7,649.43	776.2	776.2	791.4			
28 Ni	7,478.15	7,460.89	8,264.66	851.5	851.5	868.8			
29 Cu	8,047.78	8,027.83	8,905.29	929.7	929.7	949.8			
30 Zn	8,638.86	8,615.78	9,572.0	1,011.7	1,011.7	1,034.7			
31 Ga	9,251.74	9,224.82	10,264.2	1,097.92	1,097.92	1,124.8			
32 Ge	9,886.42	9,855.32	10,982.1	1,188.00	1,188.00	1,218.5			
33 As	10,543.72	10,507.99	11,726.2	1,282.0	1,282.0	1,317.0			
34 Se	11,222.4	11,181.4	12,495.9	1,379.10	1,379.10	1,419.23			
35 Br	11,924.2	11,877.6	13,291.4	1,480.43	1,480.43	1,525.90			
36 Kr	12,649	12,598	14,112	1,586.0	1,586.0	1,636.6			
37 Rb	13,395.3	13,335.8	14,961.3	1,694.13	1,692.56	1,752.17			
38 Sr	14,165	14,097.9	15,835.7	1,806.56	1,804.74	1,871.72			
39 Y	14,958.4	14,882.9	16,737.8	1,922.56	1,920.47	1,995.84			
40 Zr	15,775.1	15,690.9	17,667.8	2,042.36	2,039.9	2,124.4*	2,219.4	2,302.7	

Table 1-2. Energies of x-ray emission lines (continued).

Element	K α_1	K α_2	K β_1	L α_1	L α_2	L β_1	L β_2	L γ	M α_1
63 Eu	41,542.2	40,901.9	47,037.9	5,845.7	5,816.6	6,456.4	6,843.2	7,480.3	1,131
64 Gd	42,996.2	42,308.9	48,697	6,057.2	6,025.0	6,713.2	7,102.8	7,785.8	1,185
65 Tb	44,481.6	43,744.1	50,382	6,272.8	6,238.0	6,978	7,366.7	8,102	1,240
66 Dy	45,998.4	45,207.8	52,119	6,495.2	6,457.7	7,247.7	7,635.7	8,418.8	1,293
67 Ho	47,546.7	46,699.7	53,877	6,719.8	6,679.5	7,525.3	7,911	8,747	1,348
68 Er	49,127.7	48,221.1	55,681	6,948.7	6,905.0	7,810.9	8,189.0	9,089	1,406
69 Tm	50,741.6	49,772.6	57,517	7,179.9	7,133.1	8,101	8,468	9,426	1,462
70 Yb	52,388.9	51,354.0	59,370	7,415.6	7,367.3	8,401.8	8,758.8	9,780.1	1,521.4
71 Lu	54,069.8	52,965.0	61,283	7,655.5	7,604.9	8,709.0	9,048.9	10,143.4	1,581.3
72 Hf	55,790.2	54,611.4	63,234	7,899.0	7,844.6	9,022.7	9,347.3	10,515.8	1,644.6
73 Ta	57,532	56,277	65,223	8,146.1	8,087.9	9,343.1	9,651.8	10,895.2	1,710
74 W	59,318.24	57,981.7	67,244.3	8,397.6	8,335.2	9,672.35	9,961.5	11,285.9	1,775.4
75 Re	61,140.3	59,717.9	69,310	8,652.5	8,586.2	10,010.0	10,275.2	11,685.4	1,842.5
76 Os	63,000.5	61,486.7	71,413	8,911.7	8,841.0	10,355.3	10,598.5	12,095.3	1,910.2
77 Ir	64,895.6	63,286.7	73,560.8	9,175.1	9,099.5	10,708.3	10,920.3	12,512.6	1,979.9
78 Pt	66,832	65,112	75,748	9,442.3	9,361.8	11,070.7	11,250.5	12,942.0	2,050.5
79 Au	68,803.7	66,989.5	77,984	9,713.3	9,628.0	11,442.3	11,584.7	13,381.7	2,122.9
80 Hg	70,819	68,895	80,253	9,988.8	9,897.6	11,822.6	11,924.1	13,830.1	2,195.3
81 Tl	72,871.5	70,831.9	82,576	10,268.5	10,172.8	12,213.3	12,271.5	14,291.5	2,270.6

82 Pb	74,969.4	72,804.2	84,936	10,551.5	10,449.5	12,613.7	12,622.6	14,764.4	2,345.5
83 Bi	77,107.9	74,814.8	87,343	10,838.8	10,730.91	13,023.5	12,979.9	15,247.7	2,422.6
84 Po	79,290	76,862	89,800	11,130.8	11,015.8	13,447	13,340.4	15,744	—
85 At	81,520	78,950	92,300	11,426.8	11,304.8	13,876	—	16,251	—
86 Rn	83,780	81,070	94,870	11,727.0	11,597.9	14,316	—	16,770	—
87 Fr	86,100	83,230	97,470	12,031.3	11,895.0	14,770	14,450	17,303	—
88 Ra	88,470	85,430	100,130	12,339.7	12,196.2	15,235.8	14,841.4	17,849	—
89 Ac	90,884	87,670	102,850	12,652.0	12,500.8	15,713	—	18,408	—
90 Th	93,350	89,953	105,609	12,968.7	12,809.6	16,202.2	15,623.7	18,982.5	2,996.1
91 Pa	95,868	92,287	108,427	13,290.7	13,122.2	16,702	16,024	19,568	3,082.3
92 U	98,439	94,665	111,300	13,614.7	13,438.8	17,220.0	16,428.3	20,167.1	3,170.8
93 Np	—	—	—	13,944.1	13,759.7	17,750.2	16,840.0	20,784.8	—
94 Pu	—	—	—	14,278.6	14,084.2	18,293.7	17,255.3	21,417.3	—
95 Am	—	—	—	14,617.2	14,411.9	18,852.0	17,676.5	22,065.2	—

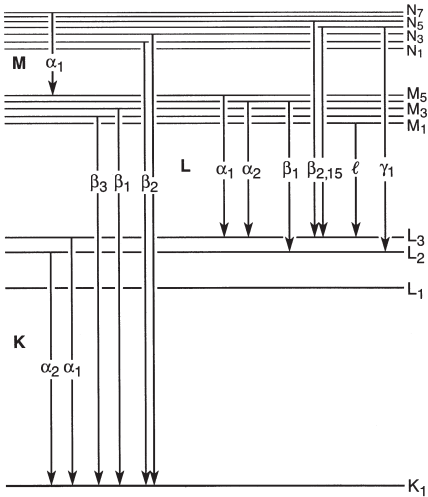


Fig. 1-1. Transitions that give rise to the emission lines in Table 1-3.

Table 1-3. Photon energies and relative intensities of K-, L-, and M-shell lines shown in Fig. 1-1, arranged by increasing energy. An intensity of 100 is assigned to the strongest line in each shell for each element.

Energy (eV)	Element	Line	Relative intensity	524.9	8 O	K $\alpha_{1,2}$	151	851.5	28 Ni	L $\alpha_{1,2}$	111
54.3	3 Li	K $\alpha_{1,2}$	150	556.3	25 Mn	L β_1	15	868.8	28 Ni	L β_1	68
108.5	4 Be	K $\alpha_{1,2}$	150	572.8	24 Cr	L $\alpha_{1,2}$	111	883	58 Ce	M α_1	100
183.3	5 B	K $\alpha_{1,2}$	151	582.8	24 Cr	L β_1	79	884	30 Zn	L β_1	7
277	6 C	K $\alpha_{1,2}$	147	615.2	26 Fe	L β_1	10	929.2	59 Pr	M α_1	100
348.3	21 Sc	L β_1	21	637.4	25 Mn	L $\alpha_{1,2}$	111	929.7	29 Cu	L $\alpha_{1,2}$	111
392.4	7 N	K $\alpha_{1,2}$	150	648.8	25 Mn	L β_1	77	949.8	29 Cu	L β_1	65
395.3	22 Ti	L β_1	46	676.8	9 F	K $\alpha_{1,2}$	148	957.2	31 Ga	L β_1	7
395.4	21 Sc	L $\alpha_{1,2}$	111	677.8	27 Co	L β_1	10	978	60 Nd	M α_1	100
399.6	21 Sc	L β_1	77	705.0	26 Fe	L $\alpha_{1,2}$	111	1,011.7	30 Zn	L $\alpha_{1,2}$	111
446.5	23 V	L β_1	28	718.5	26 Fe	L β_1	66	1,034.7	30 Zn	L β_1	65
452.2	22 Ti	L $\alpha_{1,2}$	111	742.7	28 Ni	L β_1	9	1,036.2	32 Ge	L β_1	6
458.4	22 Ti	L β_1	79	776.2	27 Co	L $\alpha_{1,2}$	111	1,041.0	11 Na	K $\alpha_{1,2}$	150
500.3	24 Cr	L β_1	17	791.4	27 Co	L β_1	76	1,081	62 Sm	M α_1	100
511.3	23 V	L $\alpha_{1,2}$	111	811.1	29 Cu	L β_1	8	1,097.9	31 Ga	L $\alpha_{1,2}$	111
519.2	23 V	L β_1	80	833	57 La	M α_1	100	1,120	33 As	L β_1	6
				848.6	10 Ne	K $\alpha_{1,2}$	150	1,124.8	31 Ga	L β_1	66

Table I-3. Energies and intensities of x-ray emission lines (continued).

Energy (eV)	Element	Line	Relative intensity	1,462	69 Tm	M α_1	100	1,740.0	14 Si	K α_1	100
1,131	63 Eu	M α_1	100	1,480.4	35 Br	L $\alpha_{1,2}$	111	1,752.2	37 Rb	L β_1	58
1,185	64 Gd	M α_1	100	1,482.4	37 Rb	L1	5	1,775.4	74 W	M α_1	100
1,188.0	32 Ge	L $\alpha_{1,2}$	111	1,486.3	13 Al	K α_2	50	1,792.0	40 Zr	L1	5
1,204.4	34 Se	L1	6	1,486.7	13 Al	K α_1	100	1,804.7	38 Sr	L α_2	11
1,218.5	32 Ge	L β_1	60	1,521.4	70 Yb	M α_1	100	1,806.6	38 Sr	L α_1	100
1,240	65 Tb	M α_1	100	1,525.9	35 Br	L β_1	59	1,835.9	14 Si	K β_1	2
1,253.6	12 Mg	K $\alpha_{1,2}$	150	1,557.4	13 Al	K β_1	1	1,842.5	75 Re	M α_1	100
1,282.0	33 As	L $\alpha_{1,2}$	111	1,581.3	71 Lu	M α_1	100	1,871.7	38 Sr	L β_1	58
1,293	66 Dy	M α_1	100	1,582.2	38 Sr	L1	5	1,902.2	41 Nb	L1	5
1,293.5	35 Br	L1	5	1,586.0	36 Kr	L $\alpha_{1,2}$	111	1,910.2	76 Os	M α_1	100
1,317.0	33 As	L β_1	60	1,636.6	36 Kr	L β_1	57	1,920.5	39 Y	L α_2	11
1,348	67 Ho	M α_1	100	1,644.6	72 Hf	M α_1	100	1,922.6	39 Y	L α_1	100
1,379.1	34 Se	L $\alpha_{1,2}$	111	1,685.4	39 Y	L1	5	1,979.9	77 Ir	M α_1	100
1,386	36 Kr	L1	5	1,692.6	37 Rb	L α_2	11	1,995.8	39 Y	L β_1	57
1,406	68 Er	M α_1	100	1,694.1	37 Rb	L α_1	100	2,012.7	15 P	K α_2	50
1,419.2	34 Se	L β_1	59	1,709.6	73 Ta	M α_1	100	2,013.7	15 P	K α_1	100
				1,739.4	14 Si	K α_2	50	2,015.7	42 Mo	L1	5

2,039.9	40 Zr	L α_2	11	2,367.0	41 Nb	L $\beta_{2,15}$	3	2,696.7	45 Rh	L α_1	100
2,042.4	40 Zr	L α_1	100	2,376.5	45 Rh	L γ_1	4	2,767.4	48 Cd	L γ_1	4
2,050.5	78 Pt	M α_1	100	2,394.8	42 Mo	L β_1	53	2,792	43 Tc	L γ_1	3
2,122	43 Tc	L γ_1	5	2,420	43 Tc	L α_2	11	2,815.6	17 Cl	K β_1	6
2,122.9	79 Au	M α_1	100	2,422.6	83 Bi	M α_1	100	2,833.3	46 Pd	L α_2	11
2,124.4	40 Zr	L β_1	54	2,424	43 Tc	L α_1	100	2,834.4	45 Rh	L β_1	52
2,139.1	15 P	K β_1	3	2,461.8	41 Nb	L γ_1	2	2,836.0	44 Ru	L $\beta_{2,15}$	10
2,163.0	41 Nb	L α_2	11	2,464.0	16 S	K β_1	5	2,838.6	46 Pd	L α_1	100
2,165.9	41 Nb	L α_1	100	2,503.4	46 Pd	L γ_1	4	2,904.4	49 In	L γ_1	4
2,195.3	80 Hg	M α_1	100	2,518.3	42 Mo	L $\beta_{2,15}$	5	2,955.6	18 Ar	K α_2	50
2,219.4	40 Zr	L $\beta_{2,15}$	1	2,538	43 Tc	L β_1	54	2,957.7	18 Ar	K α_1	100
2,252.8	44 Ru	L γ_1	4	2,554.3	44 Ru	L α_2	11	2,964.5	44 Ru	L γ_1	4
2,257.4	41 Nb	L β_1	52	2,558.6	44 Ru	L α_1	100	2,978.2	47 Ag	L α_2	11
2,270.6	81 Tl	M α_1	100	2,620.8	17 Cl	K α_2	50	2,984.3	47 Ag	L α_1	100
2,289.8	42 Mo	L α_2	11	2,622.4	17 Cl	K α_1	100	2,990.2	46 Pd	L β_1	53
2,293.2	42 Mo	L α_1	100	2,623.5	42 Mo	L γ_1	3	2,996.1	90 Th	M α_1	100
2,302.7	40 Zr	L γ_1	2	2,633.7	47 Ag	L γ_1	4	3,001.3	45 Rh	L $\beta_{2,15}$	10
2,306.6	16 S	K α_2	50	2,674	43 Tc	L $\beta_{2,15}$	7	3,045.0	50 Sn	L γ_1	4
2,307.8	16 S	K α_1	100	2,683.2	44 Ru	L β_1	54	3,126.9	48 Cd	L α_2	11
2,345.5	82 Pb	M α_1	100	2,692.0	45 Rh	L α_2	11	3,133.7	48 Cd	L α_1	100

Table I-3. Energies and intensities of x-ray emission lines (continued).

Energy (eV)	Element	Line	Relative intensity	3,487.2	49 In	$L\beta_1$	58	3,937.6	53 I	$L\alpha_1$	100
3,143.8	45 Rh	$L\gamma_1$	5	3,519.6	47 Ag	$L\gamma_1$	6	3,954.1	56 Ba	L1	4
3,150.9	47 Ag	$L\beta_1$	56	3,528.1	48 Cd	$L\beta_{2,15}$	15	4,012.7	20 Ca	$K\beta_{1,3}$	13
3,170.8	92 U	$M\alpha_1$	100	3,589.6	19 K	$K\beta_{1,3}$	11	4,029.6	52 Te	$L\beta_1$	61
3,171.8	46 Pd	$L\beta_{2,15}$	12	3,595.3	51 Sb	$L\alpha_2$	11	4,086.1	21 Sc	$K\alpha_2$	50
3,188.6	51 Sb	L1	4	3,604.7	51 Sb	$L\alpha_1$	100	4,090.6	21 Sc	$K\alpha_1$	100
3,190.5	18 Ar	$K\beta_{1,3}$	10	3,636	54 Xe	L1	4	4,093	54 Xe	$L\alpha_2$	11
3,279.3	49 In	$L\alpha_2$	11	3,662.8	50 Sn	$L\beta_1$	60	4,100.8	51 Sb	$L\beta_{2,15}$	17
3,286.9	49 In	$L\alpha_1$	100	3,688.1	20 Ca	$K\alpha_2$	50	4,109.9	54 Xe	$L\alpha_1$	100
3,311.1	19 K	$K\alpha_2$	50	3,691.7	20 Ca	$K\alpha_1$	100	4,124	57 La	L1	4
3,313.8	19 K	$K\alpha_1$	100	3,713.8	49 In	$L\beta_{2,15}$	15	4,131.1	50 Sn	$L\gamma_1$	7
3,316.6	48 Cd	$L\beta_1$	58	3,716.9	48 Cd	$L\gamma_1$	6	4,220.7	53 I	$L\beta_1$	61
3,328.7	46 Pd	$L\gamma_1$	6	3,758.8	52 Te	$L\alpha_2$	11	4,272.2	55 Cs	$L\alpha_2$	11
3,335.6	52 Te	L1	4	3,769.3	52 Te	$L\alpha_1$	100	4,286.5	55 Cs	$L\alpha_1$	100
3,347.8	47 Ag	$L\beta_{2,15}$	13	3,795.0	55 Cs	L1	4	4,287.5	58 Ce	L1	4
3,435.4	50 Sn	$L\alpha_2$	11	3,843.6	51 Sb	$L\beta_1$	61	4,301.7	52 Te	$L\beta_{2,15}$	18
3,444.0	50 Sn	$L\alpha_1$	100	3,904.9	50 Sn	$L\beta_{2,15}$	16	4,347.8	51 Sb	$L\gamma_1$	8
3,485.0	53 I	L1	4	3,920.8	49 In	$L\gamma_1$	6	4,414	54 Xe	$L\beta_1$	60
				3,926.0	53 I	$L\alpha_2$	11	4,450.9	56 Ba	$L\alpha_2$	11

4,453.2	59 Pr	L1	4	4,952.2	23 V	K α_1	100	5,531.1	56 Ba	L γ_1	9
4,460.5	21 Sc	K $\beta_{1,3}$	15	4,994.5	62 Sm	L1	4	5,546.7	65 Tb	L1	4
4,466.3	56 Ba	L α_1	100	5,013.5	59 Pr	L α_2	11	5,609.0	62 Sm	L α_2	11
4,504.9	22 Ti	K α_2	50	5,033.7	59 Pr	L α_1	100	5,613.4	58 Ce	L $\beta_{2,15}$	21
4,507.5	53 I	L $\beta_{2,15}$	19	5,034	54 Xe	L γ_1	8	5,636.1	62 Sm	L α_1	100
4,510.8	22 Ti	K α_1	100	5,042.1	57 La	L β_1	60	5,721.6	60 Nd	L β_1	60
4,570.9	52 Te	L γ_1	8	5,156.5	56 Ba	L $\beta_{2,15}$	20	5,743.1	66 Dy	L1	4
4,619.8	55 Cs	L β_1	61	5,177.2	63 Eu	L1	4	5,788.5	57 La	L γ_1	9
4,633.0	60 Nd	L1	4	5,207.7	60 Nd	L α_2	11	5,816.6	63 Eu	L α_2	11
4,634.2	57 La	L α_2	11	5,230.4	60 Nd	L α_1	100	5,845.7	63 Eu	L α_1	100
4,651.0	57 La	L α_1	100	5,262.2	58 Ce	L β_1	61	5,850	59 Pr	L $\beta_{2,15}$	21
4,714	54 Xe	L $\beta_{2,15}$	20	5,280.4	55 Cs	L γ_1	8	5,887.6	25 Mn	K α_2	50
4,800.9	53 I	L γ_1	8	5,362.1	64 Gd	L1	4	5,898.8	25 Mn	K α_1	100
4,809	61 Pm	L1	4	5,383.5	57 La	L $\beta_{2,15}$	21	5,943.4	67 Ho	L1	4
4,823.0	58 Ce	L α_2	11	5,405.5	24 Cr	K α_2	50	5,946.7	24 Cr	K $\beta_{1,3}$	15
4,827.5	56 Ba	L β_1	60	5,408	61 Pm	L α_2	11	5,961	61 Pm	L β_1	61
4,840.2	58 Ce	L α_1	100	5,414.7	24 Cr	K α_1	100	6,025.0	64 Gd	L α_2	11
4,931.8	22 Ti	K $\beta_{1,3}$	15	5,427.3	23 V	K $\beta_{1,3}$	15	6,052	58 Ce	L γ_1	9
4,935.9	55 Cs	L $\beta_{2,15}$	20	5,432	61 Pm	L α_1	100	6,057.2	64 Gd	L α_1	100
4,944.6	23 V	K α_2	50	5,488.9	59 Pr	L β_1	61	6,089.4	60 Nd	L $\beta_{2,15}$	21

Table 1-3. Energies and intensities of x-ray emission lines (continued).

Energy (eV)	Element	Line	Relative intensity	64 Gd	67 Ho	70 Yb	73 Ta	76 Os	79 Au	82 Pb
6,152	68 Er	L1	4	6,713.2	67 Ho	70 Yb	73 Ta	76 Os	79 Au	82 Pb
6,205.1	62 Sm	Lβ ₁	61	6,719.8	67 Ho	70 Yb	73 Ta	76 Os	79 Au	82 Pb
6,238.0	65 Tb	Lα ₂	11	6,752.8	71 Lu	L1	4	7,415.6	70 Yb	Lα ₁
6,272.8	65 Tb	Lα ₁	100	6,843.2	63 Eu	Lβ _{2,15}	21	7,460.9	28 Ni	Kα ₂
6,322.1	59 Pr	Lγ ₁	9	6,892	61 Pm	Lγ ₁	10	7,478.2	28 Ni	Kα ₁
6,339	61 Pm	Lβ ₂	21	6,905.0	68 Er	Lα ₂	11	7,480.3	63 Eu	Lγ ₁
6,341.9	69 Tm	L1	4	6,915.3	27 Co	Kα ₂	51	7,525.3	67 Ho	Lβ ₁
6,390.8	26 Fe	Kα ₂	50	6,930.3	27 Co	Kα ₁	100	7,603.6	75 Re	L1
6,403.8	26 Fe	Kα ₁	100	6,948.7	68 Er	Lα ₁	100	7,604.9	71 Lu	Lα ₂
6,456.4	63 Eu	Lβ ₁	62	6,959.6	72 Hf	L1	5	7,635.7	66 Dy	Lβ ₂
6,457.7	66 Dy	Lα ₂	11	6,978	65 Tb	Lβ ₁	61	7,649.4	27 Co	Kβ _{1,3}
6,490.4	25 Mn	Kβ _{1,3}	17	7,058.0	26 Fe	Kβ _{1,3}	17	7,655.5	71 Lu	Lα ₁
6,495.2	66 Dy	Lα ₁	100	7,102.8	64 Gd	Lβ _{2,15}	21	7,785.8	64 Gd	Lγ ₁
6,545.5	70 Yb	L1	4	7,133.1	69 Tm	Lα ₂	11	7,810.9	68 Er	Lβ ₁
6,587.0	62 Sm	Lβ _{2,15}	21	7,173.1	73 Ta	L1	5	7,822.2	76 Os	L1
6,602.1	60 Nd	Lγ ₁	10	7,178.0	62 Sm	Lγ ₁	10	7,844.6	72 Hf	Lα ₂
6,679.5	67 Ho	Lα ₂	11	7,179.9	69 Tm	Lα ₁	100	7,899.0	72 Hf	Lα ₁
				7,247.7	66 Dy	Lβ ₁	62	7,911	67 Ho	Lβ _{2,15}
				7,366.7	65 Tb	Lβ _{2,15}	21	8,027.8	29 Cu	Kα ₂

8,045.8	77 Ir	L ₁	5	8,721.0	80 Hg	L ₁	5	9,442.3	78 Pt	L α_1	100
8,047.8	29 Cu	K α_1	100	8,747	67 Ho	L γ_1	11	9,572.0	30 Zn	K $\beta_{1,3}$	17
8,087.9	73 Ta	L α_2	11	8,758.8	70 Yb	L $\beta_{2,15}$	20	9,628.0	79 Au	L α_2	11
8,101	69 Tm	L β_1	64	8,841.0	76 Os	L α_2	11	9,651.8	73 Ta	L β_2	20
8,102	65 Tb	L γ_1	11	8,905.3	29 Cu	K $\beta_{1,3}$	17	9,672.4	74 W	L β_1	67
8,146.1	73 Ta	L α_1	100	8,911.7	76 Os	L α_1	100	9,713.3	79 Au	L α_1	100
8,189.0	68 Er	L $\beta_{2,15}$	20	8,953.2	81 Tl	L ₁	6	9,780.1	70 Yb	L γ_1	12
8,264.7	28 Ni	K $\beta_{1,3}$	17	9,022.7	72 Hf	L β_1	67	9,855.3	32 Ge	K α_2	51
8,268	78 Pt	L ₁	5	9,048.9	71 Lu	L β_2	19	9,886.4	32 Ge	K α_1	100
8,335.2	74 W	L α_2	11	9,089	68 Er	L γ_1	11	9,897.6	80 Hg	L α_2	11
8,397.6	74 W	L α_1	100	9,099.5	77 Ir	L α_2	11	9,961.5	74 W	L β_2	21
8,401.8	70 Yb	L β_1	65	9,175.1	77 Ir	L α_1	100	9,988.8	80 Hg	L α_1	100
8,418.8	66 Dy	L γ_1	11	9,184.5	82 Pb	L ₁	6	10,010.0	75 Re	L β_1	66
8,468	69 Tm	L $\beta_{2,15}$	20	9,224.8	31 Ga	K α_2	51	10,143.4	71 Lu	L γ_1	12
8,493.9	79 Au	L ₁	5	9,251.7	31 Ga	K α_1	100	10,172.8	81 Tl	L α_2	11
8,586.2	75 Re	L α_2	11	9,343.1	73 Ta	L β_1	67	10,260.3	31 Ga	K β_3	5
8,615.8	30 Zn	K α_2	51	9,347.3	72 Hf	L β_2	20	10,264.2	31 Ga	K β_1	66
8,638.9	30 Zn	K α_1	100	9,361.8	78 Pt	L α_2	11	10,268.5	81 Tl	L α_1	100
8,652.5	75 Re	L α_1	100	9,420.4	83 Bi	L ₁	6	10,275.2	75 Re	L β_2	22
8,709.0	71 Lu	L β_1	66	9,426	69 Tm	L γ_1	12	10,355.3	76 Os	L β_1	67

Table I-3. Energies and intensities of x-ray emission lines (continued).

Energy (eV)	Element	Line	Relative intensity	11,250.5	78 Pt	L β_2	23	12,598	36 Kr	K α_2	52
10,449.5	82 Pb	L α_2	11	11,285.9	74 W	L γ_1	13	12,613.7	82 Pb	L β_1	66
10,508.0	33 As	K α_2	51	11,442.3	79 Au	L β_1	67	12,622.6	82 Pb	L β_2	25
10,515.8	72 Hf	L γ_1	12	11,584.7	79 Au	L β_2	23	12,649	36 Kr	K α_1	100
10,543.7	33 As	K α_1	100	11,618.3	92 U	L γ_1	7	12,652	34 Se	K β_2	1
10,551.5	82 Pb	L α_1	100	11,685.4	75 Re	L γ_1	13	12,809.6	90 Th	L α_2	11
10,598.5	76 Os	L β_2	22	11,720.3	33 As	K β_3	6	12,942.0	78 Pt	L γ_1	13
10,708.3	77 Ir	L β_1	66	11,726.2	33 As	K β_1	13	12,968.7	90 Th	L α_1	100
10,730.9	83 Bi	L α_2	11	11,822.6	80 Hg	L β_1	67	12,979.9	83 Bi	L β_2	25
10,838.8	83 Bi	L α_1	100	11,864	33 As	K β_2	1	13,023.5	83 Bi	L β_1	67
10,895.2	73 Ta	L γ_1	12	11,877.6	35 Br	K α_2	52	13,284.5	35 Br	K β_3	7
10,920.3	77 Ir	L β_2	22	11,924.1	80 Hg	L β_2	24	13,291.4	35 Br	K β_1	14
10,978.0	32 Ge	K β_3	6	11,924.2	35 Br	K α_1	100	13,335.8	37 Rb	K α_2	52
10,982.1	32 Ge	K β_1	60	12,095.3	76 Os	L γ_1	13	13,381.7	79 Au	L γ_1	13
11,070.7	78 Pt	L β_1	67	12,213.3	81 Tl	L β_1	67	13,395.3	37 Rb	K α_1	100
11,118.6	90 Th	L γ_1	6	12,271.5	81 Tl	L β_2	25	13,438.8	92 U	L α_2	11
11,181.4	34 Se	K α_2	52	12,489.6	34 Se	K β_3	6	13,469.5	35 Br	K β_2	1
11,222.4	34 Se	K α_1	100	12,495.9	34 Se	K β_1	13	13,614.7	92 U	L α_1	100
				12,512.6	77 Ir	L γ_1	13	13,830.1	80 Hg	L γ_1	14

14,097.9	38 Sr	K α_2	52	16,202.2	90 Th	L β_1	69	19,150.4	44 Ru	K α_2	53
14,104	36 Kr	K β_3	7	16,428.3	92 U	L β_2	26	19,279.2	44 Ru	K α_1	100
14,112	36 Kr	K β_1	14	16,521.0	41 Nb	K α_2	52	19,590.3	42 Mo	K β_3	8
14,165.0	38 Sr	K α_1	100	16,615.1	41 Nb	K α_1	100	19,608.3	42 Mo	K β_1	15
14,291.5	81 Tl	L γ_1	14	16,725.8	39 Y	K β_3	8	19,965.2	42 Mo	K β_2	3
14,315	36 Kr	K β_2	2	16,737.8	39 Y	K β_1	15	20,073.7	45 Rh	K α_2	53
14,764.4	82 Pb	L γ_1	14	17,015.4	39 Y	K β_2	3	20,167.1	92 U	L γ_1	15
14,882.9	39 Y	K α_2	52	17,220.0	92 U	L β_1	61	20,216.1	45 Rh	K α_1	100
14,951.7	37 Rb	K β_3	7	17,374.3	42 Mo	K α_2	52	20,599	43 Tc	K β_3	8
14,958.4	39 Y	K α_1	100	17,479.3	42 Mo	K α_1	100	20,619	43 Tc	K β_1	16
14,961.3	37 Rb	K β_1	14	17,654	40 Zr	K β_3	8	21,005	43 Tc	K β_2	4
15,185	37 Rb	K β_2	2	17,667.8	40 Zr	K β_1	15	21,020.1	46 Pd	K α_2	53
15,247.7	83 Bi	L γ_1	14	17,970	40 Zr	K β_2	3	21,177.1	46 Pd	K α_1	100
15,623.7	90 Th	L β_2	26	18,250.8	43 Tc	K α_2	53	21,634.6	44 Ru	K β_3	8
15,690.9	40 Zr	K α_2	52	18,367.1	43 Tc	K α_1	100	21,656.8	44 Ru	K β_1	16
15,775.1	40 Zr	K α_1	100	18,606.3	41 Nb	K β_3	8	21,990.3	47 Ag	K α_2	53
15,824.9	38 Sr	K β_3	7	18,622.5	41 Nb	K β_1	15	22,074	44 Ru	K β_2	4
15,835.7	38 Sr	K β_1	14	18,953	41 Nb	K β_2	3	22,162.9	47 Ag	K α_1	100
16,084.6	38 Sr	K β_2	3	18,982.5	90 Th	L γ_1	16	22,698.9	45 Rh	K β_3	8

Table 1-3. Energies and intensities of x-ray emission lines (continued).

Energy (eV)	Element	Line	Relative intensity	51 Sb	51 Sb	48 Cd	48 Cd	52 Te	52 Te	56 Ba	56 Ba	57 La	57 La	54 Xe	54 Xe	58 Ce	58 Ce	55 Cs	55 Cs	59 Pr	59 Pr	55 Cs	55 Cs
22,723.6	45 Rh	K β_1	16	26,359.1	K α_1	100	30,972.8	55 Cs	K α_1	100													
22,984.1	48 Cd	K α_2	53	26,643.8	K β_2	4	30,995.7	52 Te	K β_1	18													
23,172.8	45 Rh	K β_2	4	27,201.7	K α_2	54	31,700.4	52 Te	K β_2	5													
23,173.6	48 Cd	K α_1	100	27,237.7	K β_3	9	31,817.1	56 Ba	K α_2	54													
23,791.1	46 Pd	K β_3	8	27,275.9	K β_1	17	32,193.6	56 Ba	K α_1	100													
23,818.7	46 Pd	K β_1	16	27,472.3	K α_1	100	32,239.4	53 I	K β_3	9													
24,002.0	49 In	K α_2	53	27,860.8	K β_2	5	32,294.7	53 I	K β_1	18													
24,209.7	49 In	K α_1	100	28,317.2	K α_2	54	33,034.1	57 La	K α_2	54													
24,299.1	46 Pd	K β_2	4	28,444.0	K β_3	9	33,042	53 I	K β_2	5													
24,911.5	47 Ag	K β_3	9	28,486.0	K β_1	17	33,441.8	57 La	K α_1	100													
24,942.4	47 Ag	K β_1	16	28,612.0	K α_1	100	33,562	54 Xe	K β_3	9													
25,044.0	50 Sn	K α_2	53	29,109.3	K β_2	5	33,624	54 Xe	K β_1	18													
25,271.3	50 Sn	K α_1	100	29,458	K α_2	54	34,278.9	58 Ce	K α_2	55													
25,456.4	47 Ag	K β_2	4	29,679.2	K β_3	9	34,415	54 Xe	K β_2	5													
26,061.2	48 Cd	K β_3	9	29,725.6	K β_1	18	34,719.7	58 Ce	K α_1	100													
26,095.5	48 Cd	K β_1	17	29,779	K α_1	100	34,919.4	55 Cs	K β_3	9													
26,110.8	51 Sb	K α_2	54	30,389.5	K β_2	5	34,986.9	55 Cs	K β_1	18													
				30,625.1	K α_2	54	35,550.2	59 Pr	K α_2	55													
				30,944.3	K β_3	9	35,822	55 Cs	K β_2	6													

36,026.3	59 Pr	K α_1	100	41,542.2	63 Eu	K α_1	100	47,037.9	63 Eu	K β_1	19
36,304.0	56 Ba	K β_3	10	41,773	59 Pr	K β_2	6	47,546.7	67 Ho	K α_1	100
36,378.2	56 Ba	K β_1	18	42,166.5	60 Nd	K β_3	10	48,221.1	68 Er	K α_2	56
36,847.4	60 Nd	K α_2	55	42,271.3	60 Nd	K β_1	19	48,256	63 Eu	K β_2	6
37,257	56 Ba	K β_2	6	42,308.9	64 Gd	K α_2	56	48,555	64 Gd	K β_3	10
37,361.0	60 Nd	K α_1	100	42,996.2	64 Gd	K α_1	100	48,697	64 Gd	K β_1	20
37,720.2	57 La	K β_3	10	43,335	60 Nd	K β_2	6	49,127.7	68 Er	K α_1	100
37,801.0	57 La	K β_1	19	43,713	61 Pm	K β_3	10	49,772.6	69 Tm	K α_2	57
38,171.2	61 Pm	K α_2	55	43,744.1	65 Tb	K α_2	56	49,959	64 Gd	K β_2	7
38,724.7	61 Pm	K α_1	100	43,826	61 Pm	K β_1	19	50,229	65 Tb	K β_3	10
38,729.9	57 La	K β_2	6	44,481.6	65 Tb	K α_1	100	50,382	65 Tb	K β_1	20
39,170.1	58 Ce	K β_3	10	44,942	61 Pm	K β_2	6	50,741.6	69 Tm	K α_1	100
39,257.3	58 Ce	K β_1	19	45,207.8	66 Dy	K α_2	56	51,354.0	70 Yb	K α_2	57
39,522.4	62 Sm	K α_2	55	45,289	62 Sm	K β_3	10	51,698	65 Tb	K β_2	7
40,118.1	62 Sm	K α_1	100	45,413	62 Sm	K β_1	19	51,957	66 Dy	K β_3	10
40,233	58 Ce	K β_2	6	45,998.4	66 Dy	K α_1	100	52,119	66 Dy	K β_1	20
40,652.9	59 Pr	K β_3	10	46,578	62 Sm	K β_2	6	52,388.9	70 Yb	K α_1	100
40,748.2	59 Pr	K β_1	19	46,699.7	67 Ho	K α_2	56	52,965.0	71 Lu	K α_2	57
40,901.9	63 Eu	K α_2	56	46,903.6	63 Eu	K β_3	10	53,476	66 Dy	K β_2	7

Table 1-3. Energies and intensities of x-ray emission lines (continued).

Energy (eV)	Element	Line	Relative intensity	59,370	70 Yb	K β_1	21	66,989.5	79 Au	K α_2	59
53,711	67 Ho	K β_3	11	59,717.9	75 Re	K α_2	58	66,990	73 Ta	K β_2	7
53,877	67 Ho	K β_1	20	60,980	70 Yb	K β_2	7	67,244.3	74 W	K β_1	22
54,069.8	71 Lu	K α_1	100	61,050	71 Lu	K β_3	11	68,803.7	79 Au	K α_1	100
54,611.4	72 Hf	K α_2	57	61,140.3	75 Re	K α_1	100	68,895	80 Hg	K α_2	59
55,293	67 Ho	K β_2	7	61,283	71 Lu	K β_1	21	68,994	75 Re	K β_3	12
55,494	68 Er	K β_3	11	61,486.7	76 Os	K α_2	58	69,067	74 W	K β_2	8
55,681	68 Er	K β_1	21	62,970	71 Lu	K β_2	7	69,310	75 Re	K β_1	22
55,790.2	72 Hf	K α_1	100	62,980	72 Hf	K β_3	11	70,819	80 Hg	K α_1	100
56,277	73 Ta	K α_2	57	63,000.5	76 Os	K α_1	100	70,831.9	81 Tl	K α_2	60
57,210	68 Er	K β_2	7	63,234	72 Hf	K β_1	22	71,077	76 Os	K β_3	12
57,304	69 Tm	K β_3	11	63,286.7	77 Ir	K α_2	58	71,232	75 Re	K β_2	8
57,517	69 Tm	K β_1	21	64,895.6	77 Ir	K α_1	100	71,413	76 Os	K β_1	23
57,532	73 Ta	K α_1	100	64,948.8	73 Ta	K β_3	11	72,804.2	82 Pb	K α_2	60
57,981.7	74 W	K α_2	58	64,980	72 Hf	K β_2	7	72,871.5	81 Tl	K α_1	100
59,090	69 Tm	K β_2	7	65,112	78 Pt	K α_2	58	73,202.7	77 Ir	K β_3	12
59,140	70 Yb	K β_3	11	65,223	73 Ta	K β_1	22	73,363	76 Os	K β_2	8
59,318.2	74 W	K α_1	100	66,832	78 Pt	K α_1	100	73,560.8	77 Ir	K β_1	23
				66,951.4	74 W	K β_3	11	74,814.8	83 Bi	K α_2	60

1.3 FLUORESCENCE YIELDS FOR K, L, AND M SHELLS

Jeffrey B. Kortright

Fluorescence yields for the *K*, *L*, and *M* shells for the elements $3 \leq Z \leq 110$ are plotted in Fig. 1-2. The data are based on Ref. 1 for $Z \leq 100$, and on Ref. 2 for $Z > 100$ for *K* and *L* shells. These yields represent the probability of a core hole in the *K* or *L* shells being filled by a radiative process, in competition with nonradiative processes. Auger processes are the only nonradiative processes competing with fluorescence for the *K* shell and

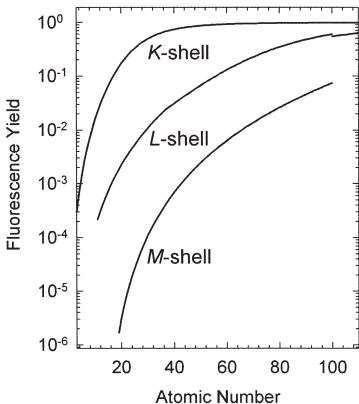


Fig. 1-2. Fluorescence yields for *K*, *L*, and *M* shells for $3 \leq Z \leq 110$. The plotted curves for the *L* and *M* shells represent average subshell effective yields.

L_3 subshell holes. Auger and Coster-Kronig nonradiative processes complete with fluorescence to fill L_1 and L_2 subshell holes. Only one curve is presented for the three L subshells, representing the average of the L_1 , L_2 , and L_3 effective fluorescence yields. Likewise the M shell curve represents an average over subshells. See Refs. 1 and 2 for details and citations to the theoretical and experimental work upon which Fig. 1-2 is based. Widths of K and L fluorescence lines can be found in Ref. 3.

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1.4 PRINCIPAL AUGER ELECTRON ENERGIES

Figure 1-3 has been reproduced by permission of Physical Electronics, Inc., and is taken from Ref. 1. For each element, dots indicate the energies of principal Auger peaks, the predominant ones represented by the heavier dots. The families of Auger transitions are denoted by labels of the form WXY , where W is the shell in which the original vacancy occurs, X is the shell from which the W vacancy is filled, and Y is the shell from which the Auger electron is ejected. The listed references should be consulted for detailed tabulations and for shifted values in several common compounds.

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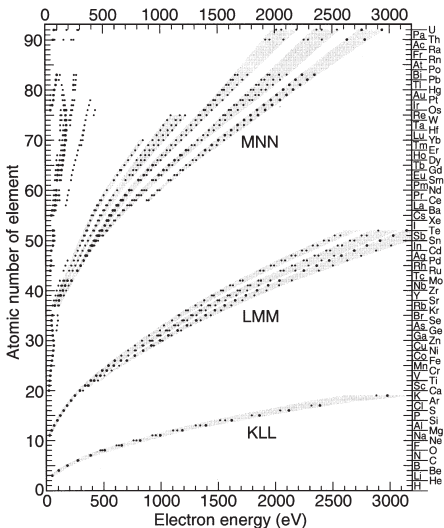


Fig. 1-3. Auger electron energies for the elements. Points indicate the electron energies of the principal Auger peaks for each element. The larger points represent the most intense peaks. (Reproduced by permission of Physical Electronics, Inc.)

1.5 SUBSHELL PHOTOIONIZATION CROSS SECTIONS

Ingolf Lindau

The atomic subshell photoemission cross sections plotted in Fig. 1-4 have been calculated for isolated atoms by Yeh and Lindau [1,2]. The calculations were done with a one-electron central-field frozen-core model using first-order perturbation theory. No single model accurately predicts the photoionization process of all orbitals for all elements from the VUV to 1.5 keV. The complexity of the physics of different atomic orbitals makes it impossible for any single rule to describe all of them. The accuracy of the model used has been discussed in detail by Cooper and Manson [3-5].

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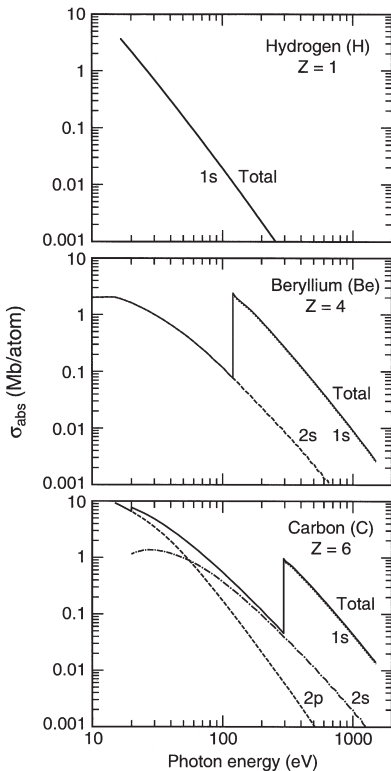


Fig. 1-4. Plots of atomic subshell photoemission cross sections, calculated for isolated atoms.

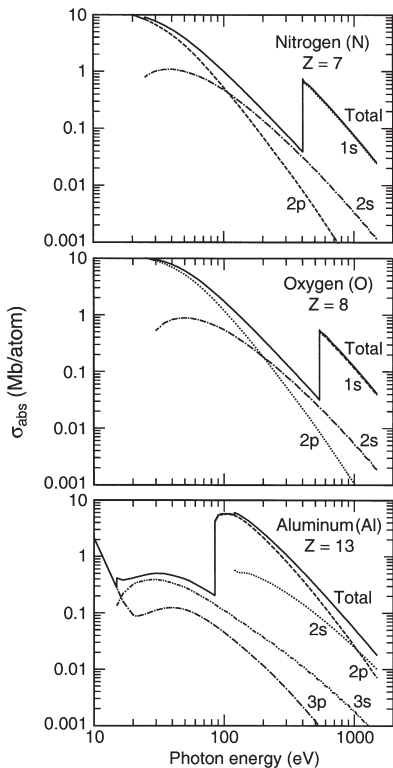


Fig. 1-4. Subshell photoemission cross sections (continued).

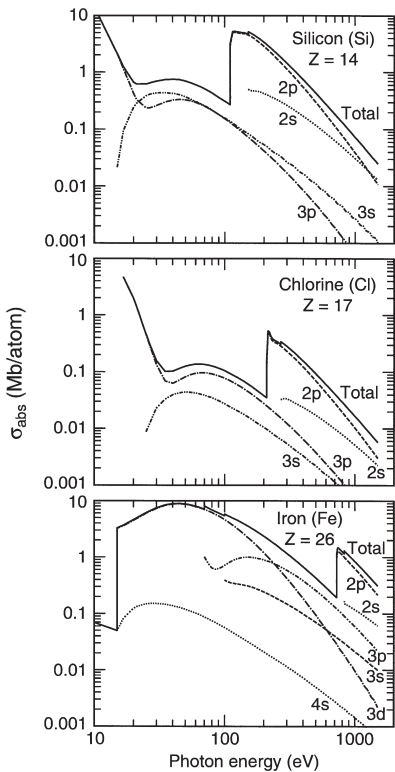


Fig. 1-4. Subshell photoemission cross sections (continued).

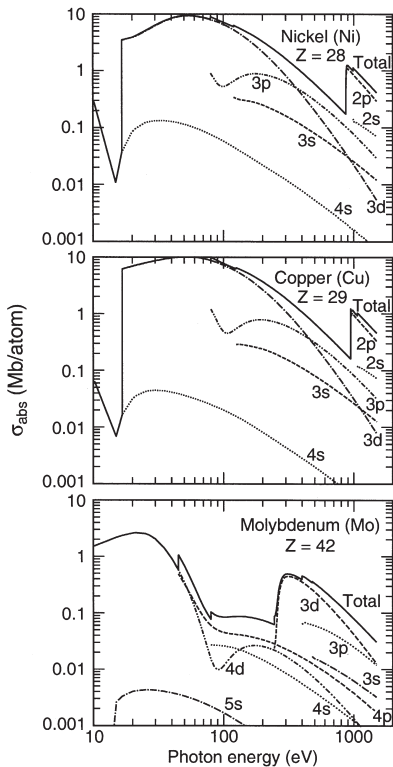


Fig. 1-4. Subshell photoemission cross sections (continued).

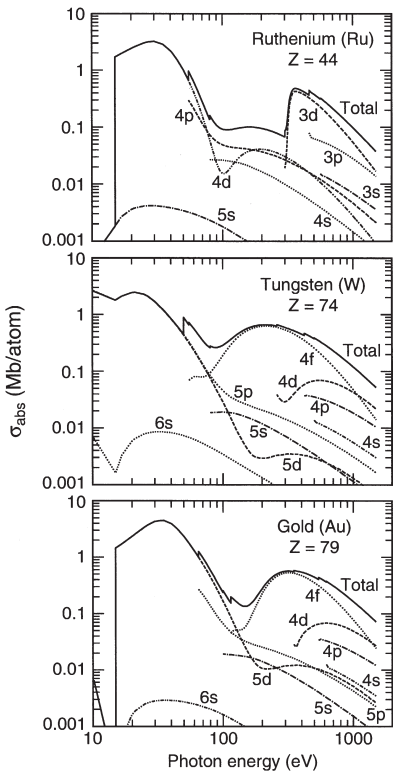


Fig. 1-4. Subshell photoemission cross sections (continued).

1.6 MASS ABSORPTION COEFFICIENTS

Eric M. Gullikson

Mass absorption coefficients have been tabulated for elements $Z \leq 92$, based on both measured values and theoretical calculations [see B. L. Henke, E. M. Gullikson, and J. C. Davis, "X-Ray Interactions: Photoabsorption, Scattering, Transmission, and Reflection at $E = 50\text{--}30,000$ eV, $Z = 1\text{--}92$," *At. Data Nucl. Data Tables* **54**, 181 (1993); for updated values, see http://www.cxro.lbl.gov/optical_constants/]. The mass absorption coefficient μ (cm^2/g) is related to the transmitted intensity through a material of density ρ (g/cm^3) and thickness d by

$$I = I_0 e^{-\mu \rho d} \quad (1)$$

Thus, the linear absorption coefficient is μ_ℓ (cm^{-1}) = $\mu\rho$. For a pure material, the mass absorption coefficient is directly related to the total atomic absorption cross section σ_a (cm^2/atom) by

$$\mu = \frac{N_A}{A} \sigma_a \quad (2)$$

where N_A is Avogadro's number and A is the atomic weight. For a compound material, the mass absorption coefficient is obtained from the sum of the absorption cross sections of the constituent atoms by

$$\mu = \frac{N_A}{MW} \sum_i x_i \sigma_{ai} \quad (3)$$

where the molecular weight of a compound containing x_i atoms of type i is $MW = \sum_i x_i A_i$. This approximation, which neglects interactions among the atoms in the material, is generally applicable for photon energies above about 30 eV and sufficiently far from absorption edges.

In Fig. 1-5, the mass absorption coefficient is plotted for 15 elements over the photon energy range 10–30,000 eV. In much of this range, the absorption coefficient is dominated by photoabsorption. However, for H, Be, C, N, and O, Compton (inelastic) scattering is significant at the higher energies. In these cases, the total cross section is shown as a solid curve and the photoabsorption cross section as a separate dashed curve.

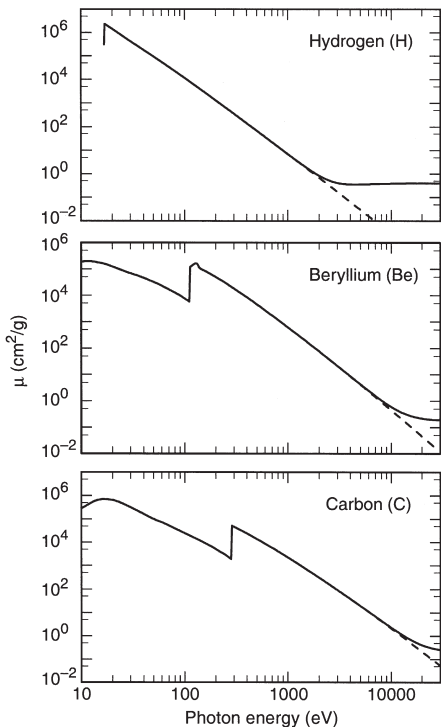


Fig. I-5. Plots of mass absorption coefficients for several elements in their natural forms. For H, Be, C, N, and O, the photoabsorption cross section is shown as a dashed curve.

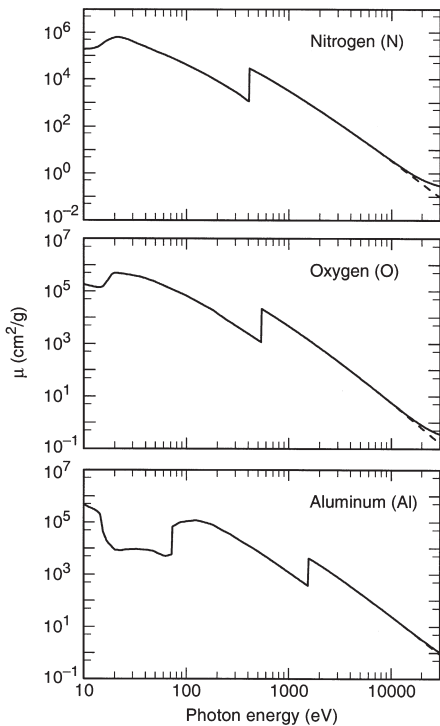


Fig. I-5. Mass absorption coefficients (continued).

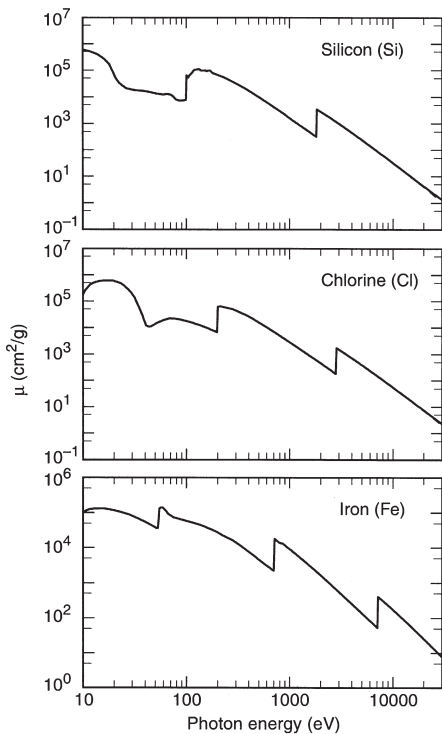


Fig. 1-5. Mass absorption coefficients (continued).

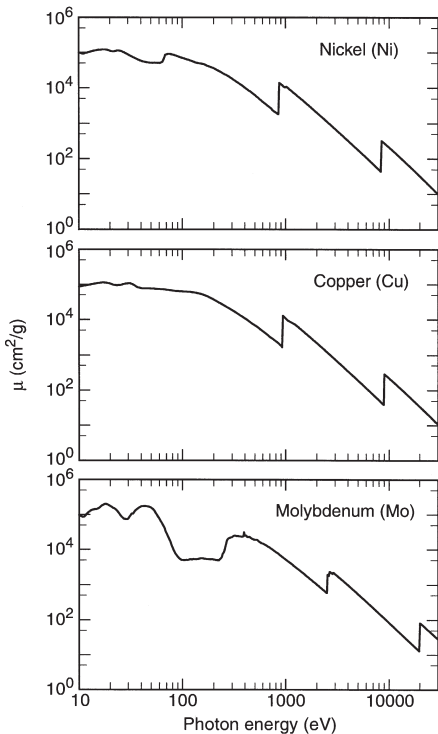


Fig. 1-5. Mass absorption coefficients (continued).

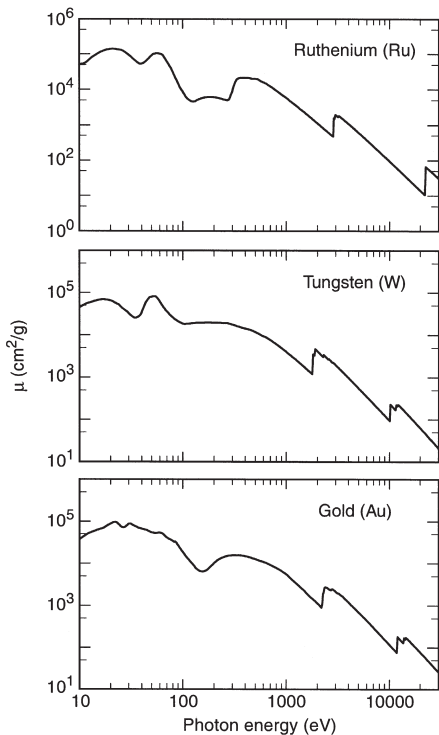


Fig. 1-5. Mass absorption coefficients (continued).

1.7 ATOMIC SCATTERING FACTORS

Eric M. Gullikson

The optical properties of materials in the photon energy range above about 30 eV can be described by the atomic scattering factors. The index of refraction of a material is related to the scattering factors of the individual atoms by

$$n = 1 - \delta - i\beta = 1 - \frac{r_e}{2\pi} \lambda^2 \sum_i n_i f_i(0) \quad , \quad (1)$$

where r_e is the classical electron radius, λ is the wavelength, and n_i is the number of atoms of type i per unit volume. The parameters δ and β are called the refractive index decrement and the absorption index, respectively. The complex atomic scattering factor for the forward scattering direction is

$$f(0) = f_1 + if_2 \quad . \quad (2)$$

The imaginary part is derived from the atomic photoabsorption cross section:

$$f_2 = \frac{\sigma_a}{2r_e\lambda} \quad . \quad (3)$$

The real part of the atomic scattering factor is related to the imaginary part by the Kramers-Kronig dispersion relation:

$$f_1 = Z^* + \frac{1}{\pi r_e hc} \int_0^\infty \frac{\varepsilon^2 \sigma_a(\varepsilon)}{E^2 - \varepsilon^2} d\varepsilon \quad . \quad (4)$$

In the high-photon-energy limit, f_1 approaches Z^* , which differs from the atomic number Z by a small relativistic correction:

$$Z^* \approx Z - (Z/82.5)^{2.37} \quad . \quad (5)$$

On the following pages, Fig. 1-6 presents the scattering factors for 15 elements in their natural forms. Complete tables are given in B. L. Henke, E. M. Gullikson, and J. C. Davis, "X-Ray Interactions: Photoabsorption, Scattering, Transmission, and Reflection at $E = 50\text{--}30,000$ eV, $Z = 1\text{--}92$," *At. Data Nucl. Data Tables* **54**, 181 (1993).

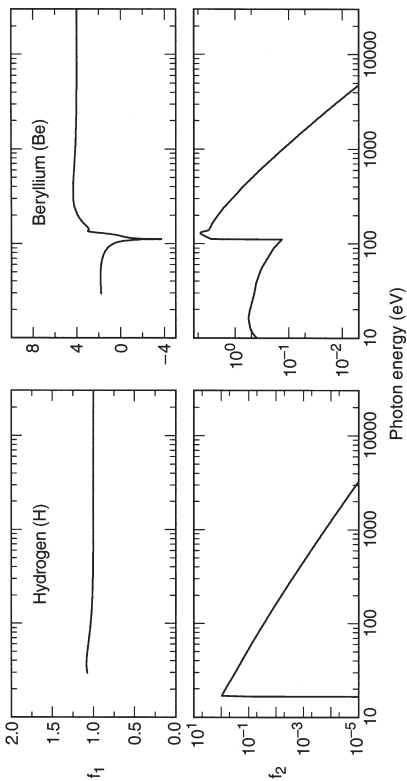


Fig. 1-6. Plots of scattering factors for several elements in their natural forms.

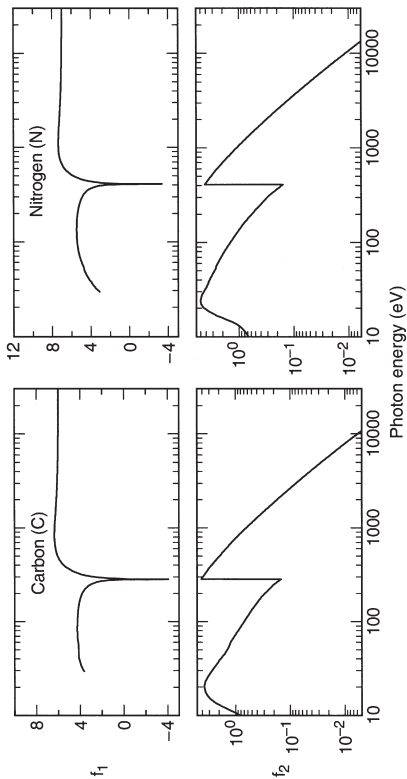


Fig. 1-6. Scattering factors (continued).

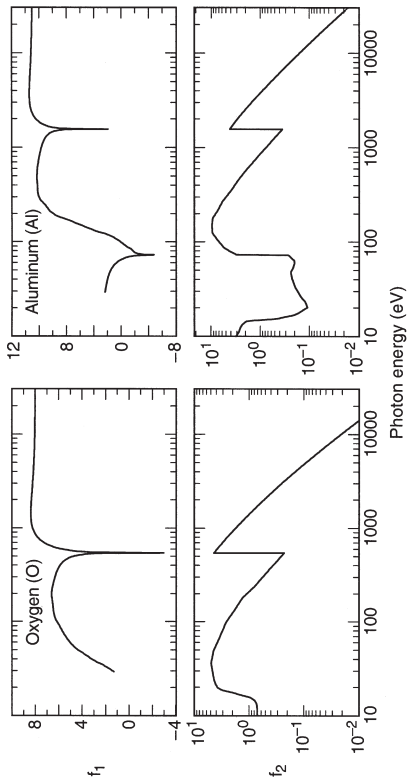


Fig. I-6. Scattering factors (continued).

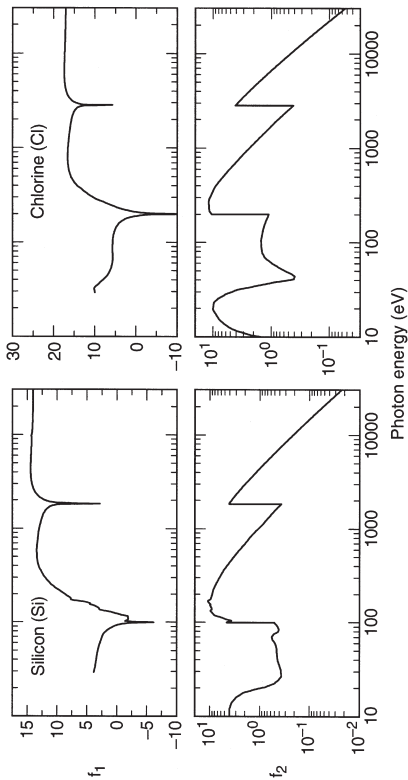


Fig. I-6. Scattering factors (continued).

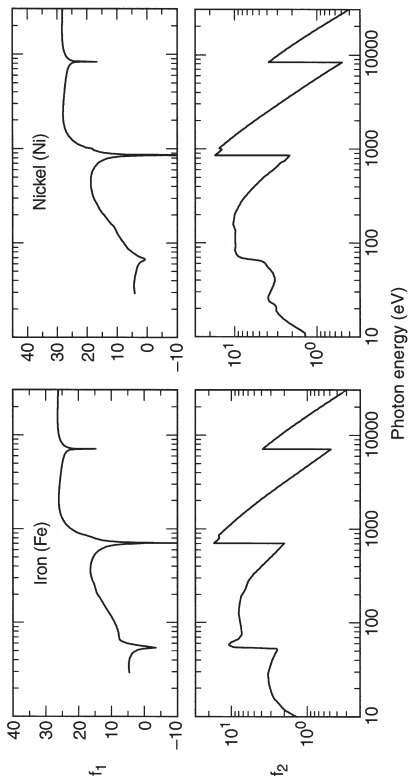


Fig. I-6. Scattering factors (continued).

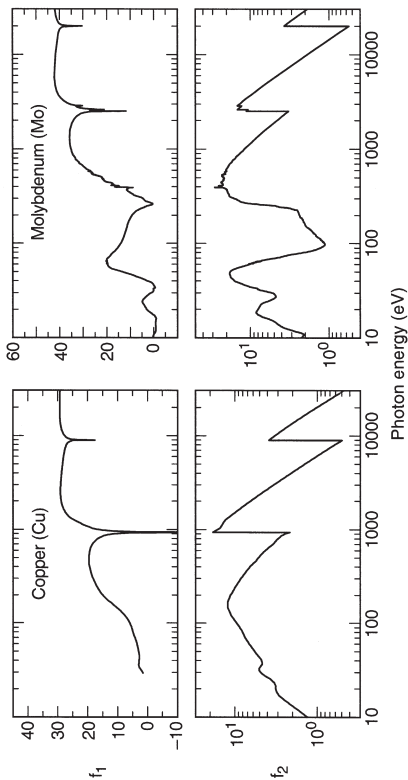


Fig. I-6. Scattering factors (continued).

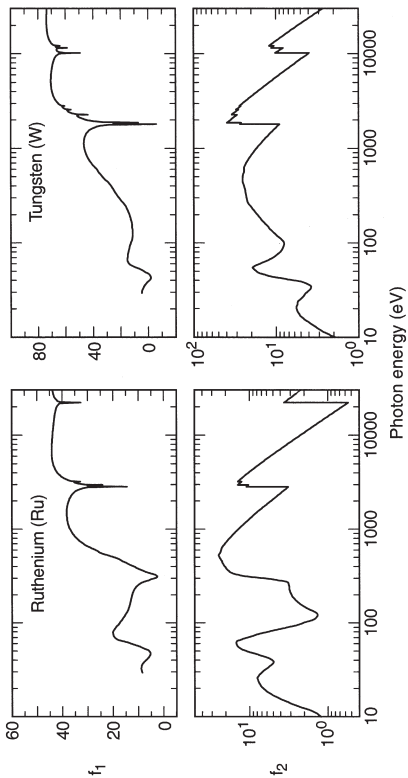


Fig. I-6. Scattering factors (continued).

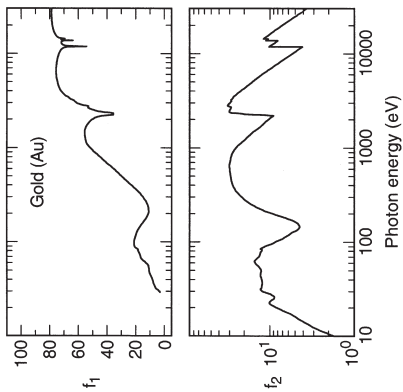


Fig. I-6. Scattering factors (continued).

1.8 ENERGY LEVELS OF FEW-ELECTRON IONIC SPECIES

James H. Scofield

Table 1-4 presents ionization energies for selected few-electron ions with $6 \leq Z \leq 54$. Table 1-5 gives the energies of the resonant $2p$ transitions in hydrogen- and heliumlike ions. The energy values in this section have been generated using the relativistic Hartree-Fock code of I. P. Grant and collaborators [1] with a correction term of the form $A + B/(Z - Q)$ added to bring about agreement with the experimental values known for low atomic numbers. Nuclear size effects, radiative corrections, and the Breit interaction accounting for retardation and the magnetic electron-electron interaction are included in the calculations. The hydrogenic values are uncorrected as they come from the code, but to the accuracy given here, they agree with more detailed calculations. The values in Table 1-4 for Co-, Ni-, and Cu-like ions are based on data from C. E. Moore [2], J. Sugar and A. Musgrove [3], and others referenced therein.

REFERENCES

1. I. P. Grant, B. J. McKenzie, P. H. Norrington, D. F. Mayers, and N. C. Pyper, "An Atomic Multiconfigurational Dirac-Fock Package," *Comput. Phys. Commun.* **21**, 207 (1980).
2. C. E. Moore, *Ionization Potentials and Ionization Limits Derived from the Analysis of Optical Spectra*, NBS Pub. NSRDS-NBS 34 (1970).
3. J. Sugar and A. Musgrove, "Energy Levels of Zinc, Zn I through Zn XXX," *J. Phys. Chem. Ref. Data* **24**, 1803 (1995).

Table 1-4. Ionization energies, in electron volts, for selected few-electron ionic species. Each column is labeled with the number of electrons in the ion before ionization and with the symbol for the neutral atom with the same number of electrons.

Element	1 (H)	2 (He)	3 (Li)	4 (Be)	10 (Ne)	11 (Na)	12 (Mg)	27 (Co)	28 (Ni)	29 (Cu)
6 C	490.0	392.1	64.49	47.89						
7 N	667.1	552.1	97.89	77.48						
8 O	871.4	739.3	138.11	113.90						
9 F	1103.1	953.9	185.18	157.15						
10 Ne	1362.2	1195.8	239.09	207.26	21.564					
11 Na	1648.7	1465.1	299.86	264.21	47.286	5.139				
12 Mg	1962.7	1761.8	367.5	328.0	80.143	15.035	7.646			
13 Al	2304.2	2086.0	442.0	398.7	119.99	28.447	18.828			
14 Si	2673.2	2437.7	523.4	476.3	166.42	45.12	33.64			
15 P	3070	2816.9	611.7	560.8	220.31	65.02	51.50			
16 S	3494	3224	707.0	652.1	281.00	88.05	72.59			
17 Cl	3946	3658	809.2	750.5	348.5	114.20	96.84			
18 Ar	4426	4121	918.4	855.8	422.8	143.46	124.24			
19 K	4934	4611	1034.6	968.0	503.9	175.82	154.75			
20 Ca	5470	5129	1157.7	1087.3	591.9	211.28	188.38			
21 Sc	6034	5675	1288.0	1213.6	686.6	249.84	225.13			
22 Ti	6626	6249	1425.3	1346.9	788.2	291.50	264.98			
23 V	7246	6851	1569.7	1487.3	896.6	336.3	307.9			
24 Cr	7895	7482	1721.2	1634.8	1011.8	384.2	354.0			
25 Mn	8572	8141	1879.9	1789.5	1133.8	435.2	403.2			
26 Fe	9278	8828	2045.8	1951.3	1262.7	489.3	455.6			

27 Co	10012	9544	2218.9	2120.4	1398.3	546.6	511.0	7.86	
28 Ni	10775	10289	2399.3	2296.7	1540.8	607.0	569.7	18.17	7.63
29 Cu	11568	11063	2587.0	2480.2	1690.2	670.6	631.4	36.83	20.29
30 Zn	12389	11865	2782.0	2671.1	1846.4	737.3	696.4	59.57	39.72
31 Ga	13239	12696	2984.4	2869.4	2009.4	807.3	764.5	86.0	63.4
32 Ge	14119	13557	3194	3075	2179.3	880.4	835.8	115.9	90.5
33 As	15029	14448	3412	3288	2356.0	956.8	910.3	149.2	121.2
34 Se	15968	15367	3637	3509	2539.6	1036.3	988.1	185.5	155.4
35 Br	16937	16317	3869	3737	2730.1	1119.1	1069.1	225.4	192.8
36 Kr	17936	17296	4109	3973	2927.4	1205.2	1153.3	268.2	233.4
37 Rb	18965	18306	4357	4216	3132	1294.5	1240.8	314.2	277.1
38 Sr	20025	19345	4612	4467	3343	1387.2	1331.5	363.3	324.1
39 Y	21115	20415	4876	4726	3561	1483.1	1425.6	413.6	374.0
40 Zr	22237	21516	5147	4993	3786	1582.4	1523.0	471	427.4
41 Nb	23389	22648	5426	5268	4017	1684.9	1623.7	530	483.8
42 Mo	24572	23810	5713	5550	4256	1790.9	1727.8	592	541.7
43 Tc	25787	25004	6008	5841	4502	1900.3	1835.2	656	605.8
44 Ru	27033	26230	6312	6140	4754	2013.0	1946.1	724	671.4
45 Rh	28312	27487	6623	6447	5014	2129.2	2060.3	795	740.1
46 Pd	29623	28776	6943	6762	5280	2248.9	2178.0	869	811.8
47 Ag	30966	30097	7271	7086	5553	2372.0	2299.2	946	886.6
48 Cd	32341	31451	7608	7418	5834	2498.6	2423.9	1026	964.5
49 In	33750	32837	7953	7758	6121	2628.8	2552.1	1109	1045.4
50 Sn	35192	34257	8307	8107	6415	2762.5	2683.9	1196	1129.1
51 Sb	36668	35710	8670	8465	6717	2899.8	2819.2	1285	1215.3
52 Te	38177	37196	9041	8832	7025	3041	2958.1	1377	1306.3
53 I	39721	38716	9421	9207	7340	3185	3101	1472	1399.3
54 Xe	41300	40271	9810	9591	7663	3334	3247	1571	1495.4

Table 1-5. *Transition energies, in electron volts, for transitions from the $n = 2$ states to the $n = 1$ ground state of H- and He-like ions.*

Element	Hydrogenlike		Heliumlike	
	$2p_{1/2}$	$2p_{3/2}$	$2p\ ^3P_1$	$2p\ ^1P_1$
5 B	255.17	255.20	202.78	205.37
6 C	367.5	367.5	304.3	307.8
7 N	500.3	500.4	426.3	430.7
8 O	653.5	653.7	568.7	574.0
9 F	827.3	827.6	731.5	737.8
10 Ne	1021.5	1022.0	914.9	922.1
11 Na	1236.3	1237.0	1118.8	1126.9
12 Mg	1471.7	1472.7	1343.2	1352.3
13 Al	1727.7	1729.0	1588.3	1598.4
14 Si	2004.3	2006.1	1853.9	1865.1
15 P	2301.7	2304.0	2140.3	2152.6
16 S	2619.7	2622.7	2447.3	2460.8
17 Cl	2958.5	2962.4	2775.1	2789.8
18 Ar	3318	3323	3124	3140
19 K	3699	3705	3493	3511
20 Ca	4100	4108	3883	3903
21 Sc	4523	4532	4295	4316
22 Ti	4966	4977	4727	4750
23 V	5431	5444	5180	5205
24 Cr	5917	5932	5655	5682
25 Mn	6424	6442	6151	6181
26 Fe	6952	6973	6668	6701
27 Co	7502	7526	7206	7242
28 Ni	8073	8102	7766	7806
29 Cu	8666	8699	8347	8392
30 Zn	9281	9318	8950	8999
31 Ga	9917	9960	9575	9628
32 Ge	10575	10624	10221	10280
33 As	11255	11311	10889	10955
34 Se	11958	12021	11579	11652
35 Br	12682	12753	12292	12372
36 Kr	13429	13509	13026	13114
37 Rb	14199	14288	13783	13880
38 Sr	14990	15090	14562	14669
39 Y	15805	15916	15364	15482
40 Zr	16643	16765	16189	16318
41 Nb	17503	17639	17036	17178
42 Mo	18387	18537	17907	18062
43 Tc	19294	19459	18800	18971
44 Ru	20224	20406	19717	19904
45 Rh	21178	21377	20658	20861
46 Pd	22156	22374	21622	21843
47 Ag	23157	23396	22609	22851
48 Cd	24183	24444	23621	23884
49 In	25233	25518	24657	24942
50 Sn	26308	26617	25717	26027

SECTION 2

SYNCHROTRON RADIATION

2.1A CHARACTERISTICS OF SYNCHROTRON RADIATION

Kwang-Je Kim

Synchrotron radiation occurs when a charge moving at relativistic speeds follows a curved trajectory. In this section, formulas and supporting graphs are used to quantitatively describe characteristics of this radiation for the cases of circular motion (bending magnets) and sinusoidal motion (periodic magnetic structures).

We will first discuss the ideal case, where the effects due to the angular divergence and the finite size of the electron beam—the emittance effects—can be neglected.

A. BENDING MAGNETS

The angular distribution of radiation emitted by electrons moving through a bending magnet with a circular trajectory in the horizontal plane is given by

$$\frac{d^2\mathcal{G}_b(\omega)}{d\theta d\psi} = \frac{3\alpha}{4\pi^2} \gamma^2 \frac{\Delta\omega I}{\omega e} y^2 (1 + X^2)^2 \quad (1)$$
$$\times \left[K_{2/3}^2(\xi) + \frac{X^2}{1 + X^2} K_{1/3}^2(\xi) \right],$$

where

- \mathcal{F}_B = photon flux (number of photons per second)
 θ = observation angle in the horizontal plane
 ψ = observation angle in the vertical plane
 α = fine-structure constant
 γ = electron energy/ $m_e c^2$ (m_e = electron mass, c = velocity of light)
 ω = angular frequency of photon ($\varepsilon = \hbar \omega$ = energy of photon)
 I = beam current
 e = electron charge = 1.602×10^{-19} coulomb
 y = $\omega' \omega_c = \varepsilon' \varepsilon_c$
 ω_c = critical frequency, defined as the frequency that divides the emitted power into equal halves,
 $= 3 \gamma^3 c / 2 \rho$
 ρ = radius of instantaneous curvature of the electron trajectory (in practical units,
 $\rho[\text{m}] = 3.3 E[\text{GeV}] / B[\text{T}]$)
 E = electron beam energy
 B = magnetic field strength
 ε_c = $\hbar \omega_c$ (in practical units,
 $\varepsilon_c [\text{keV}] = 0.665 E^2 [\text{GeV}] B[\text{T}]$)
 X = $\gamma \psi$
 ξ = $y(1 + X^2)^{3/2} / 2$

The subscripted K 's are modified Bessel functions of the second kind. In the horizontal direction ($\psi = 0$), Eq. (1) becomes

$$\left. \frac{d^2 \mathcal{F}_B}{d\theta d\psi} \right|_{\psi=0} = \frac{3\alpha}{4\pi^2} \gamma^2 \frac{\Delta\omega}{\omega} \frac{I}{e} H_2(y), \quad (2)$$

where

$$H_2(y) = y^2 K_{2/3}^2(y/2) \quad . \quad (3)$$

In practical units [photons \cdot s $^{-1}$ \cdot mr $^{-2}$ \cdot (0.1% bandwidth) $^{-1}$],

$$\left. \frac{d^2 \mathcal{F}_B}{d\theta d\psi} \right|_{\psi=0} = 1.327 \times 10^{13} E^2 [\text{GeV}] I [\text{A}] H_2(y).$$

The function $H_2(y)$ is shown in Fig. 2-1.

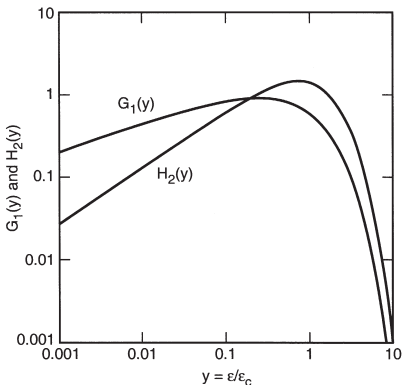


Fig. 2-1. The functions $G_1(y)$ and $H_2(y)$, where y is the ratio of photon energy to critical photon energy.

The distribution integrated over ψ is given by

$$\frac{d^3\mathbf{B}}{d\theta} = \frac{\sqrt{3}}{2\pi} \alpha \gamma \frac{\Delta\omega}{\omega} \frac{I}{e} G_1(y), \quad (4)$$

where

$$G_1(y) = y \int_y^\infty K_{5/3}(y') dy' \quad . \quad (5)$$

In practical units [photons \cdot s $^{-1}$ \cdot mr $^{-1}$ \cdot (0.1% bandwidth) $^{-1}$],

$$\frac{d^3\mathbf{B}}{d\theta} = 2.457 \times 10^{13} E[\text{GeV}] I[\text{A}] G_1(y).$$

The function $G_1(y)$ is also plotted in Fig. 2-1.

Radiation from a bending magnet is linearly polarized when observed in the bending plane. Out of this plane, the polarization is elliptical and can be decomposed into its horizontal and vertical components. The first

and second terms in the last bracket of Eq. (1) correspond, respectively, to the intensity of the horizontally and vertically polarized radiation. Figure 2-2 gives the normalized intensities of these two components, as functions of emission angle, for different energies. The square root of the ratio of these intensities is the ratio of the major and minor axes of the polarization ellipse. The sense of the electric field rotation reverses as the vertical observation angle changes from positive to negative.

Synchrotron radiation occurs in a narrow cone of nominal angular width $\sim 1/\gamma$. To provide a more specific measure of this angular width, in terms of electron and photon energies, it is convenient to introduce the effective rms half-angle σ_ψ as follows:

$$\left. \frac{d^2\mathcal{I}_B}{d\theta} \frac{d^2\mathcal{I}_B}{d\theta d\psi} \right|_{\psi=0} = \sqrt{2\pi} \sigma_\psi, \quad (6)$$

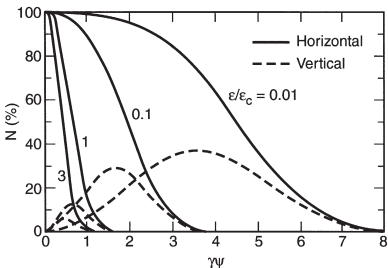


Fig. 2-2. Normalized intensities of horizontal and vertical polarization components, as functions of the vertical observation angle ψ , for different photon energies. (Adapted from Ref. 1.)

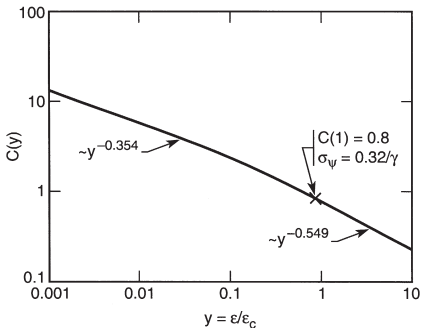


Fig. 2-3. The function $C(y)$. The limiting slopes, for $\epsilon/\epsilon_c \ll 1$ and $\epsilon/\epsilon_c \gg 1$, are indicated.

where σ_ψ is given by

$$\sigma_\psi = \frac{2}{\gamma\sqrt{2\pi}} C(y) = 0.408 \frac{C(y)[\text{mr}]}{E[\text{GeV}]} \quad (7)$$

The function $C(y)$ is plotted in Fig. 2-3. In terms of σ_ψ , Eq. (2) may now be rewritten as

$$\left. \frac{d^2 \mathfrak{F}_B}{d\theta d\psi} \right|_{\psi=0} = \frac{d\mathfrak{F}_B}{d\theta} \frac{1}{\sigma_\psi \sqrt{2\pi}} \quad (2a)$$

B. PERIODIC MAGNETIC STRUCTURES

In a wiggler or an undulator, electrons travel through a periodic magnetic structure. We consider the case where the magnetic field B varies sinusoidally and is in the vertical direction:

$$B(z) = B_0 \cos(2\pi z/\lambda_u) \quad , \quad (8)$$

where z is the distance along the wiggler axis, B_0 the peak magnetic field, and λ_u the magnetic period. Electron motion is also sinusoidal and lies in

the horizontal plane. An important parameter characterizing the electron motion is the deflection parameter K given by

$$K = eB_0\lambda_u / 2\pi mc = 0.934\lambda_u [\text{cm}] B_0[\text{T}] \quad . \quad (9)$$

In terms of K , the maximum angular deflection of the orbit is $\delta = K/\gamma$. For $K \leq 1$, radiation from the various periods can exhibit strong interference phenomena, because the angular excursions of the electrons are within the nominal $1/\gamma$ radiation cone; in this case, the structure is referred to as an undulator. In the case $K \gg 1$, interference effects are less important, and the structure is referred to as a wiggler.

B.1 Wiggler radiation

In a wiggler, K is large (typically ≥ 10) and radiation from different parts of the electron trajectory adds incoherently. The flux distribution is then given by $2N$ (where N is the number of magnet periods) times the appropriate formula for bending magnets, either Eq. (1) or Eq. (2). However, ρ or B must be taken at the point of the electron's trajectory tangent to the direction of observation. Thus, for a horizontal angle θ ,

$$\varepsilon_c(\theta) = \varepsilon_{c\text{max}} \sqrt{1 - (\theta / \delta)^2} \quad , \quad (10)$$

where

$$\varepsilon_{c\text{max}} = 0.665 E^2[\text{GeV}] B_0[\text{T}] \quad .$$

When $\psi = 0$, the radiation is linearly polarized in the horizontal plane, as in the case of the bending magnet. As ψ increases, the direction of the polarization changes, but because the elliptical polarization from one half-period of the motion combines with the elliptical polarization (of opposite sense of rotation) from the next, the polarization remains linear.

B.2 Undulator radiation

In an undulator, K is moderate (≤ 1) and radiation from different periods interferes coherently, thus producing sharp peaks at harmonics of the fundamental ($n = 1$). The wavelength of the fundamental on axis ($\theta = \psi = 0$) is given by

$$\lambda_1 = \frac{(1 + K^2/2)}{2\gamma^2} \lambda_u \quad (11)$$

or

$$\lambda_1[\text{\AA}] = \frac{13.056 \lambda_u[\text{cm}]}{E^2[\text{GeV}]} (1 + K^2/2) \quad .$$

The corresponding energy, in practical units, is

$$\varepsilon_1[\text{keV}] = 0.950 \frac{E^2[\text{GeV}]}{(1 + K^2/2)\lambda_u[\text{cm}]} \quad .$$

The relative bandwidth at the n th harmonic is

$$\frac{\Delta\lambda}{\lambda} \equiv \frac{\Delta\omega}{\omega} \equiv \frac{1}{nN} \quad (n = 1, 2, 3, \dots) \quad . \quad (12)$$

On axis the peak intensity of the n th harmonic is given by

$$\left. \frac{d^2\mathfrak{F}_n}{d\theta d\psi} \right|_0 = \alpha N^2 \gamma^2 \frac{\Delta\omega}{\omega} \frac{I}{e} F_n(K) \quad (n = 1, 3, 5, \dots)$$

$$= 0 \quad (n = 2, 4, 6, \dots) \quad (13)$$

where

$$F_n(K) = \frac{K^2 n^2}{(1 + K^2/2)^2} \left\{ J_{\frac{n-1}{2}} \left[\frac{nK^2}{4(1 + K^2/2)} \right] \right. \\ \left. - J_{\frac{n+1}{2}} \left[\frac{nK^2}{4(1 + K^2/2)} \right] \right\}^2 \quad . \quad (14)$$

Here, the J 's are Bessel functions. The function $F_n(K)$ is plotted in Fig. 2-4. In practical units [photons \cdot s $^{-1}\cdot$ mr $^{-2}\cdot$ (0.1% bandwidth) $^{-1}$], Eq. (13) becomes

$$\left. \frac{d^2\mathfrak{F}_n}{d\theta d\psi} \right|_0 = 1.744 \times 10^{14} N^2 E^2 [\text{GeV}] I [\text{A}] F_n(K).$$

The angular distribution of the n th harmonic is concentrated in a narrow cone whose half-width is given by

$$\sigma_r \equiv \sqrt{\frac{\lambda_n}{L}} = \frac{1}{\gamma} \sqrt{\frac{(1 + K^2/2)}{2Nn}} \quad . \quad (15)$$

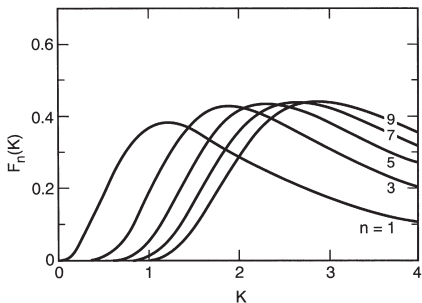


Fig. 2-4. The function $F_n(K)$ for different values of n , where K is the deflection parameter.

Here L is the length of the undulator ($L = N\lambda_w$). Additional rings of radiation of the same frequency also appear at angular distances

$$\theta_{n,\ell} = \frac{1}{\gamma} \sqrt{\frac{\ell}{n} (1 + K^2/2)} \quad (\ell = 1, 2, 3, \dots) \quad (16)$$

The angular structure of undulator radiation is illustrated in Fig. 2-5 for the limiting case of zero beam emittance.

We are usually interested in the central cone. An approximate formula for the flux integrated over the central cone is

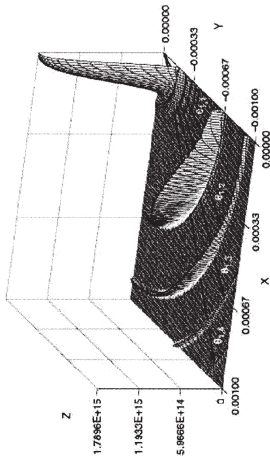
$$\mathcal{F}_n = \pi\alpha N \frac{\Delta\omega}{\omega} \frac{I}{e} Q_n(K), \quad (17)$$

or, in units of photons \cdot s $^{-1}$ \cdot (0.1% bandwidth) $^{-1}$,

$$\mathcal{F}_n = 1.431 \times 10^{14} N Q_n I [A].$$

The function $Q_n(K) = (1 + K^2/2)F_n/n$ is plotted in Fig. 2-6. Equation (13) can also be written as

Fig. 2-5. The angular distribution of fundamental ($n = 1$) undulator radiation for the limiting case of zero beam emittance. The x and y axes correspond to the observation angles θ_x and θ_y (in radians), respectively, and the z axis is the intensity in photons $\cdot s^{-1} \cdot A^{-1} \cdot (0.1 \text{ mr})^{-2} \cdot (1\% \text{ bandwidth})^{-1}$. The undulator parameters for this theoretical calculation were $N = 14$, $K = 1.87$, $\sigma_u = 3.5 \text{ cm}$, and $E = 1.3 \text{ GeV}$. (Figure courtesy of R. Taichyn, Stanford University.)



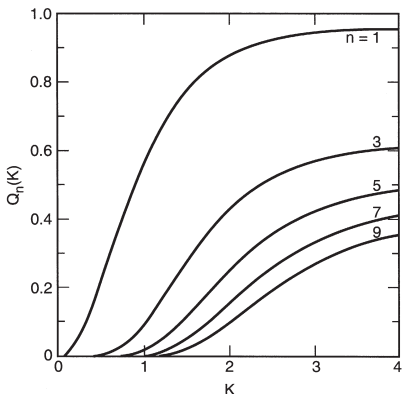


Fig. 2-6. The function $Q_n(K)$ for different values of n .

$$\left. \frac{d^2 \mathcal{F}_n}{d\theta d\psi} \right|_0 = \frac{\mathcal{F}_n}{2\pi\sigma_r^2} \quad (13a)$$

Away from the axis, there is also a change in wavelength: The factor $(1 + K^2/2)$ in Eq. (11) must be replaced by $[1 + K^2/2 + \gamma^2(\theta^2 + \psi^2)]$. Because of this wavelength shift with emission angle, the angle-integrated spectrum consists of peaks at λ_n superposed on a continuum. The peak-to-continuum ratio is large for $K \ll 1$, but the continuum increases with K , as one shifts from undulator to wiggler conditions.

B.3 Power

The total power radiated by an undulator or wiggler is

$$P_T = \frac{N}{6} Z_0 I_e \frac{2\pi c}{\lambda_u} \gamma^2 K^2 \quad , \quad (18)$$

where $Z_0 = 377$ ohms, or, in practical units,

$$P_T[\text{kW}] = 0.633 E^2[\text{GeV}] B_0^2[\text{T}] L[\text{m}] I[\text{A}] \quad .$$

The angular distribution of the radiated power is

$$\frac{d^2P}{d\theta d\psi} = P_T \frac{21\gamma^2}{16\pi K} G(K) f_K(\gamma\theta, \gamma\psi) \quad , \quad (19)$$

or, in units of $\text{W} \cdot \text{mr}^{-2}$,

$$\frac{d^2P}{d\theta d\psi} = 10.84 B_0[\text{T}] E^4 [\text{GeV}] I[\text{A}] N G(K) f_K(\gamma\theta, \gamma\psi) \quad .$$

The behavior of the angular function $f_K(\gamma\theta, \gamma\psi)$, which is normalized as $f_K(0,0) = 1$, is shown in Fig. 2-7. The function $G(K)$, shown in Fig. 2-8, quickly approaches unity as K increases from zero.

C. EMITTANCE EFFECTS

Electrons in storage rings are distributed in a finite area of transverse phase space—position \times angle. We introduce the rms beam sizes σ_x (horizontal) and σ_y (vertical), and beam divergences $\sigma_{x'}$ (horizontal) and $\sigma_{y'}$ (vertical). The quantities $\epsilon_x = \sigma_x \sigma_{x'}$ and $\epsilon_y = \sigma_y \sigma_{y'}$ are known as the horizontal and vertical emittances, respectively. In general, owing to the finite emittances of real electron beams, the intensity of the radiation observed in the forward direction is less than that given by Eqs. (2a) and (13a). Finite emittances can be taken into account approximately by replacing these equations by

$$\left. \frac{d^2\mathcal{F}_B}{d\theta d\psi} \right|_{\psi=0} = \frac{d\mathcal{F}_B}{d\theta} \frac{1}{\sqrt{2\pi(\sigma_\psi^2 + \sigma_{\psi'}^2)}} \quad (20)$$

and

$$\left. \frac{d^2\mathcal{F}_n}{d\theta d\psi} \right|_0 = \frac{\mathcal{F}_n}{2\pi\sqrt{(\sigma_r^2 + \sigma_{x'}^2)(\sigma_r^2 + \sigma_{y'}^2)}} \quad . \quad (21)$$

for bends and undulators, respectively. For bending magnets, the electron beam divergence effect is usually negligible in the horizontal plane.

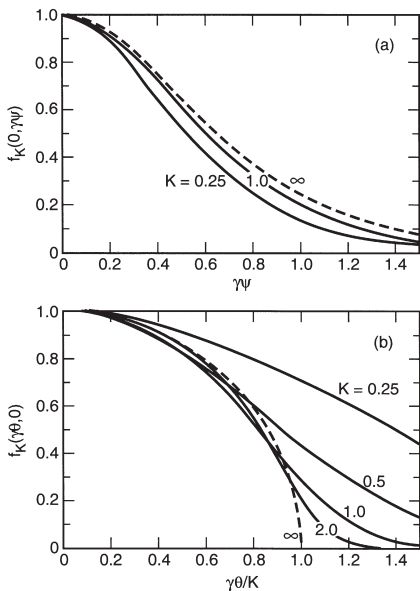


Fig. 2-7. The angular function f_K , for different values of the deflection parameter K , (a) as a function of the vertical observation angle ψ when the horizontal observation angle $\theta = 0$ and (b) as a function of θ when $\psi = 0$.

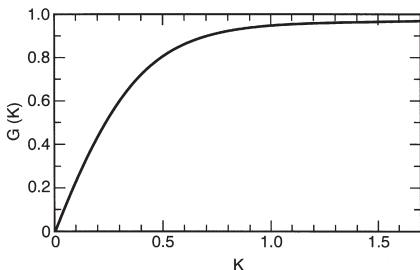


Fig. 2-8. The function $G(K)$.

D. SPECTRAL BRIGHTNESS AND TRANSVERSE COHERENCE

For experiments that require a small angular divergence and a small irradiated area, the relevant figure of merit is the beam brightness \mathcal{B} , which is the photon flux per unit phase space volume, often given in units of $\text{photons} \cdot \text{s}^{-1} \cdot \text{m}^{-2} \cdot \text{mm}^{-2} \cdot (0.1\% \text{ bandwidth})^{-1}$. For an undulator, an approximate formula for the peak brightness is

$$\mathcal{B}_n(0,0) = \frac{\mathcal{F}_n}{(2\pi)^2 \sigma_{Tx} \sigma_{Ty} \sigma_{Tx'} \sigma_{Ty'}}, \quad (22)$$

where, for example,

$$\begin{aligned} \sigma_{Tx} &= \sqrt{\sigma_x^2 + \sigma_y^2}, \\ \sigma_{Tx'} &= \sqrt{\sigma_{x'}^2 + \sigma_{r'}^2}, \end{aligned} \quad (23)$$

and where the single-electron radiation from an axially extended source of finite wavelength is described by

$$\begin{aligned} \sigma_r &= \frac{1}{4\pi} \sqrt{\lambda L}, \\ \sigma_{r'} &= \sqrt{\lambda / L}. \end{aligned} \quad (24)$$

Brightness is shown in Fig. 2-9 for several sources of synchrotron radiation, as well as some conventional x-ray sources.

That portion of the flux that is transversely coherent is given by

$$\mathcal{F}_c = \mathcal{B}_n \left(\frac{\lambda}{2} \right)^2 = \frac{\mathcal{F}_n \lambda^2}{(4\pi)^2 \sigma_{Tx} \sigma_{Ty} \sigma_{Tx'} \sigma_{Ty'}}. \quad (25)$$

A substantial fraction of undulator flux is thus transversely coherent for a low-emittance beam satisfying $\varepsilon_x \varepsilon_y \lesssim (\lambda/4\pi)^2$.

E. LONGITUDINAL COHERENCE

Longitudinal coherence is described in terms of a coherence length

$$\ell_c = \lambda^2 / \Delta\lambda \quad . \quad (26)$$

For an undulator, the various harmonics have a natural spectral purity of $\Delta\lambda/\lambda = 1/nN$ [see Eq. (12)]; thus, the coherence length is given by

$$\ell_c = nN\lambda \quad , \quad (27)$$

which corresponds to the relativistically contracted length of the undulator. Thus, undulator radiation from low-emittance electron beams [$\varepsilon_x \varepsilon_y \lesssim (\lambda/4\pi)^2$] is transversely coherent and is longitudinally coherent within a distance described by Eq. (27). In the case of finite beam emittance or finite angular acceptance, the longitudinal coherence is reduced because of the change in wavelength with emission angle. In this sense, undulator radiation is partially coherent. Transverse and longitudinal coherence can be enhanced when necessary by the use of spatial and spectral filtering (i.e., by use of apertures and monochromators, respectively).

The references listed below provide more detail on the characteristics of synchrotron radiation.

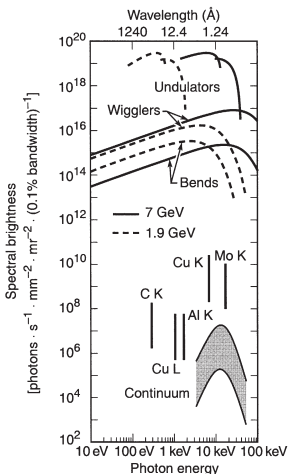


Fig. 2-9. Spectral brightness for several synchrotron radiation sources and conventional x-ray sources. The data for conventional x-ray tubes should be taken as rough estimates only, since brightness depends strongly on such parameters as operating voltage and take-off angle. The indicated two-order-of-magnitude ranges show the approximate variation that can be expected among stationary-anode tubes (lower end of range), rotating-anode tubes (middle), and rotating-anode tubes with microfocusing (upper end of range). The envelope of spectral brightness curves for today's third-generation synchrotron facilities is somewhat higher than the curves above due to increased average current (due to "top up" mode) and decreased electron beam emittance.

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2.1B. FREE ELECTRON LASERS (FELS) AT EXTREME ULTRAVIOLET AND X-RAY WAVELENGTHS

Yanwei Liu

As fourth-generation radiation facilities, FELs can generate very short duration, spatially coherent beams at short wavelengths. The lack of suitable mirrors prevents x-ray FELs from using a cavity oscillator. Instead, x-ray FELs are based on the Self Amplified Spontaneous Emission (SASE) process in a very long undulator [1]. The long undulator allows interaction between the radiated electric field and the electron beam to the extent that the light field can effectively modulate the energy of the electrons, forcing them into microbunches separated by one wavelength of the radiation (see Fig. 2-10). This microbunching process results in coherently radiating electrons and self-amplification of the light field, producing extremely bright, spatially coherent radiation. As the SASE process starts from shot noise, the x-ray FEL beams generally contain multiple spikes in their time structure, and thus have a complicated temporal phase and somewhat limited temporal coherence. Future FELs will likely use a seed pulse to improve temporal coherence.

Fully operational x-ray FELs include FLASH at DESY (Deutsches Elektronen-Synchrotron, Hamburg, Germany) and LCLS (Linac Coherent Light Source, Stanford, CA, USA). FLASH started user operation in August 2005, with a fundamental wavelength ranging from 6.5 nm (starting 2007) to 47 nm [2]. In April 2009, LCLS realized lasing in the hard x-ray region, with saturation at a wavelength of 1.5 Å [3]. X-ray FELs under construction worldwide include the SCSS at Japan, FERMI in Italy, the European XFEL, and others.

Examples of FEL Electron Beam and Photon Beam
 Characteristics at FLASH and LCLS

	FLASH	LCLS
Electron beam energy	0.2–1 GeV	13.6 GeV
Undulator period	2.73 cm	3 cm
Undulator length	30 m	112 m
Wavelength (fundamental)	6.5–47 nm	1.5 Å
Pulse energy	100 μJ	1.1 mJ
Peak power	5 GW	15 GW
Pulse duration	10–50 fs	75 fs
Peak brightness <i>[phs/(s mrad² mm² 0.1% BW)]</i>	$10^{29} - 10^{30}$	$10^{32} - 10^{33}$

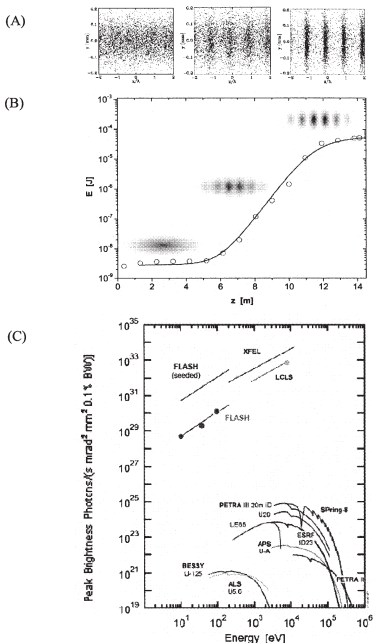


Figure 2-10. (A) Simulation of the electron microbunching process and (B) the exponential growth of FEL output energy due to microbunching (following S. Reiche and K.-J. Kim). (C) FEL beams have very short pulse duration and full spatial coherence, with peak brightness many orders of magnitude higher than 3rd generation synchrotron facilities [4, modified].

List of Some X-Ray FEL Facilities (operational and under construction)

FLASH at DESY (Deutsches Elektronen-Synchrotron),
Hamburg, Germany <http://flash.desy.de/>

LCLS (Linac Coherent Light Source) at Stanford, CA, USA
<http://lcls.slac.stanford.edu/>

SCSS (Spring-8 Compact SASE Source)
<http://www-xfel.spring8.or.jp/>

FERMI@Elettra, Trieste, Italy
www.elettra.trieste.it/FERMI/

European XFEL
<http://xfel.desy.de/>

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3. P. Emma, "First lasing of the *LCLS* X-ray FEL at 1.5 Å." for the *LCLS* Commissioning Team; *SLAC, Stanford, CA 94309, USA*. Available at <http://ssrl.slac.stanford.edu/lcls/commissioning/documents/th3pbi01.pdf>
4. H. Weise et al., "The TTF/VUV-FEL (FLASH) as the prototype for the European XFEL project." *Proceedings of LINAC 2006, Knoxville, TN, USA*

2.2 HISTORY OF SYNCHROTRON RADIATION

Arthur L. Robinson

Synchrotron radiation generated by relativistic electrons in circular accelerators is only a little more than 60 years old. The first observation—literally, since it was visible light that was generated—came at the General Electric Research Laboratory in Schenectady, New York, on April 24, 1947. In the six decades since, synchrotron radiation in the x-ray and ultraviolet spectral regions has become a premier research tool for the study of matter in all its varied manifestations, as facilities around the world have evolved to provide this light in ever more useful forms.

A. BACKGROUND

Both scientists and society have long recognized the exceptional importance of x-rays, beginning immediately after their discovery in 1895. The first Nobel Prize in Physics ever awarded went to Röntgen in 1901 “in recognition of the extraordinary services he has rendered by the discovery of the remarkable rays subsequently named after him.” By the time synchrotron radiation was observed almost 50 years later, the scientific use of x-rays was well established. Some highlights include the following:

- 1909: Barkla and Sadler discover characteristic x-ray radiation (1917 Nobel Prize to Barkla)
- 1912: von Laue, Friedrich, and Knipping observe x-ray diffraction (1914 Nobel Prize to von Laue)
- 1913: Bragg, father and son, build an x-ray spectrometer (1915 Nobel Prize)
- 1913: Moseley develops quantitative x-ray spectroscopy and Moseley’s Law
- 1916: Siegbahn and Stenstrom observe emission satellites (1924 Nobel Prize to Siegbahn)
- 1921: Wentzel observes two-electron excitations
- 1922: Meitner discovers Auger electrons
- 1924: Lindh and Lundquist resolve chemical shifts

- 1927: Coster and Druyvesteyn observe valence-core multiplets
- 1931: Johann develops bent-crystal spectroscopy

B. DISCOVERY OF SYNCHROTRON RADIATION

In the 1920s, physicists began contemplating magnetic-induction electron accelerators (betatrons) as machines to produce intense beams of x-rays by directing the accelerated beam onto a suitable target. Ivanenko and Pomeranchuk in the Soviet Union published calculations in 1944 showing that energy losses due to radiating electrons would limit the energy obtainable in a betatron. Subsequent theoretical work proceeded independently in the Soviet Union and in the U.S., where by 1945 Schwinger had worked out in considerable detail the classical (i.e., nonquantum) theory of radiation from accelerated relativistic electrons. After GE began testing a 100-MeV betatron in 1944, Blewett suggested a search for the radiation losses, which he expected to be significant at this energy. Although quantitative measurements reported in 1946 of the shrinking electron-orbit radius with energy were in accord with predicted losses, there was no direct observation of synchrotron radiation, owing to an opaque coating on the tube in which the electrons circulated.

Advances on another accelerator front led to the 1947 visual observation of synchrotron radiation at GE, where Pollack assembled a team to build a 70-MeV electron synchrotron to test the idea of phase stability in accelerators, independently proposed by McMillan in the U.S. and Veksler in the Soviet Union. Fortunately for the future of synchrotron radiation, the coating on the doughnut-shaped electron tube was transparent, which allowed a technician to look around the shielding with a mirror and see a bright arc of light, which the GE group quickly realized was actually coming from the electron beam. Their subsequent measurements launched the experimental investigation of the spectral and polarization properties of the radiation. Characterization measurements were also carried out in the 1950s at a 250-MeV synchrotron at the Lebedev Institute in Moscow.

The next big step came with the 1956 experiments of Tombouliau and Hartman, who were granted a two-week run at the 320-MeV electron synchrotron at Cornell. However,

despite the advantages of synchrotron radiation that were detailed by the Cornell scientists and the interest their work stimulated, it was not until 1961 that an experimental program using synchrotron radiation got under way when the National Bureau of Standards (now National Institute of Standards and Technology) modified its 180-MeV electron synchrotron to allow access to the radiation.

C. THE FIRST GENERATION

Under Madden and Codling, measurements began at the new NBS facility (Synchrotron Ultraviolet Radiation Facility or SURF) to determine the potential of synchrotron radiation for standards and as a source for spectroscopy in the ultraviolet. Their findings further stimulated the growing interest in synchrotron radiation. Establishment of SURF began the first generation of synchrotron radiation facilities. If SURF headed the first generation, it was not by much, as activity was also blossoming in both Europe and Asia. At the Frascati laboratory near Rome, researchers began measuring absorption using a 1.15-GeV synchrotron. In 1962 scientists in Tokyo formed the INS-SOR (Institute for Nuclear Studies-Synchrotron Orbital Radiation) group and by 1965 were measuring soft x-ray absorption spectra of solids using light from a 750-MeV synchrotron. The trend toward higher energies and shorter wavelengths took a big leap with the use of the 6-GeV Deutsches Elektronen-Synchrotron (DESY) in Hamburg, which provided synchrotron radiation at wavelengths in the x-ray region down to 0.1 Å.

While the number of synchrotrons with budding synchrotron radiation facilities was growing, the next major advance came with the development of storage rings, the basis for all of today's synchrotron sources. In the 1950s, the Midwest Universities Research Association was formed to develop a proposal for a high-current accelerator for particle physics. As part of the project, Mills and Rowe designed a 240-MeV storage ring as a test bed for advanced accelerator concepts. Politics intervened, however, and funding disappeared. Instead, supported by the U.S. Air Force Office of Scientific Research, the University of Wisconsin took on the responsibility of completing the storage ring, known as Tantalus I, and operating it for synchrotron radiation research. The first spectrum was

measured in 1968. With Tantalus I, the superiority of the electron storage ring as a source of synchrotron radiation became evident.

A surge of interest in storage rings soon followed. In 1971 synchrotron radiation work began on the 540-MeV ACO storage ring at the Orsay laboratory in France. NBS converted its synchrotron into a 250-MeV storage ring (SURF II) in 1974. The same year, the INS-SOR group in Tokyo began commissioning a 300-MeV storage ring, generally considered the first machine designed from the start specifically for the production of synchrotron radiation. The first storage ring in the multi-GeV class to provide x-rays to a large community of synchrotron radiation users was the 2.5-GeV SPEAR ring at the Stanford Linear Accelerator Center under the auspices of the Stanford Synchrotron Radiation Project in 1974. Other large storage rings to which synchrotron radiation capabilities were added early on include DORIS at the DESY laboratory, VEPP-3 at the Institute for Nuclear Physics in Novosibirsk, DCI at Orsay, and CESR at Cornell (the CHESS facility).

D. THE SECOND GENERATION: DEDICATED SOURCES

Parasitic operation at high-energy physics facilities often meant a severely limited output of synchrotron radiation, thereby motivating a demand for storage rings designed for and dedicated to the production of synchrotron radiation. The Synchrotron Radiation Source (SRS) at the Daresbury Laboratory in the UK was the first of the resulting "second generation" of sources. Experiments began at the new facility in 1981. That same year in the U.S., construction of the National Synchrotron Light Source (NSLS) at the Brookhaven National Laboratory was completed and commissioning began. The NSLS complex included separate 700-MeV and 2.5-GeV storage rings for production of UV radiation and x-rays, respectively. During this same period, the University of Wisconsin Synchrotron Radiation Center built a new 1-GeV storage ring named Aladdin, which superseded the old Tantalus I. In Japan the Photon Factory was completed in 1982 at the KEK laboratory in Tsukuba; in Berlin the BESSY facility began serving users in 1982 with an 800-MeV storage ring; and at Orsay LURE (Laboratoire pour l'Utilisation du Rayonnement Elec-

tromagnétique) began operating an 800-MeV storage ring, SuperACO, in 1984.

Elsewhere, some of the first-generation facilities gradually evolved toward second-generation status by means of upgrades and agreements to dedicate a fraction and sometimes all of the yearly machine operations to synchrotron radiation as the high-energy physics frontier advanced. The Stanford Synchrotron Radiation Laboratory at SLAC and HASYLAB (Hamburger Synchrotronstrahlungslabor) at DESY are prime examples.

E. BRIGHTNESS

As the clamor for facilities dedicated to synchrotron radiation expanded in the 1970s, users increasingly appreciated that spectral brightness, or brilliance, was often more important than flux alone for many experiments. As the flux density in phase space, brightness is an invariant quantity, so that no optical technique can improve it. The cure therefore is proper design of the source, the electron beam in the storage ring. As planning for NSLS progressed, Chasman and Green designed what has become the prototype magnet lattice (a so-called double-bend achromat) for storage rings with a low emittance (product of beam size and divergence) and hence a light source with high brightness. The Chasman-Green lattice and variations are the basis for most of today's synchrotron sources.

Undulators provide a way to take maximum advantage of the intrinsic brightness of the synchrotron radiation source. They produce one or a few spectrally narrow peaks (a fundamental and harmonics) in a beam that is highly collimated in both the horizontal and the vertical directions. Wigglers are similar to undulators but produce a continuous spectrum with a higher flux and a spectrum that extends to shorter wavelengths than bend magnets. Together, wigglers and undulators are called insertion devices because they are placed in the otherwise empty straight sections that connect the curved arcs of large storage rings.

The undulator concept traces back to the 1947 theoretical work of Ginzburg in the Soviet Union. Motz and coworkers experimentally verified the idea in 1953 by building an undulator and using it to produce radiation from the millimeter-wave to the visible range in experiments with a linear accelerator at Stanford University. The next step came in the 1970s

with the installation of undulators in storage rings at the Lebedev Institute in Moscow and the Tomsk Polytechnic Institute. Undulators became practical devices for producing synchrotron radiation in storage rings in 1981 when Halbach at the Lawrence Berkeley Laboratory and coworkers constructed a device based on permanent magnets and successfully tested it at SSRL. Parallel work was also under way in Novosibirsk. A decade after the initial suggestion by Robinson, a wiggler was installed in 1966 at the Cambridge Electron Accelerator to enhance beam storage. In 1979 a wiggler comprising just seven electromagnet poles at SSRL was the first to be used for producing synchrotron radiation. Nowadays, wigglers may be permanent-magnet devices following the Halbach design or be based on high-field superconductors that shift the spectrum to the shortest wavelengths.

F. THE THIRD GENERATION: OPTIMIZED FOR BRIGHTNESS

Synchrotron users recognized that a new generation of storage rings with a still lower emittance and long straight sections for insertion devices would permit achieving even higher brightness and with it a considerable degree of spatial coherence. Following the NSLS two-ring model, third-generation facilities specialize in either hard x-rays or vacuum-ultraviolet radiation and soft x-rays. The European Synchrotron Radiation Facility (ESRF) in Grenoble was the first of the third-generation hard x-ray sources to operate, coming on line for experiments by users with a 6-GeV storage ring and a partial complement of commissioned beamlines in 1994. The ESRF was followed by the Advanced Photon Source at Argonne National Laboratory (7 GeV) in late 1996 and by SPring-8 (8 GeV) in Harima Science Garden City in Japan in late 1997.

Among the soft x-ray sources, the Advanced Light Source at Berkeley (1.9 GeV) began its scientific program in early 1994, as did the Synchrotrone Trieste (2.0 GeV) in Italy, followed by the Synchrotron Radiation Research Center (1.3 GeV) in Hsinchu, Taiwan, and the Pohang Light Source (2.0 GeV) in Korea. Many other soft x-ray facilities have been and are being constructed around the world. Addition of superconducting bend magnets to the storage-ring lattice in these smaller machines, as the Advanced Light Source has

done, will allow them to extend their spectral coverage to higher photon energies without sacrificing performance at lower photon energies. At the same time, storage rings at SSRL (SPEAR3) and HASYLAB (DORIS III) have been reconfigured to push them into the third-generation regime.

Following on the success of the third-generation sources, a newer round of third-and-a-half-generation facilities with still higher brightness has come into operation. Added to an often higher beam energy (typically 2.5 to 3 GeV) and circumference and advanced insertion devices (such as narrow gaps and the coming use of superconducting technology), these machines are filling in the spectral gap between the high- and low-energy facilities. Examples include the Swiss Light Source at the Paul Scherrer Institut in Switzerland; Soleil in Saclay, France; Diamond at the CCLRC Rutherford Appleton Laboratory in the UK; the Canadian Light Source at the University of Saskatchewan, Canada; the Australian Synchrotron in Melbourne; and the Shanghai Synchrotron Radiation Facility in China.

Even more advanced storage rings are also on the horizon. The PETRA III machine (6 GeV) now in operation at HASYLAB takes advantage of its 2.3-km circumference to reach an ultralow emittance. Construction has started on NSLS-II at Brookhaven on an ultrahigh-performance medium energy machine (3 GeV). Other similar projects have been proposed at locations around the world.

G. NEXT: THE FOURTH GENERATION

Plans to develop a new generation—the fourth—of synchrotron radiation sources with vastly enhanced performance are now reaching fruition. With the fourth generation, the accelerator technology shifts from storage rings to linear accelerators. In the x-ray and ultraviolet spectral regions now served by synchrotron radiation facilities, the technologies fall into two broad classes: free electron lasers (FELs) and energy recovery linacs (ERLs).

FELs differ from conventional lasers in that they use the electron beam as the lasing medium rather than a gas or a solid. To date, FELs available at longer wavelengths have been based on a high-precision insertion device and an optional optical cavity formed by mirrors. Current and future generations of

FELs will access shorter wavelengths by use of a single-pass configuration, in which the laser energy builds during the passage of each individual electron bunch, without the use of an optical cavity. The FEL can generate fully coherent light, can have peak brightnesses up to one billion times higher than that of ordinary synchrotron light, and can produce ultrashort pulses of 10 fs or shorter, properties that facilitate time-resolved investigations of extremely rapid processes. (See Section 2.1b for a fuller discussion of FELs.)

FLASH at HASYLAB is the first free-electron laser for VUV and soft x-ray radiation, having begun user operation in 2005. Currently it covers a wavelength range from 6.5 nm to about 50 nm with pulse durations between 10 fs and 50 fs. The Linac Coherent Light Source at SLAC is now undergoing commissioning and is the first facility designed specifically to produce hard x rays in the 1-Å range in 100-fs pulses. Many other projects, some including more advanced concepts to further improve performance or broaden the range of parameters available, are under way or under study in Asia, Europe, and the USA.

An Energy Recovery Linac (ERL) x-ray source takes advantage of recent advances in superconducting linear accelerators and in high-brightness electron sources. ERLs have the potential to generate synchrotron radiation with an average brightness about 1000 times greater than that of today's storage rings, together with ultrashort pulses, suitable for studying the dynamics of materials on extremely rapid time scales, and with intense x-ray nanoprobe beams to study samples that are spatially inhomogeneous on a nanometer scale or are simply ultra-tiny, also on the nanoscale.

While both ERLs and XFELs will be able to produce very fast x-ray pulses, the two sources are quite distinct in the timing of these pulses: ERLs are being designed to produce pulse times up to a billion times a second, whereas the first XFELs will produce bigger pulses with the capability of temporal as well as spatial coherence, but at a far lower repetition rate. In the meantime, the technology is advancing extremely rapidly, and advanced concepts may blur the distinction between these two extremes.

2.3 SYNCHROTRON RADIATION FACILITIES

Herman Winick and David Attwood

The number of synchrotron radiation facilities is growing rapidly. As a result, the Internet is the most reliable source of up-to-date information on facilities around the world. A very useful site is www.lightsources.org. A list of present facilities and their websites follows.

Facility	Website
Aarhus Storage Ring, Denmark	www.isa.au.dk
Advanced Light Source, Berkeley, CA	www.als.lbl.gov
Advanced Photon Source, Argonne, IL	www.aps.anl.gov
ALBA - Consortium for the Exploitation of the Synchrotron Light Laboratory, Barcelona, Spain	www.cells.es
ANKA-Angstromquelle Karlsruhe, Germany	www.anka-online.de
Australian Synchrotron, Melbourne, Australia	www.synchrotron.vic.gov.au
Beijing Synchrotron Radiation Facility, Beijing, China	www.ihep.ac.cn/bsrf/english
BESSY, Berlin, Germany	www.bessy.de
Canadian Light Source, Saskatoon, Saskatchewan, Canada	www.lightsource.ca
Center for Advanced Microstructures and Devices, Baton Rouge, LA	www.camd.lsu.edu
Cornell High Energy Synchrotron Sources, Cornell, NY	www.chess.cornell.edu
Dafne, Frascati, Italy	www.inf.infn.it
Deutsches Elektronen-Synchrotron, Hamburg, Germany	www-hASYLAB.desy.de
Diamond Light Source, Didcot, Oxfordshire, England	www.diamond.ac.uk
Dortmund Electron Test Accelerator, Dortmund, Germany	www.delta.uni-dortmund.de
Electron Stretcher Accelerator, Bonn, Germany	http://www-eisa.physik.uni-bonn.de/eisa-facility_en.html
Elettra Synchrotron Light Laboratory, Trieste, Italy	www.elettra.trieste.it
European Synchrotron Radiation Facility, Grenoble, France	www.esrf.fr

Facility	Website
Hiroshima Synchrotron Radiation Center, Hiroshima, Japan	www.hsrc.hiroshima-u.ac.jp
Indus, Indore, India	www.cat.lernet.in/technology/accel/indus/index.html
Laboratorio Nacional de Luz Sincrotron, Campinas, Brazil	www.inls.br
Linear Coherent Light Source, Stanford CA	http://lcls.siac.stanford.edu/
MAX-lab, Lund, Sweden	www.maxlab.lu.se
Metrology Light Source, Berlin, Germany	www.ptb.de/mls/
National Synchrotron Light Source, Upton, NY	www.nsls.bnl.gov
National Synchrotron Radiation Laboratory, Hefei, China	www.ustc.edu.cn/en/srl.htm
NewSUBARU, Nishi-Harima, Hyogo, Japan	www.lasfi.himeji-tech.ac.jp/NS/Index.html
Petra III	http://hasylab.desy.de/facilities/petra_iii/index_eng.html
Photon Factory, Tsukuba, Japan	http://pfwww.kek.jp
Pohang Light Source, Pohang, Korea	http://pal.postech.ac.kr
MIRRORCLE - Ritsumeikan, Kusatsu, Japan	www.ritsumeai.ac.jp/se/d11/index-e.html
SAGA-Light Source, Kyushu Synchrotron Light Research Center, Japan	http://www.saga-ls.jp/?page=206
SESAME, Amman, Jordan	www.sesame.org.jo
Shanghai Synchrotron Radiation Facility, Shanghai, China	http://ssrf.sinap.ac.cn/english/1/Introduction.htm
Siberia-2, Kurchatov Institute, Moscow, Russia	www.kiae.ru
Singapore Synchrotron Light Source, Singapore	www.nus.edu.sg/NUSinfo/SSLS

Facility	Website
SOLEIL, Saint-Aubin, France	www.synchrotron-soleil.fr
Spring-8, Nishi-Harima, Hyogo, Japan	www.spring8.or.jp
Stanford Synchrotron Radiation Lightsource, Stanford, CA	www-ssrl.siac.stanford.edu
SURF, NIST, Gaithersburg, MD	http://physics.nist.gov/MajResFac/SURF/SURF.html
Swiss Light Source, Villigen, Switzerland	http://sls.web.psi.ch/view.php/about/index.html
Synchrotron Light Research Institute, Nakhon Ratchasima, Thailand	http://sut2.sut.ac.th
Synchrotron Radiation Center, Stoughton, WI	www.src.wisc.edu
Synchrotron Radiation Research Center, Hsinchu, Taiwan	www.srrc.gov.tw/en/main2000.htm
UVSOR Facility, Myodaiji, Okazaki, Japan	www.uvsor.ims.ac.jp
VEPP-3, VEPP-4, Budker INP, Novosibirsk, Russia	http://ssrc.inp.nsk.su

SECTION 3

SCATTERING PROCESSES

3.1 SCATTERING OF X-RAYS FROM ELECTRONS AND ATOMS

Janos Kirz

A. COHERENT, RAYLEIGH, OR ELASTIC SCATTERING

Scattering from single electrons (Thomson scattering) has a total cross section

$$\sigma_T = 8\pi r_e^2 / 3 = 6.652 \times 10^{-29} \text{ m}^2 \quad , \quad (1)$$

where r_e is the classical radius of the electron, $e^2 / mc^2 = 2.818 \times 10^{-15}$ meter. The angular distribution for unpolarized incident radiation is proportional to $(1 + \cos^2 \theta)$, where θ is the scattering angle. For polarized incident radiation, the cross section vanishes at 90° in the plane of polarization.

Scattering from atoms involves the cooperative effect of all the electrons, and the cross section becomes

$$\sigma_R = \pi r_e^2 \int_{-1}^1 |f(\theta)|^2 (1 + \cos^2 \theta) d(\cos \theta) \quad , \quad (2)$$

where $f(\theta)$ is the (complex) atomic scattering factor, tabulated in Section 1.7 of this booklet. Up to about 2 keV, the scattering factor is approximately independent of scattering angle, with a real part that represents the effective number of electrons that participate in the scattering. At higher energies, the scattering factor falls off rapidly with scattering angle. For details see Ref. 1.

B. COMPTON SCATTERING

In relativistic quantum mechanics, the scattering of x-rays by a free electron is given by the Klein-Nishina formula. If we assume unpolarized x-rays and unaligned electrons, this formula can be approximated as follows for x-ray energies below 100 keV:

$$d\sigma_{\text{KN}} / d\Omega \cong \frac{r_e^2 (1 + \cos^2 \theta)}{2[1 + k(1 - \cos \theta)]^2} \quad , \quad (3)$$

where $k = E / mc^2$, the photon energy measured in units of the electron rest energy. The total cross section is approximately

$$\sigma_{\text{KN}} \cong 8\pi r_e^2 \frac{(1 + 2k + 1.2k^2)}{3(1 + 2k)^2} \quad . \quad (4)$$

Note that for very low energies ($k \rightarrow 0$), we recover the Thomson cross section. The real difference comes when we deal with atoms. In that case, if the scattering leaves the atom in the ground state, we deal with coherent scattering (see above), whereas if the electron is ejected from the atoms, the scattering is (incoherent) Compton scattering. At high energies, the total Compton cross section approaches $Z\sigma_{\text{KN}}$. At low energies and small scattering angles, however, binding effects are very important, the Compton cross section is significantly reduced, and coherent scattering dominates (see Figs. 3-1 and 3-2). For details see Refs. 1 and 2.

The scattered x-ray suffers an energy loss, which (ignoring binding effects) is given by

$$E' / E = 1 / [1 + k(1 - \cos \theta)] \quad (5)$$

or, in terms of the wavelength shift,

$$\lambda' - \lambda = \lambda_c (1 - \cos \theta) \quad , \quad (6)$$

where $\lambda_c = h / mc = 2.426 \times 10^{-12}$ meter. The kinetic energy of the recoil electron is just the energy lost by the photon in this approximation:

$$E_e = E \frac{k(1 - \cos \theta)}{1 + k(1 - \cos \theta)} \quad . \quad (7)$$

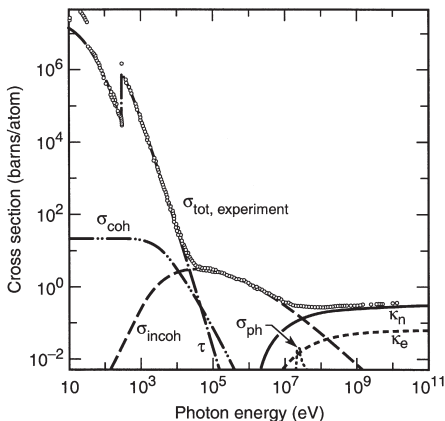


Fig. 3-1. Total photon cross section σ_{tot} in carbon, as a function of energy, showing the contributions of different processes: τ , atomic photo-effect (electron ejection, photon absorption); σ_{coh} , coherent scattering (Rayleigh scattering—atom neither ionized nor excited); σ_{incoh} , incoherent scattering (Compton scattering off an electron); κ_n , pair production, nuclear field; κ_e , pair production, electron field; σ_{ph} , photonuclear absorption (nuclear absorption, usually followed by emission of a neutron or other particle). (From Ref. 3; figure courtesy of J. H. Hubbell.)

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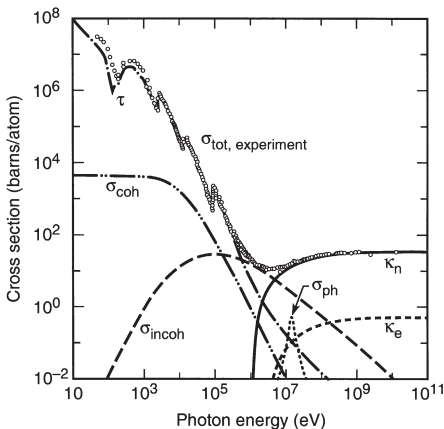


Fig. 3-2. Total photon cross section σ_{tot} in lead, as a function of energy. See Fig. 3-1. (From Ref. 3; figure courtesy of J. H. Hubbell.)

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3.2 LOW-ENERGY ELECTRON RANGES IN MATTER

Piero Pianetta

The electron range is a measure of the straight-line penetration distance of electrons in a solid [1]. Electrons with energies in the kilo-electron volt range, traveling in a solid, are scattered inelastically in collisions with the electrons in the material. For low- Z materials, such as organic insulators, scattering from the valence electrons is the major loss mechanism for incident electron energies from 10 eV to 10 keV. The core levels contribute less than 10% to the electron's energy dissipation for energies between 1 keV and 10 keV [2].

A. CSDA RANGES

For electron energies below 5 keV, the usual Bethe-Bloch formalism is inadequate for calculating the electron energy loss in a solid, and an approach using the dielectric response of the material is used [3]. The complex dielectric function $\epsilon(k, \omega)$ describes the response of a medium to a given energy transfer $\hbar\omega$ and momentum transfer $\hbar k$. The dielectric function contains contributions from both valence and core electrons. References 4 and 5 describe the steps for calculating $\epsilon(k, \omega)$ for insulators and metals, respectively. For an electron of energy E , the probability of an energy loss ω per unit distance is given by [2]

$$\tau(E, \hbar\omega) = \frac{1}{\pi a_0 E} \int_{k_-}^{k_+} \frac{dk}{k} \operatorname{Im} \left[\frac{-1}{\epsilon(k, \omega)} \right], \quad (1)$$

where $\hbar k_{\pm} = \sqrt{2m(\sqrt{E} \pm \sqrt{E - \hbar\omega})}$ and $a_0 = \hbar^2 / me^2$. The quantity $\tau(E, \hbar\omega)$ is also known as the differential inverse mean free path, because by integrating it over all allowed energy transfers, the inelastic mean free path (IMFP) is obtained. Furthermore, an integration of $\hbar\omega\tau(E, \hbar\omega)$ over all allowed energy transfers gives the energy loss per unit path length, or stopping power $S(E)$. The stopping power can then be used to calculate the distance it takes to slow an electron down

to a given energy. This distance is called the continuous slowing down approximation range, or CSDA range, because the calculation assumes that the electron slows down continuously from the initial energy E to the final energy, which is usually taken to be 10 eV [2]. The CSDA range $R_0(E)$ is given by

$$R_0(E) = \int_{10\text{eV}}^E \frac{dE'}{S(E')} \quad (2)$$

The calculations for IMFP and stopping power have been carried out down to 10 eV for a number of materials, including SiO_2 [3]; polystyrene [2]; polyethylene [6]; collodion [7]; and silicon, aluminum, nickel, copper, and gold [5]. The CSDA ranges from 15 eV to 6 keV were then calculated for polystyrene, silicon, and gold by integrating Eq. (2) and are shown in Fig. 3-3. These curves can be used with confidence down to 100 eV. However, comparisons of different available calculations with the meager experimental data below 100 eV indicate that errors as large as 100% may occur at 10 eV. An example of this is shown in the figure, where experimental range data for collodion are given. It is clear that the agreement between the collodion and polystyrene data starts to become reasonable above 100 eV. The differences below 100 eV could equally well be due to problems with the theory or to the increased difficulty of the measurement. Stopping-power calculations for polymethyl methacrylate (PMMA) have been carried out only from 100 eV, so that the CSDA range as defined above could not be calculated [4]. However, data on effective electron ranges of photoelectrons in PMMA at several energies can be found in Ref. 8.

B. ELECTRON INELASTIC MEAN FREE PATHS

A very important aspect of photoelectron spectroscopy, especially with synchrotron radiation, is the ability to effectively tune the surface sensitivity from a few angstroms or a few tens of angstroms in core-level photoemission measurements to a few hundred angstroms in total-electron-yield surface EXAFS experiments. This variation arises from the fact that the IMFP of the photoemitted electrons is a strong function of the electron kinetic energy, which can be tuned by the appropriate choice of photon energy. The definition of the IMFP

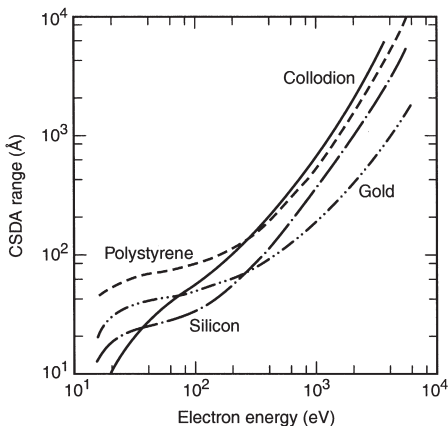


Fig. 3-3. Plot of the CSDA range, as a function of energy, for gold and silicon [5] and for polystyrene, $(C_8H_8)_n$, with a density of 1.05 g/cm^3 [2]. The measured electron range in collodion with a density of 1 g/cm^3 is also plotted [7].

[9] is the average distance traveled by an electron between inelastic collisions. Although the exact relationship between the IMFP and kinetic energy depends on the detailed electronic structure of the element or compound of interest, the general features are similar for all elements, starting at large values for kinetic energies below 10–15 eV, dropping to a minimum value of 5–10 Å at kinetic energies between 30 and 100 eV, and then rising monotonically above 100 eV.

Since the surface sensitivity is determined by the depth *perpendicular* to the surface from which electrons can escape, it is best defined using the mean escape depth (MED), which is related to the IMFP by

$$\Delta = \lambda_i \cos \alpha \quad , \quad (3)$$

where Δ is the MED, λ_i is the IMFP, and α is the emission angle of the electrons relative to the surface normal. However, it should be noted that *elastic* scattering effects within the solid could increase the MED as much as a factor of two at electron emission angles greater than 60° , depending on the angle of incidence of the incoming x-rays and the particular core level being studied [9,10]. Therefore, the standard technique of increasing the surface sensitivity by working at glancing emission angles using Eq. (3) must be qualified to take these effects into account. In addition, both angle-dependent cross sections and photoelectron diffraction effects can result in anisotropic emission from the solid that can also cause errors in the interpretation of the MEDs in solids. Because of these complications, graphs of the IMFPs, rather than the MEDs, versus electron kinetic energy will be presented here to give a measure of the surface sensitivity. The reader is referred to Ref. 9 when more complicated experimental conditions need to be considered.

Using the formalism developed by Penn that uses optical data to determine the IMFP of a material [11], Tanuma et al. have calculated the IMFPs for a large number of elements and compounds for kinetic energies up to 2000 eV [12–14]. Figure 3-4 shows IMFP curves for Ag, Al, Na, PMMA, Si, and SiO₂. These materials are representative of a fairly wide variety of materials for kinetic energies between 200 and 2000 eV. For example, the IMFPs for Ni, Ta, W, Pt, and Au all hover around the values given here for Ag; Cr, Fe, and Cu fall between Al and Ag. Likewise, C falls between Si and SiO₂, whereas GaAs overlies the PMMA curve for much of this energy range. The behavior below 200 eV is more complex, because the IMFPs are strongly dependent on the details of the electronic structure. Figure 3-5 shows the region below 250 eV for Al, Ag, GaAs, Na, PMMA, and Si. Silicon dioxide is not shown here because it overlaps the PMMA curve in this range, whereas GaAs does not. Although the calculations below 50 eV may not be reliable, owing to limitations in the theory, the values are plotted at these low energies to show the general behavior of the IMFPs in this region, as well as the location of the minima for the different materials. Calculations for additional materials

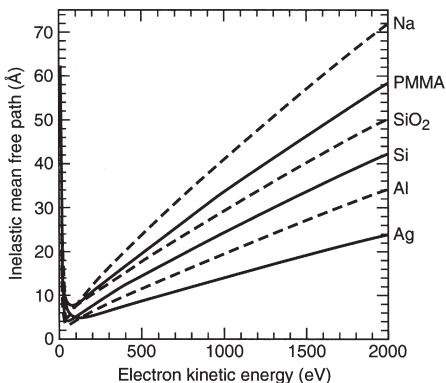


Fig. 3-4. *Inelastic mean free paths for electron kinetic energies up to 2000 eV, for Ag, Al, Na, PMMA, Si, and SiO₂.*

can be found in the literature as follows: (i) elements from C to Bi [12]; (ii) III-V and II-VI compound semiconductors, alkali halides, Si₃N₄, and several oxides [13]; and (iii) organic compounds [14]. Calculations are being presented here because they provide the most complete and consistent set of values for the IMFPs. References 9 and 10 give the historical background for both the theory and the experimental work in this field and show that it is difficult to generalize much of the experimental data in the literature, owing to the experiment-specific effects described above, as well as uncertainties in sample preparation. Seah and Dench [15] were the first to classify the material dependence of the IMFPs and presented data for kinetic energies up to 10 keV. A good example of the care that is needed in determining IMFPs is given in Ref. 8, which is a study of the Si/SiO₂ system. Finally, it should be mentioned that spin-dependent effects on the IMFP have also been observed in ferromagnetic materials [17].

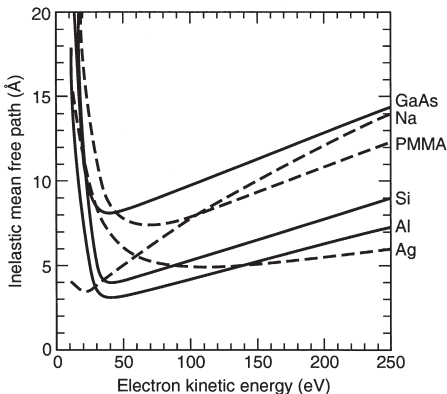


Fig. 3-5. Detail of the inelastic mean free paths in the kinetic energy range below 250 eV, for Ag, Al, GaAs, Na, PMMA, and Si.

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SECTION 4

OPTICS AND DETECTORS

4.1 MULTILAYERS AND CRYSTALS

James H. Underwood

A. MULTILAYERS

Multilayered structures consisting of alternating layers of high- and low- Z materials, with individual layers having thicknesses of the order of nanometers, can be fabricated using sputtering or evaporation on suitable smooth substrates. These structures act as multilayer interference reflectors for x-rays, soft x-rays, and extreme ultraviolet (EUV) light. Their high reflectivity and moderate energy bandwidth ($10 < E/\Delta E < 100$) make them a valuable addition to the range of optical components useful in instrumentation for EUV radiation and x-rays with photon energies from a few hundred eV to tens of keV. Multilayer reflectors have a wide range of applications in the EUV region, where normal-incidence multilayer reflectors allow the construction of space telescopes and of optics for EUV lithography. Multilayers are useful as mirrors and dispersive elements on synchrotron radiation beamlines. They may be used to produce focal spots with micrometer-scale sizes and for applications such as fluorescent microprobing, microdiffraction, and microcrystallography.

Ordinary mirrors are useful in the EUV and x-ray regions when operated at glancing incidence (see Section 4.2). However, at near-normal incidence angles the reflectivity is typically very small. The reason is that the value of the complex refractive index, $n = 1 - \delta - i\beta$, is close to unity for all materials in this region. Although the normal-incidence intensity reflectivity of a surface

might be 10^{-2} to 10^{-4} , the corresponding *amplitude* reflectivity, $r = (n - 1)/(n + 1)$, which is the square root of the intensity reflectivity, will be $1/10$ to $1/100$. This implies that, if the reflections from 10–100 surfaces could be made to add in phase, a total reflectivity approaching unity could be obtained. This is the multilayer principle. As shown in Fig. 4-1, a multilayer reflector comprises a stack of materials having alternately high and low refractive indices. The thicknesses are adjusted so that the path length difference between reflections from successive layer pairs is equal to one wavelength. Hence, x-ray/EUV multilayers are

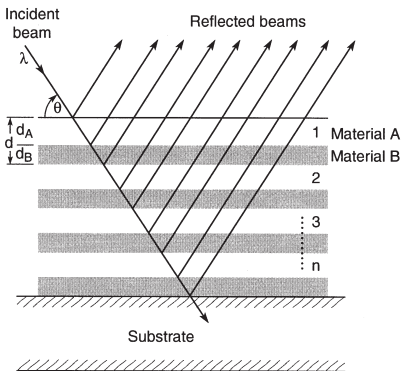


Fig. 4-1. Schematic of a multilayer reflector of n bilayer pairs. The parameters λ , θ , and d are chosen to satisfy the familiar Bragg equation, but the relative thicknesses of the high- and low- Z materials are also critical in optimizing reflectivity. The total reflectivity is the vector sum of the complex reflection coefficients at each interface, with the different path lengths taken into account.

approximately equivalent to the familiar “quarter-wave stacks” of visible-light coating technology. The equivalence is not exact, however, because of the absorption term β , which is usually negligible for visible-light multilayers. This absorption reduces the multilayer reflectivity below unity and requires the design to be optimized for highest reflectivity. This optimization normally requires modeling or simulation of the multilayer. Most thin-film calculation programs, even if designed for the visible region, will perform these calculations if given the right complex values of the refractive index. (An on-line calculation program is available at <http://www.cxro.lbl.gov/>.) Either elemental materials or compounds can be used to fabricate multilayer reflectors. The performance obtained from a multilayer depends largely on whether there exists a fortuitous combination of materials having the right refractive indices (good contrast, low absorption), whether these materials can be deposited in smooth thin layers, and whether they remain stable (low reactivity, low diffusion) in the deposited state. The roughness of the underlying substrate is also of prime importance; an rms roughness of the same order of magnitude as the layer thicknesses will spoil the performance of most coatings. “Superpolished” substrates, with roughness $\sigma \approx 0.1$ nm, are available. On such substrates, the peak reflectivity of the coatings can approach 80% to 90% of theoretical predictions.

A large variety of multilayers have been fabricated over the years. A database of reported reflectivities has been assembled from surveys taken at the biennial Physics of X-ray Multilayer Structures conferences and can be found at the website listed in Fig. 4-2. The reported near-normal incidence reflectivity is shown in Fig. 4-2 for photon energies below 600 eV. The highest reflectivity ($\sim 70\%$) is obtained for the EUV region around 100 eV by multilayers Mo-Si and Mo-Be (see Fig. 4-2). This relatively high reflectivity is the basis for current efforts in the field of EUV lithography. At hard x-ray wavelengths near 10 keV, a commonly used multilayer is made with tungsten as the high-Z material and boron carbide (B_4C) as the low-Z material. A reflectivity of 84% has been achieved with this combination at 8048 eV, the energy of the Cu $K\alpha$ line (see Fig. 4-3).

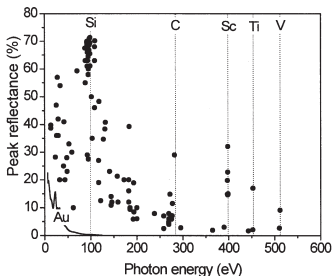


Fig. 4-2. The peak reflectivity achieved for multilayer mirrors at near-normal incidence versus photon energy. The data are based on the results of surveys taken at the biennial Physics of X-Ray Multilayer Structures conferences (maintained at <http://www.cxro.lbl.gov/multilayer/survey.html>). The normal incidence reflectivity of a gold mirror is shown for reference.

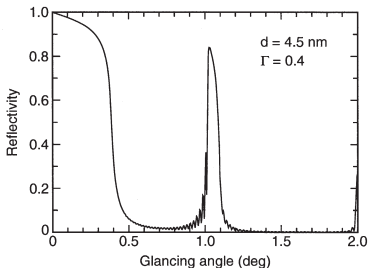


Fig. 4-3. The reflectivity of a tungsten-boron carbide multilayer at 8048 eV. The parameters d and Γ are discussed in the text.

B. CRYSTALS

Dispersion of radiation by a periodic structure is formally equivalent to multiple-beam interferometry. Crystals and multilayer structures produce the N interfering beams by division of the incident *amplitude*. (Structures periodic across their surface and that produce the N interfering beams by division of the incident wave front are called gratings and are treated in Section 4.3.) The spectrum of the incident radiation is dispersed in angle according to the Bragg equation $n\lambda = 2d \sin \theta$, where n is an integer representing the order of the reflection, λ is the wavelength of the incident radiation, d is the period of the multilayer or crystal structure, and θ is the angle of glancing incidence. For a crystal, d is the lattice spacing, the perpendicular distance between the successive planes of atoms contributing to the reflection. These planes are designated by their Miller indices [(hkl) or, in the case of crystals belonging to the hexagonal group, $(hkil)$]. The value of $2d$ also represents the longest wavelength that the structure can diffract.

For $2d$ values greater than about 25 \AA , the choice of natural crystals is very limited, and those available (such as prochlorite) are likely to be small and of poor quality. Sputtered or evaporated multilayers can be used as dispersing elements at longer wavelengths. (Langmuir-Blodgett films have seen limited use since the development of vacuum-deposited multilayers.)

Table 4-1 is a revision of one compiled by E. P. Bertin [1]. The crystals are arranged in order of increasing $2d$ spacing.

REFERENCE

1. E. P. Bertin, "Crystals and Multilayer Langmuir-Blodgett Films Used as Analyzers in Wavelength-Dispersive X-Ray Spectrometers," in J. W. Robinson, Ed., *Handbook of Spectroscopy* (CRC Press, Cleveland, 1974), vol. 1, p. 238.

Table 4-1. Data for selected crystals used as dispersive elements in x-ray spectrometers and monochromators. The Miller indices [(hkl), or (hkl) for hexagonal crystals] are given for the diffracting planes parallel to the surface of the dispersive element. A question mark (?) indicates that the crystal is developmental and that the indices have not been ascertained. An asterisk following the indices indicates that literature references to this crystal without specification of (hkl) or 2d are likely to be references to this "cut." The indicated useful wavelength region lies in the 2θ interval between 10° and 140°. The analyzer should be used outside this region in special cases only.

No.	Crystal	Miller indices	2d (Å)	Chemical formula	Useful wavelength region (Å)	Applications, remarks
1	α-Quartz, silicon dioxide	(50 $\bar{5}2$)	1.624	SiO ₂	0.142–1.55	Shortest 2d of any practical crystal. Good for high-Z K-lines excited by 100-kV generators.
2	Lithium fluoride	(422)	1.652	LiF	0.144–1.58	Better than quartz (50 $\bar{5}2$) for the same applications.
3	Corundum, aluminum oxide	(146)	1.660	Al ₂ O ₃	0.145–1.58	Same applications as quartz (50 $\bar{5}2$)
4	Lithium fluoride	(420)	1.801	LiF	0.157–1.72	Similar to LiF (422).
5	Calcite, calcium carbonate	(633)	2.02	CaCO ₃	0.176–1.95	
6	α-Quartz, silicon dioxide	(2243)	2.024	SiO ₂	0.177–1.96	
7	α-Quartz, silicon dioxide	(3140)	2.3604	SiO ₂	0.205–2.25	Transmission-crystal optics.
8	α-Quartz, silicon dioxide	(2240)	2.451	SiO ₂	0.213–2.37	

9	Topaz, hydrated aluminum fluorosilicate	(303)*	2.712	$\text{Al}_2(\text{F,OH})_2\text{SiO}_4$	0.236–2.59	Improves dispersion for V–Ni <i>K</i> -lines and rare earth <i>L</i> -lines.
10	Corundum, aluminum oxide, sapphire, alumina	(030)	2.748	Al_2O_3	0.240–2.62	Diffracted intensity ~2–4X topaz (303) and quartz (203) with the same or better resolution.
11	α -Quartz, silicon dioxide	(20 $\bar{2}$ 3)	2.749	SiO_2	0.240–2.62	Same applications as topaz (303) and LiF (220).
12	Topaz	(006)	2.795	$\text{Al}_2(\text{F,OH})_2\text{SiO}_4$	0.244–2.67	
13	Lithium fluoride	(220)	2.848	LiF	0.248–2.72	Same applications as topaz (303) and quartz (203), with 2–4X their diffracted intensity. Diffracted intensity ~0.4–0.8X LiF (200).
14	Mica, muscovite	(331)	3.00	$\text{K}_2\text{O}\cdot 3\text{Al}_2\text{O}_3\cdot 6\text{SiO}_2\cdot 2\text{H}_2\text{O}$	0.262–2.86	Transmission-crystal optics (Cauchois, DuMond types).
15	Calcite, calcium carbonate	(422)	3.034	CaCO_3	0.264–2.93	
16	α -Quartz, silicon dioxide	(21 $\bar{3}$ 1)	3.082	SiO_2	0.269–2.94	
17	α -Quartz, silicon dioxide	(11 $\bar{2}$ 2)	3.636	SiO_2	0.317–3.47	
18	Silicon	(220)	3.8403117	Si	0.335–3.66	Lattice period known to high accuracy.
19	Fluorite, calcium fluoride	(220)	3.862	CaF_2	0.337–3.68	
20	Germanium	(220)	4.00	Ge	0.349–3.82	

Table 4-1. Selected data for crystals (continued).

No.	Crystal	Miller indices	$2d$ (Å)	Chemical formula	Useful wavelength region (Å)	Applications, remarks
21	Lithium fluoride	(200)*	4.027	LiF	0.351-3.84	Best general crystal for K α - to Lr L-lines. Highest intensity for largest number of elements of any crystal. Combines high intensity and high dispersion.
22	Aluminum	(200)	4.048	Al	0.353-3.86	Curved, especially doubly curved, optics.
23	α -Quartz, silicon dioxide	(20 $\bar{2}$ 0)	4.246	SiO ₂	0.370-4.11	"Prism" cut.
24	α -Quartz, silicon dioxide	(10 $\bar{1}$ 2)	4.564	SiO ₂	0.398-4.35	Used in prototype Laue multichannel spectrometer.
25	Topaz	(200)	4.638	Al ₂ (F,OH) ₂ SiO ₄	0.405-4.43	
26	Aluminum	(111)	4.676	Al	0.408-4.46	Curved, especially doubly curved, optics.
27	α -Quartz, silicon dioxide	(11 $\bar{2}$ 0)	4.912	SiO ₂	0.428-4.75	
28	Gypsum, calcium sulfate dihydrate	(002)	4.990	CaSO ₄ ·2H ₂ O	0.435-4.76	Efflorescent: loses water in vacuum to become Plaster of Paris.
29	Rock salt, sodium chloride	(200)	5.641	NaCl	0.492-5.38	S K α and Cl K α in light matrixes. Like LiF (200), good general crystal for S K to Lr L.

30	Calcite, calcium carbonate	(200)	6.071	CaCO ₃	0.529-5.79	Very precise wavelength measurements. Extremely high degree of crystal perfection with resultant sharp lines.
31	Ammonium dihydrogen phosphate (ADP)	(112)	6.14	NH ₄ H ₂ PO ₄	0.535-5.86	
32	Silicon	(111)*	6.2712	Si	0.547-5.98	Very rugged and stable general-purpose crystal. High degree of perfection obtainable.
33	Sylvite, potassium chloride	(200)	6.292	KCl	0.549-6.00	
34	Fluorite, calcium fluoride	(111)	6.306	CaF ₂	0.550-6.02	Very weak second order, strong third order.
35	Germanium	(111)*	6.532	Ge	0.570-6.23	Eliminates second order. Useful for intermediate- and low-Z elements where Ge K α emission is eliminated by pulse-height selection.
36	Potassium bromide	(200)	6.584	KBr	0.574-6.28	
37	α -Quartz, silicon dioxide	(10 $\bar{1}$ 0)	6.687	SiO ₂	0.583-6.38	P K α in low-Z matrixes, especially in calcium. Intensity for P-K K-lines greater than EDDT, but less than PET.
38	Graphite	(002)	6.708	C	0.585-6.40	P, S, Cl K-lines, P K α intensity > 5X, EDDT. Relatively poor resolution but high integrated reflectivity.
39	Indium antimonide	(111)	7.4806	InSb	0.652-7.23	Important for K-edge of Si.

Table 4-1. Selected data for crystals (continued).

No.	Crystal	Miller indices	$2d$ (Å)	Chemical formula	Useful wavelength region (Å)	Applications, remarks
40	Ammonium dihydrogen phosphate (ADP)	(200)	7.5	$\text{NH}_4\text{H}_2\text{PO}_4$	0.654–7.16	Higher intensity than EDDT.
41	Topaz	(002)	8.374	$\text{Al}_2(\text{F},\text{OH})_2\text{SiO}_4$	0.730–7.99	
42	α -Quartz, silicon dioxide	(10 $\bar{1}$ 0)*	8.512	SiO_2	0.742–8.12	Same applications as EDDT and PET; higher resolution, but lower intensity.
43	Pentaerythritol (PET)	(002)	8.742	$\text{C}(\text{CH}_2\text{OH})_4$	0.762–8.34	Al, Si, P, S, Cl $K\alpha$. Intensities ~ 1.5 -2X EDDT, ~ 2.5 X KHP. Good general crystal for Al-Sc $K\alpha$. Soft; deteriorates with age and exposure to x-rays.
44	Ammonium tartrate	(?)	8.80	$(\text{CHOH})_2(\text{COONH}_4)_2$	0.767–8.4	
45	Ethylenediamine- <i>d</i> -tartrate (EDDT, EDdT, EDT)	(020)	8.808	$\text{NH}_2 - \text{CH}_2 - \text{CH}_2 - \text{NH}_2$ COOH - (CHOH) $_2$ - COOH	0.768–8.40	Same applications as PET, but lower intensity, substantially lower thermal expansion coefficient. Rugged and stable.
46	Ammonium dihydrogen phosphate (ADP)	(101)*	10.640	$\text{NH}_4\text{H}_2\text{PO}_4$	0.928–10.15	Mg $K\alpha$. Same applications as PET, EDDT, but lower intensity.
47	Na β -alumina	(0004)	11.24	$\text{NaAl}_{11}\text{O}_{17}$	0.980–10.87	
48	Oxalic acid dihydrate	(001)	11.92	$(\text{COOH})_2 \cdot 2\text{H}_2\text{O}$	1.04–11.37	

49	Sorbitol hexaacetate (SHA)	(110)	13.98	CHOH - CO - CH ₃ (COH - CO - CH ₃) ₄ CHOH - CO - CH ₃	1.22-13.34	Applications similar to ADP (101) and gypsum (020). High resolution; stable in vacuum. Available in small pieces only.
50	Rock sugar, sucrose	(001)	15.12	C ₁₂ H ₂₂ O ₁₁	1.32-14.42	
51	Gypsum, calcium sulfate dihydrate	(020)*	15.185	CaSO ₄ ·2H ₂ O	1.32-14.49	Na K α . Inferior to KHP, RHP, and beryl. Poor in vacuum (efflorescent).
52	Beryl	(10 $\bar{1}$ 0)	15.954	3BeO·Al ₂ O ₃ ·6SiO ₂	1.39-15.22	Difficult to obtain. Good specimens have $\lambda/\delta\lambda \sim 2500-3000$ at 12 Å. 2d may vary among specimens.
53	Bismuth titanate	(040)	16.40	Bi ₂ (TiO ₃) ₃	1.43-15.65	
54	Mica, muscovite	(002)*	19.84	K ₂ O·3Al ₂ O ₃ ·6SiO ₂ ·2H ₂ O	1.73-18.93	Easy to obtain. Easily bent; good for curved-crystal spectrometers, spectrographs.
55	Silver acetate	(001)	20.0	CH ₃ COOAg	1.74-19.08	
56	Rock sugar, sucrose	(100)	20.12	C ₁₁ H ₂₂ O ₁₁	1.75-19.19	
57	Na β -alumina	(0002)	22.49	NaAl ₁₁ O ₁₇	1.96-21.74	
58	Thallium hydrogen phthalate (THP, TIHP, TAP, TIAP)	(100)	25.9	THC ₈ H ₄ O ₄	2.26-24.7	Same applications as KHP, RHP.

Table 4-1. Selected data for crystals (continued).

No.	Crystal	Miller indices	$2d$ (Å)	Chemical formula	Useful wavelength region (Å)	Applications, remarks
59	Rubidium hydrogen phthalate (RHP, RbHP, RAP, RbAP)	(100)	26.121	$\text{RbHC}_8\text{H}_4\text{O}_4$	2.28-24.92	Diffraction intensity $\sim 3X$ KHP for Na, Mg, Al $K\alpha$ and Cu $L\alpha$; $\sim 4X$ KHP for F $K\alpha$; $\sim 8X$ KHP for O $K\alpha$
60	Potassium hydrogen phthalate (KHP, KAP)	(100)	26.632	$\text{KHC}_8\text{H}_4\text{O}_4$	2.32-25.41	Good general crystal for all low-Z elements down to O.
61	Octadecyl hydrogen maleate (OHM)	(?)	63.5	$\text{CH}_3(\text{CH}_2)_{17}\text{OOC}(\text{CH})_2\text{COOH}$	5.54-60.6	Ultralong-wavelength region down to C $K\alpha$

4.2 SPECULAR REFLECTIVITIES FOR GRAZING-INCIDENCE MIRRORS

Eric M. Gullikson

The specular reflectivity of six common materials is given in Figs. 4-4 and 4-5 for photon energies between 30 eV and 30 keV. The reflectivity for a perfectly smooth surface and for s-polarization is

$$R = |r|^2 = \left| \frac{k_{iz} - k_{tz}}{k_{iz} + k_{tz}} \right|^2, \quad (1)$$

where

$$k_{iz} = \frac{2\pi}{\lambda} \cos \theta \quad \text{and} \quad k_{tz} = \frac{2\pi}{\lambda} \sqrt{n^2 - \cos^2 \theta}.$$

The grazing angle θ is measured from the plane of the mirror surface. The normal components of the incident and transmitted wave vectors are k_{iz} and k_{tz} , respectively. The complex index of refraction n is obtained from the average atomic scattering factor of the material, as described in Section 1.7 and in Ref. 1.

The effect of high-spatial-frequency roughness on the reflection coefficient of an interface can be approximated by the multiplicative factor

$$r = r_0 \exp(-2k_{iz}k_{tz}\sigma^2), \quad (2)$$

where r_0 is the complex reflection coefficient of a perfectly smooth interface and σ is the rms roughness. For updated values of the atomic scattering factors and for on-line reflectivity calculations, see http://www.cxro.lbl.gov/optical_constants/.

REFERENCE

1. B. L. Henke, E. M. Gullikson, and J. C. Davis, "X-Ray Interactions: Photoabsorption, Scattering, Transmission, and Reflection at $E = 50\text{--}30,000$ eV, $Z = 1\text{--}92$," *At. Data Nucl. Data Tables* **54**, 181 (1993).

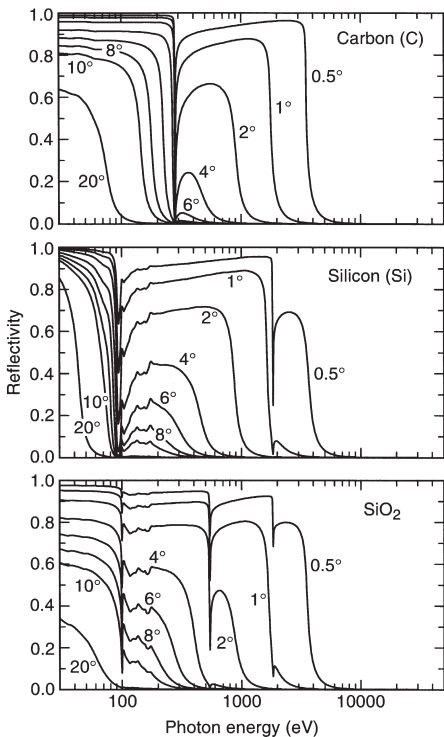


Fig. 4-4. Specular reflectivities of carbon ($\rho = 2.2 \text{ g/cm}^3$), silicon ($\rho = 2.33 \text{ g/cm}^3$), and silicon dioxide ($\rho = 2.2 \text{ g/cm}^3$). The reflectivity is calculated for s-polarization at grazing angles of 0.5, 1, 2, 4, 6, 8, 10, and 20 degrees.

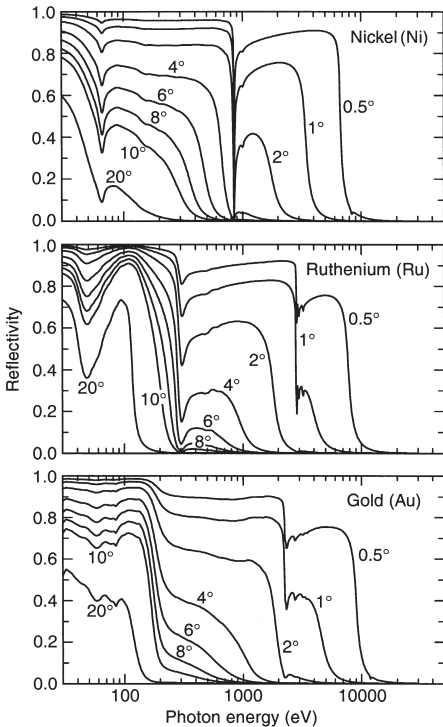


Fig. 4-5. Specular reflectivities of nickel ($\rho = 8.90 \text{ g/cm}^3$), ruthenium ($\rho = 12.41 \text{ g/cm}^3$), and gold ($\rho = 19.3 \text{ g/cm}^3$). The reflectivity is calculated for s-polarization at grazing angles of 0.5, 1, 2, 4, 6, 8, 10, and 20 degrees.

4.3 GRATINGS AND MONOCHROMATORS

Malcolm R. Howells

A more complete version of this document is available at <http://www.cxro.lbl.gov/>. The web version includes additional references to original sources, and values for coefficients given here to fourth order are given there to sixth order.

A. DIFFRACTION PROPERTIES

A.1 Notation and sign convention

We adopt the notation of Fig. 4-6, in which α and β have opposite signs if they are on opposite sides of the normal.

A.2 Grating equation

The grating equation may be written

$$m\lambda = d_0(\sin\alpha + \sin\beta) \quad . \quad (1)$$

The angles α and β are both arbitrary, so it is possible to impose various conditions relating them. If this is done, then for each λ , there will be a unique α and β . The following conditions are used:

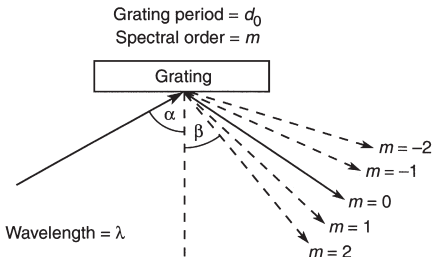


Fig. 4-6. Grating equation notation.

(i) *On-blaze condition:*

$$\alpha + \beta = 2\theta_B \quad , \quad (2)$$

where θ_B is the blaze angle (the angle of the sawtooth). The grating equation is then

$$m\lambda = 2d_0 \sin \theta_B \cos(\beta + \theta_B) \quad . \quad (3)$$

(ii) *Fixed in and out directions:*

$$\alpha - \beta = 2\theta \quad , \quad (4)$$

where 2θ is the (constant) included angle. The grating equation is then

$$m\lambda = 2d_0 \cos \theta \sin(\theta + \beta) \quad . \quad (5)$$

In this case, the wavelength scan ends when α or β reaches 90° , which occurs at the horizon wavelength $\lambda_H = 2d_0 \cos^2 \theta$.

(iii) *Constant incidence angle:* Equation (1) gives β directly.

(iv) *Constant focal distance (of a plane grating):*

$$\frac{\cos \beta}{\cos \alpha} = \text{a constant } c_{\text{ff}} \quad , \quad (6)$$

leading to a grating equation

$$1 - \left(\frac{m\lambda}{d} - \sin \beta \right)^2 = \frac{\cos^2 \beta}{c_{\text{ff}}^2} \quad . \quad (7)$$

Equations (3), (5), and (7) give β (and thence α) for any λ . Examples of the above α - β relationships are (for references see <http://www.cxro.lbl.gov/>):

- (i) Kunz et al. plane-grating monochromator (PGM), Hunter et al. double PGM, collimated-light SX700 PGM
- (ii) Toroidal-grating monochromators (TGMs), spherical-grating monochromators (SGMs, "Dragon" system), Seya-Namioka, most aberration-reduced holographic SGMs, variable-angle SGM, PGMs
- (iii) Spectrographs, "Grasshopper" monochromator
- (iv) Standard SX700 PGM and most variants

B. FOCUSING PROPERTIES

The study of diffraction gratings (for references see <http://www.cxro.lbl.gov/>) goes back more than a century and has included plane, spherical [1], toroidal, and ellipsoidal surfaces and groove patterns made by classical ("Rowland") ruling [2], holography [3,4], and variably spaced ruling [5,6]. In recent years the optical design possibilities of holographic groove patterns and variably spaced rulings have been extensively developed. Following normal practice, we provide an analysis of the imaging properties of gratings by means of the path function F [7]. For this purpose we use the notation of Fig. 4-7, in which the zeroth groove (of width d_0) passes through the grating pole O , while the n th groove passes through the variable point $P(\xi, w, l)$. The holographic groove pattern is taken to be made using two coherent point sources C and D with cylindrical polar coordinates (r_C, γ, z_C) , (r_D, δ, z_D) relative to O . The lower (upper) sign in Eq. (9) refers to C and D both real or both virtual (one real and one virtual), for which case the equiphase surfaces are confocal hyperboloids (ellipses) of revolution about CD . Gratings with varied line spacing $d(w)$ are assumed to be ruled according to $d(w)$

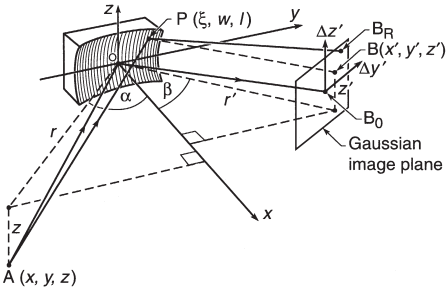


Fig. 4-7. Focusing properties notation.

$= d_0(1 + v_1 w + v_2 w^2 + \dots)$. We consider all the gratings to be ruled on the general surface

$$x = \sum_{ij} a_{ij} w^i l^j$$

and the a_{ij} coefficients are given below.

Ellipse coefficients a_{ij}

$$\begin{aligned} a_{20} &= \frac{\cos \theta}{4} \left(\frac{1}{r} + \frac{1}{r'} \right) & a_{12} &= \frac{a_{20} A}{\cos^2 \theta} \\ a_{30} &= a_{20} A & a_{22} &= \frac{a_{20} (2A^2 + C)}{2 \cos^2 \theta} \\ a_{40} &= \frac{a_{20} (4A^2 + C)}{4} & a_{04} &= \frac{a_{20} C}{8 \cos^2 \theta} \\ a_{02} &= \frac{a_{20}}{\cos^2 \theta} \end{aligned}$$

The other a_{ij} 's with $i + j \leq 4$ are zero. In the expressions above, r , r' , and θ are the object distance, image distance, and incidence angle to the normal, respectively, and

$$A = \frac{\sin \theta}{2} \left(\frac{1}{r} - \frac{1}{r'} \right), \quad C = A^2 + \frac{1}{rr'}$$

Toroid coefficients a_{ij}

$$\begin{aligned} a_{20} &= \frac{1}{2R} & a_{22} &= \frac{1}{4\rho R^2} \\ a_{40} &= \frac{1}{8R^3} & a_{04} &= \frac{1}{8\rho^3} \\ a_{02} &= \frac{1}{2\rho} \end{aligned}$$

Other a_{ij} 's with $i + j \leq 4$ are zero. Here, R and ρ are the major and minor radii of the bicycle-tire toroid.

The a_{ij} 's for spheres; circular, parabolic, and hyperbolic cylinders; paraboloids; and hyperboloids can also be obtained

from the values above by suitable choices of the input parameters r , r' , and θ .

Values for the ellipse and toroid coefficients are given to sixth order at <http://www.cxro.lbl.gov/>.

B.1 Calculation of the path function F

F is expressed as

$$F = \sum_{ijk} F_{ijk} w^{iLj} \quad , \quad (8)$$

where

$$F_{ijk} = z^k C_{ijk}(\alpha, r) + z'^k C_{ijk}(\beta, r') + \frac{m\lambda}{d_0} f_{ijk}$$

and the f_{ijk} term, originating from the groove pattern, is given by one of the following expressions:

$$f_{ijk} = \begin{cases} 1 \text{ when } ijk = 100, 0 \text{ otherwise} & \text{Rowland} \\ \frac{d_0}{\lambda_0} [z_C^k C_{ijk}(\gamma, r_C) \pm z_D^k C_{ijk}(\delta, r_D)] & \text{holographic} \\ n_{ijk} & \text{varied line spacing} \end{cases} \quad (9)$$

The coefficient F_{ijk} is related to the strength of the ij aberration of the wavefront diffracted by the grating. The coefficients C_{ijk} and n_{ijk} are given below, where the following notation is used:

$$T = T(r, \alpha) = \frac{\cos^2 \alpha}{r} - 2a_{20} \cos \alpha \quad (10a)$$

and

$$S = S(r, \alpha) = \frac{1}{r} - 2a_{02} \cos \alpha \quad . \quad (10b)$$

Coefficients C_{ijk} of the expansion of F

$$C_{011} = -\frac{1}{r} \quad C_{020} = \frac{S}{2} \quad C_{022} = -\frac{S}{4r^2} - \frac{1}{2r^3}$$

$$C_{031} = \frac{S}{2r^2} \quad C_{040} = \frac{4a_{02}^2 - S^2}{8r} - a_{04} \cos \alpha$$

$$C_{100} = -\sin \alpha \quad C_{102} = \frac{\sin \alpha}{2r^2}$$

$$C_{111} = -\frac{\sin \alpha}{r^2} \quad C_{120} = \frac{S \sin \alpha}{2r} - a_{12} \cos \alpha$$

$$C_{200} = \frac{T}{2} \quad C_{202} = -\frac{T}{4r^2} + \frac{\sin^2 \alpha}{2r^3}$$

$$C_{211} = \frac{T}{2r^2} - \frac{\sin^2 \alpha}{r^3} \quad C_{300} = -a_{30} \cos \alpha + \frac{T \sin \alpha}{2r}$$

$$C_{220} = -a_{22} \cos \alpha + \frac{1}{4r} (4a_{20}a_{02} - TS - 2a_{12} \sin 2\alpha) + \frac{S \sin^2 \alpha}{2r^2}$$

$$C_{400} = -a_{40} \cos \alpha + \frac{1}{8r} (4a_{20}^2 - T^2 - 4a_{30} \sin 2\alpha) + \frac{T \sin^2 \alpha}{2r^2}$$

The coefficients for which $i \leq 4, j \leq 4, k \leq 2, i + j + k \leq 4, j + k = \text{even}$ are included here.

Coefficients n_{ijk} of the expansion of F

$$n_{ijk} = 0 \quad \text{for } j, k \neq 0$$

$$n_{100} = 1 \quad n_{300} = \frac{v_1^2 - v_2}{3}$$

$$n_{200} = \frac{-v_1}{2} \quad n_{400} = \frac{-v_1^3 + 2v_1v_2 - v_3}{4}$$

Values for C_{ijk} and n_{ijk} are given to sixth order at <http://www.cxro.lbl.gov/>.

B.2 Determination of the Gaussian image point

By definition the principal ray AOB_0 arrives at the Gaussian image point $B_0(r'_0, \beta_0, z'_0)$ in Fig. 4-7. Its direction is given by Fermat's principal, which implies $[\partial F/\partial w]_{w=0, l=0} = 0$ and $[\partial F/\partial l]_{w=0, l=0} = 0$, from which

$$\frac{m\lambda}{d_0} = \sin \alpha + \sin \beta_0 \quad (11a)$$

and

$$\frac{z}{r} + \frac{z'_0}{r'_0} = 0 \quad , \quad (11b)$$

which are the grating equation and the law of magnification in the vertical direction. The tangential focal distance r'_0 is obtained by setting the focusing term F_{200} equal to zero and is given by

$$T(r, \alpha) + T(r'_0, \beta_0) = \begin{cases} 0 & \text{Rowland} \\ -\frac{m\lambda}{\lambda_0} [T(r_C, \gamma) \pm T(r_D, \delta)] & \text{holographic} \\ \frac{v_1 m\lambda}{d_0} & \text{varied line spacing} \end{cases} \quad (12)$$

Equations (11) and (12) determine the Gaussian image point B_0 and, in combination with the sagittal focusing condition ($F_{020} = 0$), describe the focusing properties of grating systems under the paraxial approximation. For a Rowland spherical grating the focusing condition, Eq. (12), is

$$\left(\frac{\cos^2 \alpha}{r} - \frac{\cos \alpha}{R} \right) + \left(\frac{\cos^2 \beta}{r'_0} - \frac{\cos \beta}{R} \right) = 0 \quad , \quad (13)$$

which has important special cases: (i) plane grating, $R = \infty$, implying

$$r'_0 = -r \cos^2 \alpha / \cos^2 \beta = -r / c_{ff}^2$$

so that the focal distance and magnification are fixed if c_{ff} is held constant; (ii) object and image on the Rowland circle, i.e., $r = R \cos \alpha$, $r'_0 = R \cos \beta$, $M = 1$; and (iii) $\beta = 90^\circ$ (Wadsworth condition). The focal distances of TGMs and SGMs, with or without moving slits, are also determined using Eq. (13).

B.3 Calculation of ray aberrations

In an aberrated system, the outgoing ray will arrive at the Gaussian image plane at a point B_R displaced from the Gaussian image point B_0 by the ray aberrations $\Delta y'$ and $\Delta z'$ (Fig. 4-7). The latter are given by

$$\Delta y' = \frac{r'_0}{\cos \beta_0} \frac{\partial F}{\partial w}, \quad \Delta z' = r'_0 \frac{\partial F}{\partial l}, \quad (14)$$

where F is to be evaluated for $A = (r, \alpha, z)$ and $B = (r'_0, \beta_0, z'_0)$. By means of the expansion of F , these equations allow the ray aberrations to be calculated separately for each aberration type:

$$\Delta y'_{ijk} = \frac{r'_0}{\cos \beta_0} F_{ijk} i w^{i-1} l^j, \quad \Delta z'_{ijk} = r'_0 F_{ijk} w^i j l^{j-1}. \quad (15)$$

Moreover, provided the aberrations are not too large, they are additive, so that they may either reinforce or cancel.

C. DISPERSION PROPERTIES

Dispersion properties can be summarized by the following relations.

(i) *Angular dispersion:*

$$\left(\frac{\partial \lambda}{\partial \beta} \right)_\alpha = \frac{d \cos \beta}{m}. \quad (16)$$

(ii) *Reciprocal linear dispersion:*

$$\left(\frac{\partial \lambda}{\partial (\Delta y')} \right)_\alpha = \frac{d \cos \beta}{m r'} = \frac{10^{-3} d [\text{\AA}] \cos \beta}{m r' [\text{m}]} \text{\AA}/\text{mm}. \quad (17)$$

(iii) *Magnification:*

$$M(\lambda) = \frac{\cos \alpha}{\cos \beta} \frac{r'}{r} \quad . \quad (18)$$

(iv) *Phase-space acceptance (ϵ):*

$$\epsilon = N\Delta\lambda_{S1} = N\Delta\lambda_{S2} \quad (\text{assuming } S_2 = MS_1) \quad , \quad (19)$$

where N is the number of participating grooves.

D. RESOLUTION PROPERTIES

The following are the main contributions to the width of the instrumental line spread function. An estimate of the total width is the vector sum.

(i) *Entrance slit (width S_1):*

$$\Delta\lambda_{S1} = \frac{S_1 d \cos \alpha}{mr} \quad . \quad (20)$$

(ii) *Exit slit (width S_2):*

$$\Delta\lambda_{S2} = \frac{S_2 d \cos \beta}{mr'} \quad . \quad (21)$$

(iii) *Aberrations (of perfectly made grating):*

$$\Delta\lambda_A = \frac{\Delta y' d \cos \beta}{mr'} = \frac{d}{m} \left(\frac{\partial F}{\partial w} \right) \quad . \quad (22)$$

(iv) *Slope error $\Delta\phi$ (of imperfectly made grating):*

$$\Delta\lambda_{SE} = \frac{d(\cos \alpha + \cos \beta) \Delta\phi}{m} \quad . \quad (23)$$

Note that, provided the grating is large enough, diffraction at the entrance slit always guarantees a coherent illumination of enough grooves to achieve the slit-width-limited resolution. In such case a diffraction contribution to the width need not be added to the above.

E. EFFICIENCY

The most accurate way to calculate grating efficiencies is by the full electromagnetic theory [8]. However, approximate scalar-theory calculations are often useful and, in particular, provide a way to choose the groove depth h of a laminar grating. According to Bennett, the best value of the groove-width-to-period ratio r is the one for which the usefully illuminated area of the groove bottom is equal to that of the top. The scalar-theory efficiency of a laminar grating with $r = 0.5$ is given by Franks et al. as

$$E_0 = \frac{R}{4} \left[1 + 2(1 - P) \cos \left(\frac{4\pi h \cos \alpha}{\lambda} \right) + (1 - P)^2 \right]$$

$$E_m = \begin{cases} \left[\frac{R}{m^2 \pi^2} [1 - 2 \cos Q^+ \cos(Q^- + \delta) + \cos^2 Q^+] \right] & m = \text{odd} \\ \left[\frac{R}{m^2 \pi^2} \cos^2 Q^+ \right] & m = \text{even} \end{cases} \quad (24)$$

where

$$P = \frac{4h \tan \alpha}{d_0} \quad ,$$

$$Q^\pm = \frac{m\pi h}{d_0} (\tan \alpha \pm \tan \beta) \quad ,$$

$$\delta = \frac{2\pi h}{\lambda} (\cos \alpha + \cos \beta) \quad ,$$

and R is the reflectance at grazing angle $\sqrt{\alpha_G \beta_G}$, where

$$\alpha_G = \frac{\pi}{2} - |\alpha| \quad \text{and} \quad \beta_G = \frac{\pi}{2} - |\beta| \quad .$$

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4.4 ZONE PLATES

Janos Kirz and David Attwood

A zone plate is a circular diffraction grating. In its simplest form, a transmission Fresnel zone plate lens consists of alternate transparent and opaque rings. The radii of the zone plate edges are given by

$$r_n^2 = nf\lambda + n^2\lambda^2 / 4 \quad , \quad (1)$$

where n is the zone number (opaque and transparent zones count separately), λ is the wavelength, and f is the first-order focal length. The zone plate lens can be used to focus monochromatic, uniform plane wave (or spherical wave) radiation to a small spot, as illustrated in Fig. 4.8, or can be used for near-axis point-by-point construction of a full-field image, as illustrated in Fig. 4.9. When used in imaging applications, it obeys the thin-lens formula

$$\frac{1}{p} + \frac{1}{q} = \frac{1}{f} \quad , \quad (2)$$

where p and q are the object and image distances, respectively. Note that the second term on the right-hand side of Eq. (1) is correct only for $p \ll q$ or $p \gg q$, which are the common cases in zone plate applications. Descriptions of the diffractive properties of zone plate lenses, their use in x-ray microscopes, and extensive references to the literature are given elsewhere [1-3].

A zone plate lens is fully specified by three parameters. Most applications are dominated by the choice of photon energy $\hbar\omega$, and thus by λ . Resolution is set in large part by λ and the outer zone width, $\Delta r \equiv r_N - r_{N-1}$. To avoid chromatic blurring, the number of zones, N , must be less than the inverse relative spectral bandwidth, $\lambda/\Delta\lambda$. Thus, for many applications, λ , Δr , and N constitute a natural set of zone plate-defining parameters. In terms of these three basic parameters, other zone plate parameters are given by [3]

$$D = 4N\Delta r \quad (3)$$

$$f = 4N(\Delta r)^2 / \lambda \quad (4)$$

$$F^\# = \Delta r / \lambda \quad (5)$$

$$NA = \lambda / 2\Delta r \quad (6)$$

To avoid chromatic blurring requires that

$$N < \lambda / \Delta\lambda \quad (7)$$

where $\Delta\lambda$ is the spectral width of the illuminating radiation.

For uniform plane-wave illumination, as in Fig. 4-8, the Rayleigh criterion sets the diffraction-limited (λ , NA) spatial resolution of a perfect lens as

$$Res = \frac{0.610\lambda}{NA} = 1.22 \Delta r \quad (8)$$

where $NA = \sin\theta$ is the numerical aperture of the lens in vacuum and θ is the half-angle of the focused radiation. This resolution is obtained over an axial depth of focus given by

$$DOF = \pm \frac{\lambda}{2(NA)^2} \quad (9)$$

Spatial resolution in the full-field case can be improved somewhat from that given in Eq. (8), depending on both the object itself and the partial coherence of the illumination [4], as set by the parameter

$$\sigma = \frac{(NA)_c}{(NA)_o} \quad (10)$$

where $(NA)_c$ refers to the illumination numerical aperture of the condenser, and $(NA)_o$ is that of the zone plate objective lens, as given in Eq. (6).

The efficiency of the simple zone plate in the first order is ideally π^2 , or about 10%. The remainder of the radiation is absorbed (50%) or diffracted in other orders—zero order (25%), negative orders (12.5%), and higher positive orders (2.5%). If opaque zones are replaced by transparent but phase-shifting zones, efficiencies can be substantially improved [5]. To isolate the first order from unwanted orders, zone plate lenses are often made with a central stop, used in conjunction with a somewhat smaller collimating aperture near the focal

region. This is particularly effective in scanning x-ray microscopes [6,7], where the zone plate is used to focus the radiation to a small spot, as in Fig. 4-8. For full-field x-ray microscopes, as suggested by Fig. 4-9, first-order imaging is assisted in a similar manner by stopping the central portion of the illuminating radiation [8].

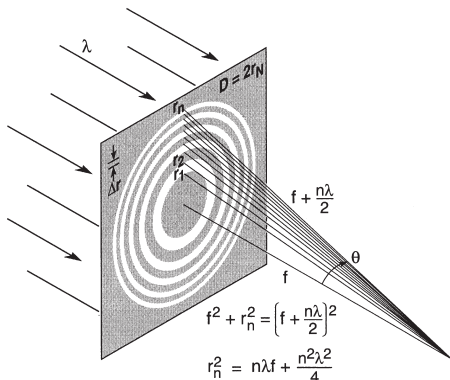


Fig. 4-8. A Fresnel zone plate lens with plane wave illumination, showing only the convergent (+1st) order of diffraction. Sequential zones of radius r_n are specified such that the incremental path length to the focal point is $n\lambda/2$. Alternate zones are opaque in the simple transmission zone plate. With a total number of zones, N , the zone plate lens is fully specified. Lens characteristics such as the focal length f , diameter D , and numerical aperture NA are described in terms of λ , N , and Δr , the outer zone width. [Courtesy of Cambridge University Press, Ref. 3.]

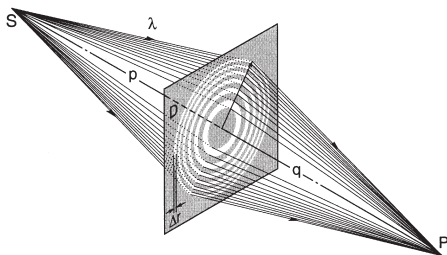


Fig. 4-9. A Fresnel zone plate used as a diffractive lens to form an x-ray image of a source point S in the image plane at P . The lens is shown as having a diameter D and outer zone width Δr . The object and image distances are p and q , respectively. A full-field image is formed concurrently in this manner. [Courtesy of Cambridge University Press, Ref. 3.]

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4.5 X-RAY DETECTORS

Albert C. Thompson

A wide variety of x-ray detectors is available, some counting single photons, some providing only measurements of count rate or total flux, others measuring the energy, position, and/or incidence time of each x-ray. Table 4-2 provides typical values for useful energy range, energy resolution, dead time per event, and maximum count rate capability for common x-ray detectors. For special applications, these specifications can often be substantially improved.

Table 4-2. *Properties of common x-ray detectors; ΔE is measured as FWHM.*

Detector	Energy range (keV)	$\Delta E/E$ at 5.9 keV (%)	Dead time/event (μ s)	Maximum count rate (s^{-1})
Gas ionization (current mode)	0.2–50	n/a	n/a	10^{11a}
Gas proportional	0.2–50	15	0.2	10^6
Multiwire and microstrip proportional	3–50	20	0.2	$10^6/\text{mm}^2$
Scintillation [NaI(Tl)]	3–10,000	40	0.25	2×10^6
Energy-resolving semiconductor	1–10,000	2.1	0.5–30	5×10^5
Surface-barrier (current mode)	0.1–20	n/a	n/a	10^8
Avalanche photodiode	0.1–50	20	0.001	10^8
CCD	0.1–70	n/a	n/a	n/a
Superconducting	0.1–4	< 0.5	100	5×10^3
Image plate	4–80	n/a	n/a	n/a

^a Maximum count rate is limited by space-charge effects to around 10^{11} photons/s per cm^2 .

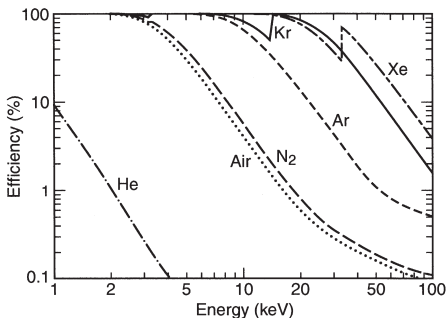


Fig. 4-10. Efficiency of a 10-cm-long gas ionization chamber as a function of energy, for different gases at normal pressure.

A. GAS IONIZATION DETECTORS

Gas ionization detectors are commonly used as integrating detectors to measure beam flux rather than individual photons. A typical detector consists of a rectangular gas cell with thin entrance and exit windows. Inside the detector, an electric field of about 100 V/cm is applied across two parallel plates. Some of the x-rays in the beam interact with the chamber gas to produce fast photoelectrons, Auger electrons, and/or fluorescence photons. The energetic electrons produce additional electron-ion pairs by inelastic collisions, and the photons either escape or are photoelectrically absorbed. The electrons and ions are collected at the plates, and the current is measured with a low-noise current amplifier. The efficiency of the detector can be calculated from the active length of the chamber, the properties of the chamber gas, and the x-ray absorption cross section at the appropriate photon energy. Figure 4-10 shows, for different gases at normal pressure, the efficiency of a 10-cm-long ion chamber as a function of energy. Once the efficiency is known, the photon flux can be estimated from

chamber current and the average energy required to produce an electron-ion pair (Table 4-3)[1].

Table 4-3. Average energy required to produce an electron-ion pair in several gases

Element	Energy (eV)
Helium	41
Nitrogen	36
Air	34.4
Neon	30
Argon	25
Krypton	24
Xenon	22

B. GAS PROPORTIONAL COUNTERS

Gas proportional detectors consist of a small-diameter anode wire in an enclosed gas volume. They are usually used to count single photon events. When a photon interacts in the gas, some gas atoms are ionized, and the electrons are attracted to the positive anode wire. Near the anode wire, the electrons are accelerated by the high electric field, producing a cascade of electrons that result in a large electrical pulse. The output is coupled to a low-noise preamplifier to give usable pulses. The pulse height resolution of the detector (about 20% at 6 keV) can be used for some energy discrimination, and the output counting rate can be as high as 10^6 counts per second.

C. MULTIWIRES AND MICROSTRIP PROPORTIONAL CHAMBERS

Multiwire and microstrip proportional chambers are widely used as position-sensitive detectors of both photons and charged particles. Multiwire chambers use a grid of fine wires spaced about 2 mm apart as the anode plane in a gas proportional chamber. Microstrip detectors use a patterned anode plane. The spatial resolution can be as good as $30 \mu\text{m}$. Gas electron multiplying (GEM) detectors have been

developed that have improved spatial resolution and lower operating voltages [2].

D. SCINTILLATION DETECTORS

Scintillation detectors work by converting x-rays to optical photons in special materials and then detecting the light with a photomultiplier tube or a photodiode. The scintillator materials can be either organic scintillators, single crystals of thallium-activated sodium iodide [commonly referred to as NaI(Tl)], single crystals of sodium-activated cesium iodide [CsI(Na)], or single crystals of bismuth germanate (BGO). Since the light output is low (about 200–300 eV is required for each optical photon), the energy resolution is also low. Organic scintillators have very poor energy resolution, whereas the NaI(Tl), CsI(Na), and BGO crystals have energy resolutions of about 40% at 10 keV. These detectors can have a time resolution of better than 1 ns and a count rate capability up to 2×10^6 photons per second. For a scintillator thickness of more than 5 mm, for both NaI and CsI, the detection efficiency between 20 and 100 keV is essentially unity.

Gas scintillation detectors combine the operation of gas ionization chambers and photon detectors to give improved performance. Electrons generated from photon or charged-particle interactions in a gas (usually pure xenon or argon with 1% xenon) are accelerated in a high-field (~ 3 kV/cm) region, where they produce UV scintillation light. This light is usually wave-shifted and then detected by a photomultiplier. These detectors have an energy resolution about two to three times better than conventional proportional chambers.

E. ENERGY-RESOLVING SEMICONDUCTOR DETECTORS

Silicon and germanium detectors can make excellent energy-resolving detectors of single photons (about 150 eV at 5.9 keV). They are basically large, reverse-biased $n^+ - i - p^+$ diodes. When a photon interacts in the intrinsic region, tracks of electron-hole pairs are produced (analogous to electron-positive ion pairs in a counting gas). In the presence of the electric field, these pairs separate and rapidly drift to the detector contacts. The average energy required to generate an

electron-hole pair is 3.6 eV for silicon and 2.98 eV for germanium. To keep the leakage current low, the detector must have very few electrically active impurities. For example, germanium detectors are made from zone-refined crystals that have fewer than 10^{10} electrically active impurities/cm³. They are usually cooled to reduce the thermal leakage current. Recently, silicon drift detectors have become available with an energy resolution of <140 eV and a count rate up to about 5×10^5 per second. To handle the high counting rates available at synchrotrons, multielement arrays of 4–30 elements have been developed for fluorescent EXAFS experiments.

F. CURRENT-MODE SEMICONDUCTOR DETECTORS

Semiconductor diodes are also used in current mode to measure x-ray flux. They have very linear responses and are available with thin entrance windows. Surface-barrier detectors are good beam monitors when used with low-noise current amplifiers. In addition, silicon avalanche detectors are now available in which the silicon is biased so that there is an internal avalanche of electron-hole pairs for each interacting photon. These devices can be used at lower beam intensities in a pulse-counting mode and in current mode at higher photon fluxes. They have excellent time resolution (≈ 1 ns) but limited energy resolution.

G. CCD DETECTORS

CCD detectors are now used in a variety of ways for x-ray imaging. They are available with up to 4096×4096 pixels, with pixel sizes of $12 \mu\text{m} \times 12 \mu\text{m}$ and readout times of less than 1 s. In most scientific applications, CCD detectors are cooled to below -30°C to reduce background noise. In most systems, a thin phosphor screen converts the incident x-rays into optical photons, which the CCD detects. A commonly used phosphor is $\text{Gd}_2\text{O}_2\text{S}(\text{Tb})$, which has a high efficiency and a light decay time of a few hundred microseconds. When used as a detector for macromolecular crystallography, a large phosphor screen (up to 300 mm^2) is usually coupled to a set of up to 9 CCD's with tapered optical fibers [3]. On the other hand, for high-spatial-resolution x-ray imaging, a very thin (5-

to 20- μm) sapphire scintillation screen is optically coupled with a high-quality microscope lens to give a spatial resolution of less than 1 μm [4]. For imaging with x-rays below 1 keV, direct exposure of back-thinned CCD detectors is used.

H. OTHER X-RAY DETECTORS

X-ray detectors operating at superconducting temperatures (0.1–4 K) have recently been developed; these devices achieve excellent energy resolution (12 eV at 700 eV). They are currently very small and very inefficient for x-rays above 1 keV, and they have maximum count rates of only about $5 \times 10^3 \text{ s}^{-1}$. With further development, however, they may make very useful x-ray spectrometers [5].

Microchannel plate detectors are compact, high-gain electron multipliers, which are often used as efficient electron or low-energy photon detectors. A typical MCP consists of about 10^7 closely packed lead-glass channels of equal diameter. Typically, the diameter of each channel, which acts as an independent, continuous dynode photomultiplier, is $\sim 10 \mu\text{m}$.

Image plate detectors are available that have many of the characteristics of film but with the advantage of excellent dynamic range, efficiency, and large area [6]. They are made with a plate containing a photosensitive material that on exposure to x-rays creates color centers that can be read out in a scanning mode with a laser as a digital image.

For fast timing experiments, silicon Avalanche Detectors are available that have a time resolution of less than 1 ns [7].

For high-speed x-ray imaging experiments, x-ray streak cameras have been developed that have time resolutions of around 350 fs.

Finally, photographic film is also available for quick x-ray–imaging experiments; however, because of the need for processing after exposure, it is no longer commonly used for scientific measurements. Special films are available that give improved efficiency, contrast, or resolution.

I. CALIBRATION OF X-RAY BEAM MONITORS

Measurement of the relative intensity of x-ray beams is usually done with a gas ionization chamber or a thin silicon diode in

the beam path. Another technique is to place a thin foil (usually plastic) in the beam and to measure the scattered photons with a scintillation detector. The approximate efficiency of a gas ionization detector can be estimated from its active length and the properties of the chamber gas at the energy of the x-ray beam. However, calibration of these detectors to measure absolute x-ray intensity is more difficult. One calibration technique is to use a well-characterized single-photon detector as a standard and to establish the x-ray flux-to-detector current calibration of the beam monitor at a reduced beam flux where the single-photon counter response is linear. A thick silicon pin diode can also be used for flux measurement [8].

At photon energies below 1000 eV, silicon photodiodes are available that can be used as absolute beam monitors [9].

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SECTION 5

MISCELLANEOUS

5.1 PHYSICAL CONSTANTS

Table 5-1 was drawn from the recommendations of CODATA (the Committee on Data for Science and Technology). The full 1998 CODATA set of constants may be found at <http://physics.nist.gov/cuu/Constants/index.html>.

Table 5-1. Physical constants.

Quantity	Symbol, equation	Value	Uncert. (ppb)
speed of light	c (see note *)	$299\,792\,458\text{ m s}^{-1}$ (10^{10} cm s^{-1})	exact
Planck constant	h	$6.626\,068\,96(33)\times 10^{-34}\text{ J s}$ (10^{-27} erg s)	50
Planck constant, reduced	$\hbar = h/2\pi$	$1.054\,571\,628(53)\times 10^{-34}\text{ J s} = 6.582\,118\,99(16)\times 10^{-22}\text{ MeV s}$	50, 25
electron charge magnitude	e	$4.803\,204\,27(12)\times 10^{-10}\text{ esu} = 1.602\,176\,87(40)\times 10^{-19}\text{ C}$	25, 25
conversion constant	hc	$197.326\,9631(49)\text{ MeV fm}$ ($=\text{ eV nm}$)	25
electron mass	m_e	$0.510\,998\,910(13)\text{ MeV}/c^2 = 9.109\,382\,15(45)\times 10^{-31}\text{ kg}$	25, 50
proton mass	m_p	$938.272\,013(23)\text{ MeV}/c^2 = 1.672\,621\,637(83)\times 10^{-27}\text{ kg}$ $= 1.007\,276\,466\,77(10)\text{ u} = 1836.152\,672\,47(80)\text{ }m_e$	25, 50 0.10, 0.43
deuteron mass	m_d	$1875.612\,793\,(47)\text{ MeV}/c^2$	25
unified atomic mass unit (u)	(mass $^{12}\text{C atom}$)/12 = (1 g)/(N_A mol)	$931.494\,028(23)\text{ MeV}/c^2 = 1.660\,538\,782(83)\times 10^{-27}\text{ kg}$	25, 50
permeability of free space	$\epsilon_0 = 1/(\mu_0 c^2)$	$8.854\,187\,817\dots\times 10^{-12}\text{ F m}^{-1}$	exact
permeability of free space	μ_0	$4\pi\times 10^{-7}\text{ N A}^{-2} = 12.566\,370\,614\dots\times 10^{-7}\text{ N A}^{-2}$	exact
fine-structure constant	$\alpha = e^2/4\pi\epsilon_0\hbar c$	$1/137.035\,999\,679(94)$	0.68
classical electron radius	$r_e = e^2/4\pi\epsilon_0 m_e c^2$	$2.817\,940\,2894(58)\times 10^{-15}\text{ m}$	2.1
Bohr radius ($m_{\text{nucleus}} = \infty$)	$a_0 = 4\pi\epsilon_0\hbar^2/m_e e^2 = r_e\alpha^{-2}$	$0.529\,177\,208\,59(36)\times 10^{-10}\text{ m}$ (10^{-8} cm)	0.68
Rydberg energy	$hcR_\infty = m_e e^4/2(4\pi\epsilon_0)^2\hbar^2$ $= m_e c^2\alpha^2/2$	$13.605\,691\,93(34)\text{ eV}$	25

Thomson cross section	$\sigma_T = 8\pi r_e^2/3$	0.665 245 8558(27) barn (10^{-28} m ²)	4.1
Bohr magneton	$\mu_B = e\hbar/2m_e$	5.788 381 7555(79) $\times 10^{-11}$ MeV T ⁻¹	1.4
nuclear magneton	$\mu_N = e\hbar/2mp$	3.152 451 2326(45) $\times 10^{-14}$ MeV T ⁻¹	1.4
electron cyclotron freq./field	$\omega_{\text{cycl}}^e/B = e/m_e$	1.758 820 150(44) $\times 10^{11}$ rad s ⁻¹ T ⁻¹	25
proton cyclotron freq./field	$\omega_{\text{cycl}}^p/B = e/mp$	9.578 833 92(24) $\times 10^7$ rad s ⁻¹ T ⁻¹	25
Avogadro constant	N_A	6.022 141 79(30) $\times 10^{23}$ mol ⁻¹	50
Boltzman constant	k	1.380 650 4(24) $\times 10^{-23}$ J K ⁻¹ = 8.617 343(15) $\times 10^{-5}$ eV K ⁻¹	1700
molar volume, ideal gas at STP	$N_A k$ (273.15 K)/(101 325 Pa)	22.413 996(39) $\times 10^{-3}$ m ³ mol ⁻¹	1700
$\pi = 3.141 592 653 589 793 238$	$e = 2.718 281 828 459 045 235$	$\gamma = 0.577 215 664 901 532 861$	
*The meter is the length of the path traveled by light in vacuum during a time interval of 1/299 792 458 of a second.			
1 in. = 2.54 cm	1 newton = 10 ⁵ dyne	1 eV/c ² = 1.782 662 $\times 10^{-33}$ g	1 coulomb = 2.997 924 58 $\times 10^9$ esu
1 Å = 10 ⁻⁸ cm	1 joule = 10 ⁷ erg	hc (1 eV) = 1.239 842 μm	1 tesla = 10 ⁴ gauss
1 fm = 10 ⁻¹³ cm	1 cal = 4.184 joule	1 eV/h = 2.417 989 $\times 10^{14}$ Hz	1 atm = 1.013 25 $\times 10^6$ dyne/cm ²
1 barn = 10 ⁻²⁴ cm ²	1 eV = 1.602 176 5 $\times 10^{-12}$ erg	1 eV/k = 11 604.5 K	0°C = 273.15 K

The NIST Reference on Constants, Units, and Uncertainty: <http://physics.nist.gov/cuu/Constants/index.html>

5.2 PHYSICAL PROPERTIES OF THE ELEMENTS

Table 5-2 lists several important properties of the elements. Data were taken mostly from D. R. Lide, Ed., *CRC Handbook of Chemistry and Physics*, 80th ed. (CRC Press, Boca Raton, Florida, 1999). Atomic weights apply to elements as they exist naturally on earth; values in parentheses are the mass numbers for the longest-lived isotopes. Some uncertainty exists in the last digit of each atomic weight. Specific heats are given for the elements at 25°C and a pressure of 100 kPa. Densities for solids and liquids are given as specific gravities at 20°C unless otherwise indicated by a superscript temperature (in °C); densities for the gaseous elements are given in g/cm³ for the liquids at their boiling points. The ionization energies were taken from <http://physics.nist.gov/PhysRefData/IonEnergy/ionEnergy.html>.

A periodic table of the elements follows Table 5-2, on page 5-10. A more detailed periodic table can be found at <http://www.cxro.lbl.gov/>.

Table 5-2. Properties of the elements.

Z	Element	Atomic weight	Density	Melting point (°C)	Boiling point (°C)	Ground-state configuration	Ground level	Ionization energy (eV)	Specific heat (J/g·K)
1	Hydrogen	1.00794	0.0708	-259.34	-252.87	1s	2S _{1/2}	13.598	14.304
2	Helium	4.002602	0.122	—	-268.93	1s ²	1S ₀	24.587	5.193
3	Lithium	6.941	0.534	180.50	1342	1s ² 2s	2S _{1/2}	5.392	3.582
4	Beryllium	9.012182	1.848	1287	2471	1s ² 2s ²	1S ₀	9.323	1.825
5	Boron	10.811	2.34	2075	4000	1s ² 2s ² 2p	2P ^o _{1/2}	8.298	1.026
6	Carbon	12.0107	1.9-2.3 (graph)	4492, 10.3 MPa	3825 ^b	1s ² 2s ² 2p ²	3P ₀	11.260	0.709
7	Nitrogen	14.00674	0.808	-210.00	-195.79	1s ² 2s ² 2p ³	4S ^o _{3/2}	14.534	1.040
8	Oxygen	15.9994	1.14	-218.79	-182.95	1s ² 2s ² 2p ⁴	3P ₂	13.618	0.918
9	Fluorine	18.9984032	1.50	-219.62	-188.12	1s ² 2s ² 2p ⁵	2P ^o _{3/2}	17.423	0.824
10	Neon	20.1797	1.207	-248.59	-246.08	1s ² 2s ² 2p ⁶	1S ₀	21.565	1.030
11	Sodium	22.989770	0.971	97.80	883	[Ne] 3s	2S _{1/2}	5.139	1.228
12	Magnesium	24.3050	1.738	650	1090	[Ne] 3s ²	1S ₀	7.646	1.023
13	Aluminum	26.981538	2.6989	660.32	2519	[Ne] 3s ² 3p	2P ^o _{1/2}	5.986	0.897
14	Silicon	28.0855	2.3325	1414	3265	[Ne] 3s ² 3p ²	3P ₀	8.152	0.705
15	Phosphorus	30.973761	1.82	44.15	280.5	[Ne] 3s ² 3p ³	4S ^o _{3/2}	10.487	0.769
16	Sulfur	32.066	2.07	119.6	444.60	[Ne] 3s ² 3p ⁴	3P ₂	10.360	0.710
17	Chlorine	35.4527	1.56-33.6	-101.5	-34.04	[Ne] 3s ² 3p ⁵	2P ^o _{3/2}	12.968	0.479
18	Argon	39.948	1.40	-189.35	-185.85	[Ne] 3s ² 3p ⁶	1S ₀	15.760	0.520
19	Potassium	39.0983	0.862	63.5	759	[Ar] 4s	2S _{1/2}	4.341	0.757
20	Calcium	40.078	1.55	842	1484	[Ar] 4s ²	1S ₀	6.113	0.647
21	Scandium	44.955910	2.98925	1541	2836	[Ar] 3d 4s ²	2D _{3/2}	6.562	0.568
22	Titanium	47.867	4.54	1668	3287	[Ar] 3d ² 4s ²	3F ₂	6.828	0.523
23	Vanadium	50.9415	6.118.7	1910	3407	[Ar] 3d ³ 4s ²	4F _{3/2}	6.746	0.489
24	Chromium	51.9961	7.18-7.20	1907	2671	[Ar] 3d ⁵ 4s	7S ₃	6.766	0.449
25	Manganese	54.938049	7.21-7.44	1246	2061	[Ar] 3d ⁵ 4s ²	6S _{5/2}	7.434	0.479
26	Iron	55.845	7.874	1538	2861	[Ar] 3d ⁶ 4s ²	5D ₄	7.902	0.449

Table 5-2. Properties of the elements (continued).

Z	Element	Atomic weight	Density	Melting point (°C)	Boiling point (°C)	Ground-state configuration	Ground level	Ionization energy (eV)	Specific heat (J/g·K)
27	Cobalt	58.933200	8.9	1495	2927	[Ar] 3d ⁷ 4s ²	4F _{9/2}	7.881	0.421
28	Nickel	58.6934	8.902 ²⁵	1455	2913	[Ar] 3d ⁸ 4s ²	3F ₄	7.640	0.444
29	Copper	63.546	8.96	1084.62	2562	[Ar] 3d ¹⁰ 4s	2S _{1/2}	7.726	0.385
30	Zinc	65.39	7.133 ²⁵	419.53	907	[Ar] 3d ¹⁰ 4s ²	1S ₀	9.394	0.388
31	Gallium	69.723	5.904 ^{29,6}	29.76	2204	[Ar] 3d ¹⁰ 4s ² 4p	2P _{1/2}	5.999	0.371
32	Germanium	72.61	5.323 ²⁵	938.25	2833	[Ar] 3d ¹⁰ 4s ² 4p ²	3P ₀	7.899	0.320
33	Arsenic	74.92160	5.73	817 ^{3,7} MPa	603 ^b	[Ar] 3d ¹⁰ 4s ² 4p ³	4S _{3/2}	9.789	0.329
34	Selenium	78.96	4.79	220.5	685	[Ar] 3d ¹⁰ 4s ² 4p ⁴	3P ₂	9.752	0.321
35	Bromine	79.904	3.12	-7.2	58.8	[Ar] 3d ¹⁰ 4s ² 4p ⁵	2P _{3/2}	11.814	0.226
36	Krypton	83.80	2.16	157.387 ^{3,2} kPa	-153.22	[Ar] 3d ¹⁰ 4s ² 4p ⁶	1S ₀	14.000	0.248
37	Rubidium	85.4678	1.532	39.30	688	[Kr] 5s	2S _{1/2}	4.177	0.363
38	Strontium	87.62	2.54	777	1382	[Kr] 5s ²	1S ₀	5.695	0.301
39	Yttrium	88.90585	4.469 ²⁵	1522	3345	[Kr] 4d 5s ²	2D _{3/2}	6.217	0.298
40	Zirconium	91.224	6.506	1855	4409	[Kr] 4d ² 5s ²	3F ₂	6.634	0.278
41	Niobium	92.90638	8.57	2477	4744	[Kr] 4d ⁴ 5s	6D _{1/2}	6.759	0.265
42	Molybdenum	95.94	10.22	2623	4639	[Kr] 4d ⁵ 5s	7S ₃	7.092	0.251
43	Technetium	(98)	11.50 ^a	2157	4265	[Kr] 4d ⁵ 5s ²	6S _{5/2}	7.28	—
44	Ruthenium	101.07	12.41	2334	4150	[Kr] 4d ⁷ 5s	5F ₅	7.360	0.238
45	Rhodium	102.90550	12.41	1964	3695	[Kr] 4d ⁸ 5s	4F _{9/2}	7.459	0.243
46	Palladium	106.42	12.02	1554.9	2963	[Kr] 4d ¹⁰	1S ₀	8.337	0.246
47	Silver	107.8682	10.50	961.78	2162	[Kr] 4d ¹⁰ 5s	2S _{1/2}	7.576	0.235
48	Cadmium	112.411	8.65	321.07	767	[Kr] 4d ¹⁰ 5s ²	1S ₀	8.994	0.232
49	Indium	114.818	7.31	156.60	2072	[Kr] 4d ¹⁰ 5s ² 5p	2P _{1/2}	5.786	0.233
50	Tin	118.710	7.31	231.93	2602	[Kr] 4d ¹⁰ 5s ² 5p ²	3P ₀	7.344	0.228
51	Antimony	121.760	6.691	630.73	1587	[Kr] 4d ¹⁰ 5s ² 5p ³	4S _{3/2}	8.608	0.207
52	Tellurium	127.60	6.24	449.51	988	[Kr] 4d ¹⁰ 5s ² 5p ⁴	3P ₂	9.010	0.202
53	Iodine	126.90447	4.93	113.7	184.4	[Kr] 4d ¹⁰ 5s ² 5p ⁵	2P _{3/2}	10.451	0.145
54	Xenon	131.29	3.52	-111.798 ^{1,6} kPa	-108.12	[Kr] 4d ¹⁰ 5s ² 5p ⁶	1S ₀	12.130	0.158

Table 5-2. Properties of the elements (continued).

Z	Element	Atomic weight	Density	Melting point (°C)	Boiling point (°C)	Ground-state configuration	Ground level	Ionization energy (eV)	Specific heat (J/g·K)
55	Cesium	132.90545	1.873	28.5	671	[Xe]	2S _{1/2}	3.894	0.242
56	Barium	137.327	3.5	727	1897	[Xe]	1S ₀	5.212	0.204
57	Lanthanum	138.9055	6.145 ²⁵	918	3464	[Xe] 5d	2D _{3/2}	5.577	0.195
58	Cerium	140.116	6.770 ²⁵	798	3443	[Xe] 4f 5d	1G ^o 4	5.539	0.192
59	Praseodymium	140.90765	6.773	931	3520	[Xe] 4f ³	4F ^o 9/2	5.473	0.193
60	Neodymium	144.24	7.008 ²⁵	1021	3074	[Xe] 4f ⁴	5I ₄	5.525	0.190
61	Promethium	(145)	7.264 ²⁵	1042	3000	[Xe] 4f ⁵	6H ^o 5/2	5.582	—
62	Samarium	150.36	7.520 ²⁵	1074	1794	[Xe] 4f ⁶	7F ₀	5.644	0.197
63	Europium	151.964	5.244 ²⁵	822	1529	[Xe] 4f ⁷	8S ^o 7/2	5.670	0.182
64	Gadolinium	157.25	7.901 ²⁵	1313	3273	[Xe] 4f ⁷ 5d	9D ^o 2	6.150	0.236
65	Terbium	158.92534	8.230	1356	3230	[Xe] 4f ⁹	6H ^o 15/2	5.864	0.182
66	Dysprosium	162.50	8.551 ²⁵	1412	2567	[Xe] 4f ¹⁰	5I ₈	5.939	0.170
67	Holmium	164.93032	8.795 ²⁵	1474	2700	[Xe] 4f ¹¹	4I ^o 15/2	6.022	0.165
68	Erbium	167.26	9.066 ²⁵	1529	2868	[Xe] 4f ¹²	3H ₆	6.108	0.168
69	Thulium	168.93421	9.321 ²⁵	1545	1950	[Xe] 4f ¹³	2F ^o 7/2	6.184	0.160
70	Ytterbium	173.04	6.966	819	1196	[Xe] 4f ¹⁴	1S ₀	6.254	0.155
71	Lutetium	174.967	9.841 ²⁵	1663	3402	[Xe] 4f ¹⁴ 5d	2D _{3/2}	5.426	0.154
72	Hafnium	178.49	13.31	2233	4603	[Xe] 4f ¹⁴ 5d ²	3F ₂	6.825	0.144
73	Tantalum	180.9479	16.654	3017	5458	[Xe] 4f ¹⁴ 5d ³	4F _{3/2}	7.550	0.140
74	Tungsten	183.84	19.3	3422	5555	[Xe] 4f ¹⁴ 5d ⁴	5D ₀	7.864	0.132
75	Rhenium	186.207	21.02	3186	5596	[Xe] 4f ¹⁴ 5d ⁵	6S _{5/2}	7.834	0.137
76	Osmium	190.23	22.57	3033	5012	[Xe] 4f ¹⁴ 5d ⁶	5D ₄	8.438	0.130
77	Iridium	192.217	22.42 ¹⁷	2446	4428	[Xe] 4f ¹⁴ 5d ⁷	4F _{9/2}	8.967	0.131
78	Platinum	195.078	21.45	1768.4	3825	[Xe] 4f ¹⁴ 5d ⁸	3D ₃	8.959	0.133
79	Gold	196.96655	~19.3	1064.18	2856	[Xe] 4f ¹⁴ 5d ⁹	2S _{1/2}	9.226	0.129
80	Mercury	200.59	13.546	-38.83	356.73	[Xe] 4f ¹⁴ 5d ¹⁰	1S ₀	10.438	0.140
81	Thallium	204.3833	11.85	304	1473	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ² 6p	2P ^o 1/2	6.108	0.129
82	Lead	207.2	11.35	327.46	1749	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ² 6p ²	3P ₀	7.417	0.129

Table 5-2. Properties of the elements (continued).

Z	Element	Atomic weight	Density	Melting point (°C)	Boiling point (°C)	Ground-state configuration	Ground level	Ionization energy (eV)	Specific heat (J/g·K)
83	Bismuth	208.98038	9.747	271.40	1564	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ² 6p ³	4s ^o 3/2	7.286	0.122
84	Polonium	(209)	9.32	254	962	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ² 6p ⁴	3p ₂	8.417 ?	—
85	Astatine	(210)	—	302	—	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ² 6p ⁵	2p ^o 3/2	—	—
86	Radon	(222)	—	-71	-61.7	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ² 6p ⁶	1s ₀	10.748	0.094
87	Francium	(223)	—	27	—	[Rn] 7s	2s _{1/2}	4.073	—
88	Radium	(226)	—	700	—	[Rn] 7s ²	1s ₀	5.278	—
89	Actinium	(227)	—	1051	3198	[Rn] 6d 7s ²	2d _{3/2}	5.17	0.120
90	Thorium	232.0381	11.72	1750	4788	[Rn] 6d ² 7s ²	3f ₂	6.307	0.113
91	Protactinium	231.03588	15.37 ^a	1572	—	[Rn] 5f ² (3H ₄) 6d 7s ²	(4, 3 ^{1/2}) 1/2	5.89	—
92	Uranium	238.0289	~18.95	1135	4131	[Rn] 5f ³ (4f ^o 9/2) 6d 7s ²	(9/2, 3 ^{1/2}) ^o 6	6.194	0.116
93	Neptunium	(237)	20.25	644	—	[Rn] 5f ⁴ (5 ^{1/4}) 6d 7s ²	(4, 3 ^{1/2}) 1/2	6.266	—
94	Plutonium	(244)	19.84 ²⁵	640	3228	[Rn] 5f ⁶ 7s ²	7f ₀	6.026	—
95	Americium	(243)	13.67	1176	2011	[Rn] 5f ⁷ 7s ²	8s ^o 7/2	5.974	—
96	Curium	(247)	13.51 ^a	1345	3100	[Rn] 5f ⁷ 6d 7s ²	9d ^o 2	5.992	—
97	Berkelium	(247)	14 (est.)	1050	—	[Rn] 5f ⁹ 7s ²	6h ^o 15/2	6.198	—
98	Californium	(251)	—	900	—	[Rn] 5f ¹⁰ 7s ²	5l ₈	6.282	—
99	Einsteinium	(252)	—	860	—	[Rn] 5f ¹¹ 7s ²	4f ^o 15/2	6.42	—
100	Fermium	(257)	—	1527	—	[Rn] 5f ¹² 7s ²	3H ₆	6.50	—
101	Mendelevium	(258)	—	827	—	[Rn] 5f ¹³ 7s ²	2f ^o 7/2	6.58	—
102	Nobelium	(259)	—	827	—	[Rn] 5f ¹⁴ 7s ²	1s ₀	6.65	—
103	Lawrencium	(262)	—	1627	—	[Rn] 5f ¹⁴ 7s ² 7p ?	2p ^o 1/2 ?	4.9 ?	—
104	Rutherfordium	(261)	—	—	—	[Rn] 5f ¹⁴ 6d ² 7s ² ?	3f ₂ ?	6.0 ?	—
105	Dubnium	(262)	—	—	—	—	—	—	—
106	Seaborgium	(266)	—	—	—	—	—	—	—
107	Bohrium	(264)	—	—	—	—	—	—	—
108	Hassium	(269)	—	—	—	—	—	—	—
109	Meitnerium	(268)	—	—	—	—	—	—	—

^aCalculated ^bSublimes

Periodic Table of the Elements

1 H 1.01																	2 He 4.00				
3 Li 6.94	4 Be 9.01															5 B 10.81	6 C 12.01	7 N 14.01	8 O 16.00	9 F 19.00	10 Ne 20.18
11 Na 22.99	12 Mg 24.31															13 Al 26.98	14 Si 28.09	15 P 30.97	16 S 32.07	17 Cl 35.45	18 Ar 39.95
19 K 39.10	20 Ca 40.08	21 Sc 44.96	22 Ti 47.87	23 V 50.94	24 Cr 52.00	25 Mn 54.94	26 Fe 55.85	27 Co 58.93	28 Ni 58.69	29 Cu 63.55	30 Zn 65.39	31 Ga 69.72	32 Ge 72.61	33 As 74.92	34 Se 78.96	35 Br 79.90	36 Kr 83.80				
37 Rb 85.47	38 Sr 87.62	39 Y 88.91	40 Zr 91.22	41 Nb 92.91	42 Mo 95.94	43 Tc (98)	44 Ru 101.07	45 Rh 102.91	46 Pd 106.42	47 Ag 107.87	48 Cd 112.41	49 In 114.82	50 Sn 118.71	51 Sb 121.76	52 Te 127.60	53 I 126.90	54 Xe 131.29				
55 Cs 132.91	56 Ba 137.33	57 La 138.91	72 Zr 91.22	73 Nb 92.91	74 Mo 95.94	75 Tc (98)	76 Ru 101.07	77 Rh 102.91	78 Pd 106.42	79 Ag 107.87	80 Cd 112.41	81 In 114.82	82 Sn 118.71	83 Sb 121.76	84 Te 127.60	85 I 126.90	86 Xe 131.29				
87 Fr (223)	88 Ra (226)	89 Ac (227)	104 Rf (261)	105 Db (262)	106 Sg (265)	107 Bh (264)	108 Hs (269)	109 Mt (268)	110 Rf (271)	111 Db (272)	112 Cn (283)	113 Nh (284)	114 Fl (287)	115 Mc (288)	116 Lv (292)	117 Ts (293)	118 Og (294)				
Lanthanide series ▶																					
58 Ce 140.12	59 Pr 140.91	60 Nd 144.24	61 Pm (145)	62 Sm 150.36	63 Eu 151.96	64 Gd 157.25	65 Tb 158.93	66 Dy 162.50	67 Ho 164.93	68 Er 167.26	69 Tm 168.93	70 Yb 173.04	71 Lu 174.97								
Actinide series ▶																					
90 Th 232.04	91 Pa 231.04	92 U 238.03	93 Np (237)	94 Pu (244)	95 Am (243)	96 Cm (247)	97 Bk (247)	98 Cf (251)	99 Es (252)	100 Fm (257)	101 Md (258)	102 No (259)	103 Lr (262)								

5.3 ELECTROMAGNETIC RELATIONS

	Gaussian CGS	SI
Units and conversions:		
Charge	$2.997\,92 \times 10^9$ esu	$= 1\text{ C} = 1\text{ A s}$
Potential	$(1/299.792)$ statvolt $= (1/299.792)$ erg/esu	$= 1\text{ V} = 1\text{ J C}^{-1}$
Magnetic field	10^4 gauss $= 10^4$ dyne/esu	$= 1\text{ T} = 1\text{ N A}^{-1}\text{ m}^{-1}$
Electron charge	$e = 4.803\,204 \times 10^{-10}$ esu	$= 1.602\,176 \times 10^{-19}\text{ C}$
Lorentz force	$\mathbf{F} = q\left(\mathbf{E} + \frac{\mathbf{v}}{c} \times \mathbf{B}\right)$	$\mathbf{F} = q(\mathbf{E} + \mathbf{v} \times \mathbf{B})$
Maxwell equations	$\nabla \cdot \mathbf{D} = 4\pi\rho$ $\nabla \times \mathbf{E} + \frac{1}{c} \frac{\partial \mathbf{B}}{\partial t} = 0$ $\nabla \cdot \mathbf{B} = 0$ $\nabla \times \mathbf{H} - \frac{1}{c} \frac{\partial \mathbf{D}}{\partial t} = \frac{4\pi}{c} \mathbf{J}$	$\nabla \cdot \mathbf{D} = \rho$ $\nabla \times \mathbf{E} + \frac{\partial \mathbf{B}}{\partial t} = 0$ $\nabla \cdot \mathbf{B} = 0$ $\nabla \times \mathbf{H} - \frac{\partial \mathbf{D}}{\partial t} = \mathbf{J}$
Linear media	$\mathbf{D} = \epsilon \mathbf{E}, \mathbf{B} = \mu \mathbf{H}$	$\mathbf{D} = \epsilon \mathbf{E}, \mathbf{B} = \mu \mathbf{H}$
Permittivity of free space	$\epsilon_{\text{vac}} = 1$	$\epsilon_{\text{vac}} = \epsilon_0$
Permeability of free space	$\mu_{\text{vac}} = 1$	$\mu_{\text{vac}} = \mu_0$
Fields from potentials	$\mathbf{E} = -\nabla V - \frac{1}{c} \frac{\partial \mathbf{A}}{\partial t}$ $\mathbf{B} = \nabla \times \mathbf{A}$	$\mathbf{E} = -\nabla V - \frac{\partial \mathbf{A}}{\partial t}$ $\mathbf{B} = \nabla \times \mathbf{A}$
Static potentials (coulomb gauge)	$V = \sum_{\text{charges}} \frac{q_i}{r_i}$ $\mathbf{A} = \frac{1}{c} \oint \frac{I \mathbf{d} \ell}{ \mathbf{r} - \mathbf{r}' }$	$V = \frac{1}{4\pi\epsilon_0} \sum_{\text{charges}} \frac{q_i}{r_i}$ $\mathbf{A} = \frac{\mu_0}{4\pi} \oint \frac{I \mathbf{d} \ell}{ \mathbf{r} - \mathbf{r}' }$
Relativistic transformations (\mathbf{v} is the velocity of primed system as seen in unprimed system)	$\mathbf{E}'_{\parallel} = \mathbf{E}_{\parallel}$ $\mathbf{E}'_{\perp} = \gamma\left(\mathbf{E}_{\perp} + \frac{1}{c} \mathbf{v} \times \mathbf{B}\right)$ $\mathbf{B}'_{\parallel} = \mathbf{B}_{\parallel}$ $\mathbf{B}'_{\perp} = \gamma\left(\mathbf{B}_{\perp} \pm \frac{1}{c} \mathbf{v} \times \mathbf{E}\right)$	$\mathbf{E}'_{\parallel} = \mathbf{E}_{\parallel}$ $\mathbf{E}'_{\perp} = \gamma(\mathbf{E}_{\perp} + \mathbf{v} \times \mathbf{B})$ $\mathbf{B}'_{\parallel} = \mathbf{B}_{\parallel}$ $\mathbf{B}'_{\perp} = \gamma\left(\mathbf{B}_{\perp} - \frac{1}{c^2} \mathbf{v} \times \mathbf{E}\right)$

$$4\pi\epsilon_0 = \frac{1}{c^2} 10^7 \text{ A}^2 \text{ N}^{-1} = \frac{1}{8.987\,55\dots} \times 10^{-9} \text{ F m}^{-1}$$

$$\frac{\mu_0}{4\pi} = 10^{-7} \text{ N A}^{-1}; c = 2.997\,924\,58 \times 10^8 \text{ m s}^{-1}$$

Impedances (SI units)

ρ = resistivity at room temperature in $10^{-8} \Omega \text{ m}$:

~ 1.7 for Cu

~ 5.5 for W

~ 2.4 for Au

~ 73 for SS 304

~ 2.8 for Al

~ 100 for Nichrome

(Al alloys may have
double this value.)

For alternating currents, instantaneous current I , voltage V ,
angular frequency ω :

$$V = V_0 e^{j\omega t} = ZI .$$

Impedance of self-inductance L : $Z = j\omega L$.

Impedance of capacitance C : $Z = 1/j\omega C$.

Impedance of free space: $Z = \sqrt{\mu_0 / \epsilon_0} = 376.7 \Omega$.

High-frequency surface impedance of a good conductor:

$$Z = \frac{(1 + j)\rho}{\delta} , \text{ where } \delta = \text{effective skin depth} ;$$

$$\delta = \sqrt{\frac{\rho}{\pi\nu\mu}} \cong \frac{6.6 \text{ cm}}{\sqrt{\nu[\text{Hz}]}} \text{ for Cu} .$$

Capacitance \hat{C} and inductance \hat{L} per unit length (SI units)

Flat rectangular plates of width w , separated by $d \ll w$ with
linear medium (ϵ, μ) between:

$$\hat{C} = \epsilon \frac{w}{d} ; \quad \hat{L} = \mu \frac{d}{w} ;$$

$$\frac{\epsilon}{\epsilon_0} = 2 \text{ to } 6 \text{ for plastics; } 4 \text{ to } 8 \text{ for porcelain, glasses;}$$

$$\frac{\mu}{\mu_0} \cong 1 .$$

Coaxial cable of inner radius r_1 , outer radius r_2 :

$$\hat{C} = \frac{2\pi\epsilon}{\ln(r_2/r_1)} ; \quad \hat{L} = \frac{\mu}{2\pi} \ln(r_2/r_1) .$$

Transmission lines (no loss):

$$\text{Impedance: } Z = \sqrt{\hat{L}/\hat{C}} .$$

$$\text{Velocity: } v = 1/\sqrt{\hat{L}\hat{C}} = 1/\sqrt{\mu\epsilon} .$$

Motion of charged particles in a uniform, static magnetic field

The path of motion of a charged particle of momentum p is a helix of constant radius R and constant pitch angle λ , with the axis of the helix along \mathbf{B} :

$$p[\text{GeV}/c]\cos\lambda = 0.29979 qB[\text{tesla}] R[\text{m}] ,$$

where the charge q is in units of the electronic charge. The angular velocity about the axis of the helix is

$$\omega[\text{rad s}^{-1}] = 8.98755 \times 10^7 qB[\text{tesla}] / E[\text{GeV}] ,$$

where E is the energy of the particle.

This section was adapted, with permission, from the 1999 web edition of the *Review of Particle Physics* (<http://pdg.lbl.gov>). See J. D. Jackson, *Classical Electrodynamics*, 2d ed. (John Wiley & Sons, New York, 1975) for more formulas and details.

5.4 RADIOACTIVITY AND RADIATION PROTECTION

The International Commission on Radiation Units and Measurements (ICRU) recommends the use of SI units. Therefore, we list SI units first, followed by cgs (or other common) units in parentheses, where they differ.

A. DEFINITIONS

Unit of activity = becquerel (curie):

$$1 \text{ Bq} = 1 \text{ disintegration s}^{-1} [= 1/(3.7 \times 10^{10}) \text{ Ci}]$$

Unit of absorbed dose = gray (rad):

$$1 \text{ Gy} = 1 \text{ J kg}^{-1} (= 10^4 \text{ erg g}^{-1} = 100 \text{ rad})$$

$$= 6.24 \times 10^{12} \text{ MeV kg}^{-1} \text{ deposited energy}$$

Unit of exposure, the quantity of x- or γ -radiation at a point in space integrated over time, in terms of charge of either sign produced by showering electrons in a small volume of air about the point:

$$= 1 \text{ C kg}^{-1} \text{ of air (roentgen; } 1 \text{ R} = 2.58 \times 10^{-4} \text{ C kg}^{-1})$$

$$= 1 \text{ esu cm}^{-3} (= 87.8 \text{ erg released energy per g of air})$$

Implicit in the definition is the assumption that the small test volume is embedded in a sufficiently large uniformly irradiated volume that the number of secondary electrons entering the volume equals the number leaving.

Unit of equivalent dose for biological damage = sievert.

1 Sv = 100 rem (roentgen equivalent for man). The equivalent dose in Sv = absorbed dose in grays $\times w_R$, where w_R is the radiation weighting factor (formerly the quality factor Q), which depends upon the type of radiation and other factors, as shown in Table 5-3. The equivalent dose expresses the long-term risk (primarily due to cancer and leukemia) from low-level chronic exposure.

B. RADIATION LEVELS

Natural annual background from all sources. In most of the world, the whole-body equivalent dose rate $\approx 0.4\text{--}4$ mSv (40–400 mrem). It can range up to 50 mSv (5 rem) in certain areas. The U.S. average ≈ 3.6 mSv, including about 2 mSv (≈ 200 mrem) from inhaled natural radioactivity, mostly

Table 5-3. Radiation weighting factors.

Type of radiation	w_R
X- and γ -rays, all energies	1
Electrons and muons, all energies	1
Neutrons:	
< 10 keV	5
10–100 keV	10
0.1–2 MeV	20
2–20 MeV	10
> 20 MeV	5
Protons (other than recoils), > 2 MeV	5
Alphas, fission fragments, and heavy nuclei	20

radon and radon daughters. This radon exposure value is for a typical house; radon exposure varies by more than an order of magnitude.

Cosmic ray background in counters (Earth's surface):

$\sim 1 \text{ min}^{-1} \text{ cm}^{-2} \text{ sr}^{-1}$.

Man-made radiation dose: The greatest contribution to man-made radiation dose has been from irradiation from x-ray diagnostics in medicine, which accounts for about 20% of the average natural radiation dose.

Fluxes (per cm^2) to deposit one Gy, assuming uniform irradiation:

For photons:

$\approx 6.24 \times 10^9 \lambda E f$, for photons of energy E [MeV],

attenuation length λ (g cm^{-2}), and fraction $f \leq 1$

expressing the fraction of the photon's energy deposited in a small volume of thickness $\ll \lambda$ but large enough to contain the secondary electrons.

$\approx 2 \times 10^{11}$ photons cm^{-2} for 1-MeV photons on carbon ($f \approx 0.5$).

For charged particles:

$\approx 6.24 \times 10^9 / (dE/dx)$; where dE/dx [$\text{MeV g}^{-1} \text{ cm}^2$], the energy loss per unit length, may be obtained from range-energy figures.

$\approx 3.5 \times 10^9 \text{ cm}^{-2}$ for minimum-ionizing singly-charged particles in carbon.

Quoted fluxes are good to about a factor of two for all materials.

Recommended exposure limits for radiation workers (whole-body dose):

ICRP: 20 mSv yr^{-1} averaged over 5 years, with the dose in any one year $\leq 50 \text{ mSv}$.

U.S.: 50 mSv yr^{-1} (5 rem yr^{-1}). Many laboratories in the U.S. and elsewhere set lower limits.

Lethal dose: Whole-body dose from penetrating ionizing radiation resulting in 50% mortality in 30 days (assuming no medical treatment), about 5 Gy (500 rads) as measured internally on body longitudinal center line. Surface dose varies owing to variable body attenuation and may be a strong function of energy.

This section was adapted, with permission, from the 1999 web edition of the *Review of Particle Physics* (<http://pdg.lbl.gov>). For further information, see ICRP Publication 60, *1990 Recommendation of the International Commission on Radiological Protection* (Pergamon Press, New York, 1991) and E. Pochin, *Nuclear Radiation: Risks and Benefits* (Clarendon Press, Oxford, 1983).

5.5 USEFUL EQUATIONS

The following pages include a number of equations useful to x-ray scientists, either expanding on subjects covered in this booklet or addressing topics not covered here. The equations have been drawn from D. T. Attwood, *Soft X-Rays and Extreme Ultraviolet Radiation: Principles and Applications* (Cambridge Univ. Press, Cambridge, 1999) [www.coe.berkeley.edu/AST/sxreuv], and the equation numbers refer to that volume, which should be consulted for further explanation and discussion. That reference also expands on the discussions in this booklet on zone plate optics, synchrotron radiation, and other topics.

General X-Ray Formulas

Wavelength and photon energy relationship:

$$h\omega \cdot \lambda = hc = 1239.842 \text{ eV} \cdot \text{nm} \quad (1.1)$$

Number of photons required for 1 joule of energy:

$$1 \text{ joule} \Rightarrow 5.034 \times 10^{15} \lambda \text{ [nm] photons} \quad (1.2a)$$

X-Ray Scattering and Absorption

Thomson cross section for a free electron:

$$\sigma_e = \frac{8\pi}{3} r_e^2 \quad (2.45)$$

$$r_e = \frac{e^2}{4\pi\epsilon_0 mc^2} = 2.82 \times 10^{-13} \text{ cm} \quad (2.44)$$

and r_e is the classical electron radius.

Scattering cross section for a bound electron:

$$\sigma = \frac{8\pi}{3} r_e^2 \frac{\omega^4}{(\omega^2 - \omega_s^2)^2 + (\gamma\omega)^4} \quad (2.51)$$

Rayleigh cross section ($\omega^2 \ll \omega_s^2$):

$$\sigma_R = \frac{8\pi}{3} r_e^2 \left(\frac{\omega}{\omega_s}\right)^4 = \frac{8\pi}{3} r_e^2 \left(\frac{\lambda_s}{\lambda}\right)^4 \quad (2.52)$$

Scattering by a multi-electron atom:

$$\frac{d\sigma(\omega)}{d\Omega} = r_e^2 |f|^2 \sin^2 \Theta \quad (2.68)$$

$$\sigma(\omega) = \frac{8\pi}{3} |f|^2 r_e^2 \quad (2.69)$$

where the complex atomic scattering factor represents the electric field scattered by an atom, normalized to that of a single electron:

$$f(\Delta\mathbf{k}, \omega) = \sum_{s=1}^Z \frac{\omega_s^2 e^{-i\Delta\mathbf{k} \cdot \Delta\mathbf{r}_s}}{(\omega^2 - \omega_s^2 + i\gamma\omega)} \quad (2.66)$$

For forward scattering or long wavelength this reduces to

$$f^0(\omega) = \sum_{s=1}^Z \frac{\omega^2}{(\omega^2 - \omega_s^2 + i\gamma\omega)} = f_1^0 - if_2^0 \quad (2.72 \text{ \& } 2.79)$$

Refractive index for x-ray radiation is commonly written * as

$$n(\omega) = 1 - \delta + i\beta = 1 - \frac{n_a r_e \lambda^2}{2\pi} (f_1^0 - if_2^0) \quad (3.9 \text{ \& } 3.12)$$

where

$$\delta = \frac{n_a r_e \lambda^2}{2\pi} f_1^0(\omega) \quad (3.13a)$$

$$\beta = \frac{n_a r_e \lambda^2}{2\pi} f_2^0(\omega) \quad (3.13b)$$

Absorption length in a material:

$$\ell_{\text{abs}} = \frac{\lambda}{4\pi\beta} = \frac{1}{2n_a r_e \lambda f_2^0(\omega)} \quad (3.22 \text{ \& } 3.23)$$

Mass-dependent absorption coefficient:

$$\mu = \frac{2r_e \lambda}{Am_u} f_2^0(\omega) \quad (3.26)$$

Atomic absorption cross section:

$$\sigma_{\text{abs}} = 2r_e \lambda f_2^0(\omega) = Am_u \mu(\omega) \quad (3.28a \text{ \& } b)$$

Relative phase shift through a medium compared to a vacuum:

$$\Delta\phi = \left(\frac{2\pi\delta}{\lambda} \right) \Delta r \quad (3.29)$$

where Δr is the thickness or propagation distance.

* The choice of $+i\beta$ is consistent with a wave description $E = E_0 \exp[-i(\omega t - kr)]$. A choice of $-i\beta$ is consistent with $E = E_0 \exp[i(\omega t - kr)]$.

Snell's law:

$$\sin \phi' = \frac{\sin \phi}{n} \quad (3.38)$$

Critical angle for total external reflection of x-rays:

$$\theta_c = \sqrt{2\bar{\delta}} \quad (3.41)$$

$$\theta_c = \sqrt{2\bar{\delta}} = \sqrt{\frac{n_a r_e \lambda^2 f_1^0(\lambda)}{\pi}} \quad (3.42a)$$

Brewster's angle (or polarizing angle):

$$\phi_B \simeq \frac{\pi}{4} - \frac{\delta}{2} \quad (3.60)$$

Multilayer Mirrors

Bragg's law:

$$m\lambda = 2d \sin \theta \quad (4.6b)$$

Correction for refraction:

$$m\lambda = 2d \sin \theta \sqrt{1 - \frac{2\bar{\delta}}{\sin^2 \theta}} = 2d \sin \theta \left(1 - \frac{4\bar{\delta}d^2}{m^2\lambda^2} \right)$$

where $\bar{\delta}$ is the period-averaged real part of the refractive index.

$$\Gamma = \frac{\Delta t_H}{\Delta t_H + \Delta t_L} = \frac{\Delta t_H}{d} \quad (4.7)$$

Plasma Equations

Electron plasma frequency:

$$\omega_p^2 = \frac{e^2 n_e}{\epsilon_0 m} \quad (6.5)$$

Debye screening distance:

$$\lambda_D = \left(\frac{\epsilon_0 \kappa T_e}{e^2 n_e} \right)^{1/2} \quad (6.6)$$

No. of electrons in Debye sphere:

$$N_D = \frac{4\pi}{3} \lambda_D^3 n_e \quad (6.7)$$

Electron cyclotron frequency:

$$\omega_c = \frac{eB}{m} \quad (6.8)$$

Maxwellian velocity distribution for electrons characterized by a single-electron temperature κT_e :

$$f(v) = \frac{1}{(2\pi)^{3/2} v_e^3} e^{-v^2/2v_e^2} \quad (6.1)$$

where

$$v_e = \left(\frac{\kappa T_e}{m} \right)^{1/2} \quad (6.2)$$

Electron sound speed:

$$a_e = \left(\frac{\gamma \kappa T_e}{m} \right)^{1/2} \quad (6.79)$$

Critical electron density:

$$n_c \equiv \frac{\epsilon_0 m \omega^2}{e^2} = 1.11 \times 10^{21} \frac{\text{e/cm}^3}{\lambda^2 (\mu\text{m})} \quad (6.112a \& b)$$

Refractive index of plasma is

$$n = \sqrt{1 - \frac{n_e}{n_c}} \quad (6.114b)$$

Ratio of electron energy in coherent oscillations to that in random motion:

$$\left| \frac{v_{os}}{v_e} \right|^2 = \frac{e^2 E^2}{m\omega^2 \kappa T_e} = \frac{I/c}{n_c \kappa T_e} \quad (6.131a)$$

$$\left| \frac{v_{os}}{v_e} \right|^2 = \frac{0.021 I (10^{14} \text{ W/cm}^2) \lambda^2 (\mu\text{m})}{\kappa T_e (\text{keV})} \quad (6.131b)$$

Spectral brightness of blackbody radiation within $\Delta\omega/\omega = 0.1\% \text{BW}$:

$$B_{\Delta\omega/\omega} = 3.146 \times 10^{11} \left(\frac{\kappa T}{\text{eV}} \right)^3 \frac{(\hbar\omega/\kappa T)^3 \text{ photons/sec}}{(e^{\hbar\omega/\kappa T} - 1) (\text{mm})^2 (\text{mr})^2 (0.1\% \text{BW})} \quad (6.136b)$$

Photon energy at peak spectral brightness:

$$\hbar\omega|_{pk} = 2.822 \kappa T \quad (6.137)$$

where κ is the Boltzmann constant.

Stefan-Boltzmann radiation law (blackbody intensity at any interface):

$$I = \sigma T^4 \quad (6.141b)$$

where the Stefan-Boltzmann constant is

$$\sigma = \frac{\pi^2 \kappa^4}{60c^2 \hbar^3} \quad (6.142)$$

With κT in eV:

$$I = \hat{\sigma} (\kappa T)^4 \quad (6.143a)$$

where $\hat{\sigma}$ is the modified Stefan-Boltzmann constant

$$\hat{\sigma} = \frac{\pi^2}{60 \hbar^3 c^2} = 1.027 \times 10^5 \frac{\text{watts}}{\text{cm}^2 (\text{eV})^4} \quad (6.143b)$$

Coherence

Longitudinal coherence length:*

$$\ell_{\text{coh}} = \lambda^2/2\Delta\lambda \quad (8.3)$$

Spatial or transverse coherence (rms quantities):

$$d \cdot \theta = \lambda/2\pi \quad (8.5)$$

or in terms of FWHM values

$$d \cdot 2\theta|_{\text{FWHM}} = 0.44\lambda$$

Spatially coherent power within a relative spectral bandwidth $\lambda/\Delta\lambda = N$ for an undulator with N periods:

$$\bar{P}_{\text{coh},N} = \frac{e\lambda_u I}{8\pi\epsilon_0 d_x d_y \theta_x \theta_y \gamma^2} \cdot \left(\frac{\hbar\omega_0}{\hbar\omega} - 1 \right) f(\hbar\omega/\hbar\omega_0) \quad (8.7c)$$

where $\hbar\omega_0$ corresponds to $K = 0$, and where

$$f(\hbar\omega/\hbar\omega_0) = \frac{7}{16} + \frac{5}{8} \frac{\hbar\omega}{\hbar\omega_0} - \frac{1}{16} \left(\frac{\hbar\omega}{\hbar\omega_0} \right)^2 + \dots \quad (8.8)$$

When the undulator condition ($\sigma' \ll \theta_{\text{cen}}$) is satisfied, the coherent power within a relative spectral bandwidth $\Delta\lambda/\lambda < 1/N$, is

$$\bar{P}_{\text{coh},\lambda/\Delta\lambda} = \frac{e\lambda_u I \eta (\lambda/\Delta\lambda) N^2}{8\pi\epsilon_0 d_x d_y} \cdot \left(1 - \frac{\hbar\omega}{\hbar\omega_0} \right) f(\hbar\omega/\hbar\omega_0) \quad (8.10c)$$

$(\sigma'^2 \ll \theta_{\text{cen}}^2)$

where η is the combined beamline and monochrometer efficiency.

* The factor of two here is somewhat arbitrary and depends, in part, on the definition of $\Delta\lambda$. Equation (26) on page 2-14 omits this factor. See Attwood, op.cit., for further discussion.

Spatially coherent power available from a laser is

$$P_{\text{coh}} = \frac{(\lambda/2\pi)^2}{(d_x\theta_x)(d_y\theta_y)} P_{\text{laser}} \quad (8.11)$$

where P_{laser} is the total laser power.

Normalized degree of spatial coherence, or complex coherence factor:

$$\mu_{12} = \frac{\langle E_1(t)E_2^*(t) \rangle}{\sqrt{\langle |E_1|^2 \rangle} \sqrt{\langle |E_2|^2 \rangle}} \quad (8.12)$$

The van Cittert-Zernike theorem for the complex coherence factor is

$$\mu_{\text{OP}} = \frac{e^{-i\psi} \iint I(\xi, \eta) e^{ik(\xi\theta_x + \eta\theta_y)} d\xi d\eta}{\iint I(\xi, \eta) d\xi d\eta} \quad (8.19)$$

For a uniformly but incoherently illuminated pinhole

$$\mu_{\text{OP}}(\theta) = e^{-i\psi} \frac{2J_1(ka\theta)}{(ka\theta)} \quad (8.27)$$

which has its first null ($\mu_{\text{OP}} = 0$) at $ka\theta = 3.832$, which for $d = 2a$ corresponds to $d \cdot \theta = 1.22\lambda$.

EUV/Soft X-Ray Lasers

Growth of stimulated emission:

$$\frac{I}{I_0} = e^{GL} \quad (7.2)$$

where L is the laser length and G is the gain per unit length. For an upper-state ion density n_u and a density inversion factor $F(\leq 1)$

$$G = n_u \sigma_{\text{stim}} F \quad (7.4)$$

where the cross section for stimulated emission is

$$\sigma_{\text{stim}} = \frac{\lambda^3 A_{u\ell}}{8\pi c (\Delta\lambda/\lambda)} \quad (7.16)$$

$$\sigma_{\text{stim}} = \frac{\pi \lambda r_e}{(\Delta\lambda/\lambda)} \left(\frac{g_\ell}{g_u} \right) f_{\ell u} \quad (7.18)$$

where $A_{u\ell}$ is the spontaneous decay rate, $f_{\ell u}$ is the oscillator strength and g_ℓ/g_u is the ratio of degeneracy factors.

Laser wavelength scaling goes as $1/\lambda^4$:

$$\frac{P}{A} = \frac{16\pi^2 c^2 \hbar (\Delta\lambda/\lambda) GL}{\lambda^4} \quad (7.22)$$

Doppler-broadened linewidth:

$$\left. \frac{(\Delta\lambda)}{\lambda} \right|_{\text{FWHM}} = \frac{v_i}{c} = \frac{2\sqrt{2 \ln 2}}{c} \sqrt{\frac{\kappa T_i}{M}} \quad (7.19a)$$

where v_i is the ion thermal velocity, κT_i is the ion temperature, and M is the ion mass. With κT_i expressed in eV and an ion mass of $2m_p Z$

$$\left. \frac{(\Delta\lambda)}{\lambda} \right|_{\text{FWHM}} = 7.68 \times 10^{-5} \left(\frac{\kappa T_i}{2Z} \right)^{1/2} \quad (7.19b)$$

Lithography

Minimum printable line width:

$$L_w = k_1 \frac{\lambda}{NA} \quad (10.1)$$

where k_1 is a constant dominated by the optical system, but affected by pattern transfer processes.

Depth of focus:

$$DOF = \pm k_2 \frac{\lambda}{(NA)^2} \quad (10.2)$$

Degree of partial coherence:

$$\sigma = \frac{NA_{cond}}{NA_{obj}} \quad (10.3)$$

where the subscript cond refers to the condenser or illumination optics, and obj refers to the objective lens of the reduction optics.

International Technology Road Map for Semiconductors

	Years			
	2005	2008	2011	2014
1:1 lines (nm)	100	70	50	35
Isolated lines (nm)	65	45	30	20

<http://www.sematech.org>



Additional information about
light sources around the world
can be found at
<http://lightsources.org>